



# Practical XRD with Confidence



Episode 1 – Powder XRD Fundamentals:  
How to Identify Phases and Trust Your Data

**Wednesday, May 6, 2026, at 1 pm CDT**

**Presenter:** Ekaterina Vinogradova, PhD

**Host:** Tom Concolino, PhD





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How to Identify Phases and Trust Your Data

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- *You will be muted during the workshop*
- *You can ask questions using the Q&A tool.*
- *You should hear music if your sound is working*





Presenter:  
**Ekaterina Vinogradova**  
Applications Scientist



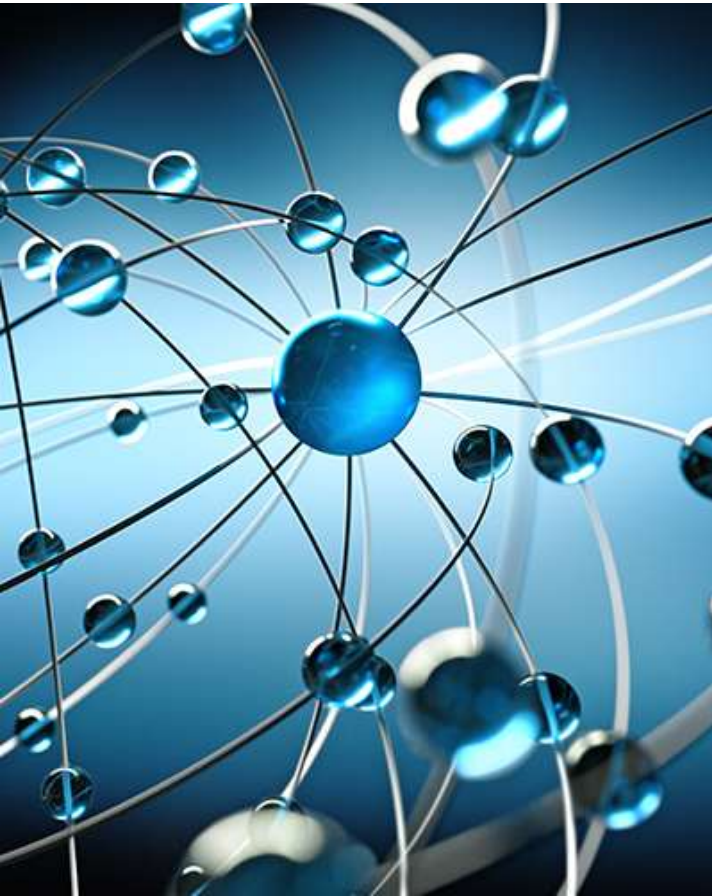
Host:  
**Tom Concolino**  
Sales Manager, XRD

You can ask questions during the presentation. Please use the Q&A to ask questions.



Recording will be  
available tomorrow.



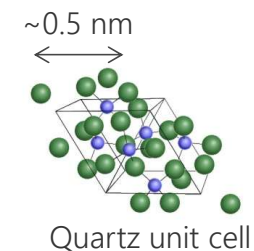
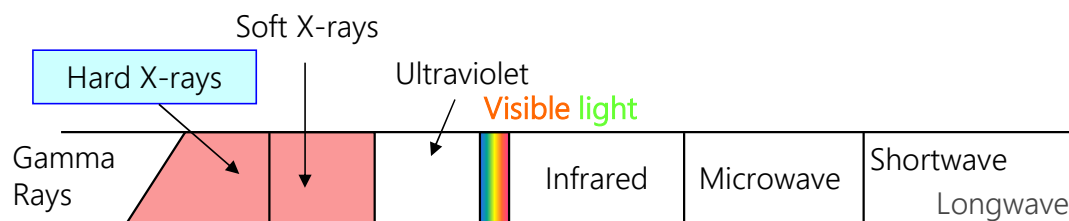


# Agenda

- The physics behind the peaks
- Phase identification foundations
- Basics of sample preparation
- Data collection quality
- Practical example

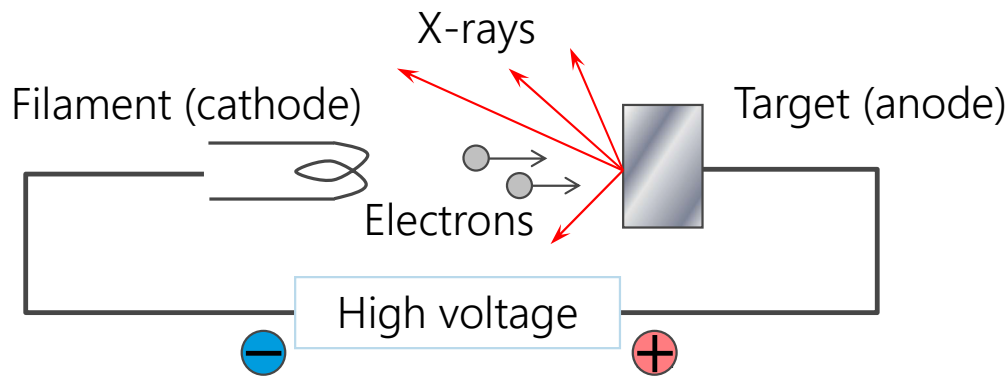
# Basics of X-rays

- Discovered by Wilhelm Roentgen (1895)
- Electromagnetic waves with very short wavelength ( $\sim 0.01\text{--}10\text{nm}$ )

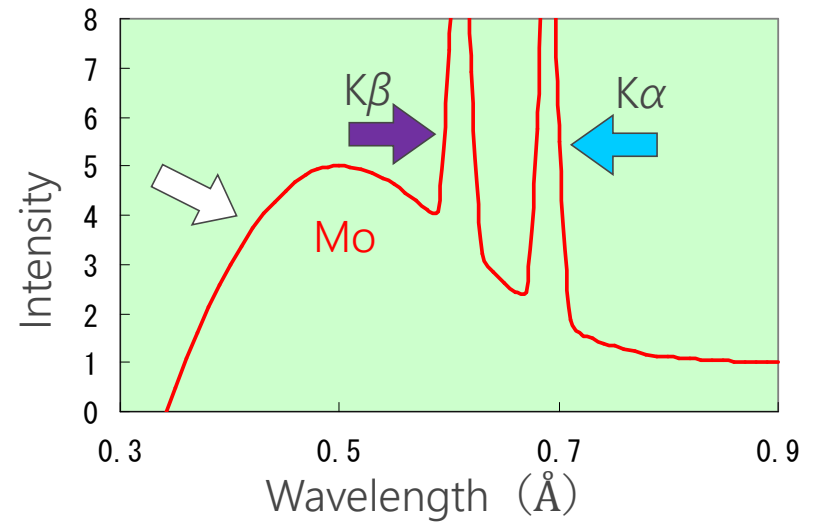


- Interaction with crystals occurs via elastic (Thomson) scattering from atomic planes
- Wavelength is comparable to interatomic distances in crystals ( $0.1\text{--}0.5\text{ nm}$ )

# Generation of X-rays

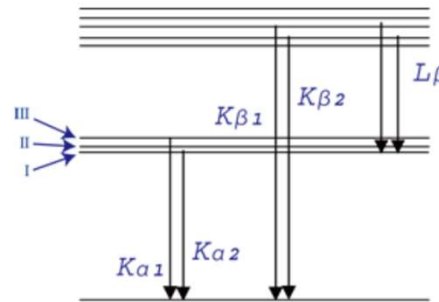
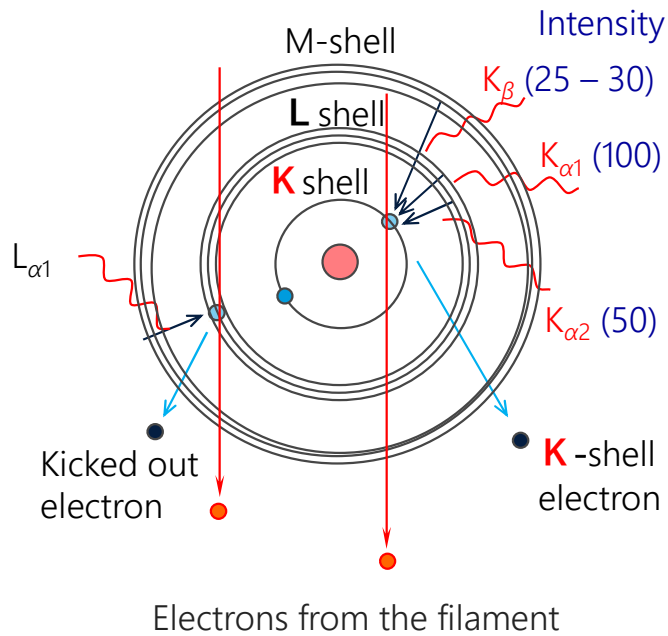


Generation efficiency  $\epsilon = 1.1 \times 10^{-9} Z V$  (Cu 0.1%)  
 $Z$  : atomic number of a target material  $V$  : applied voltage (V)



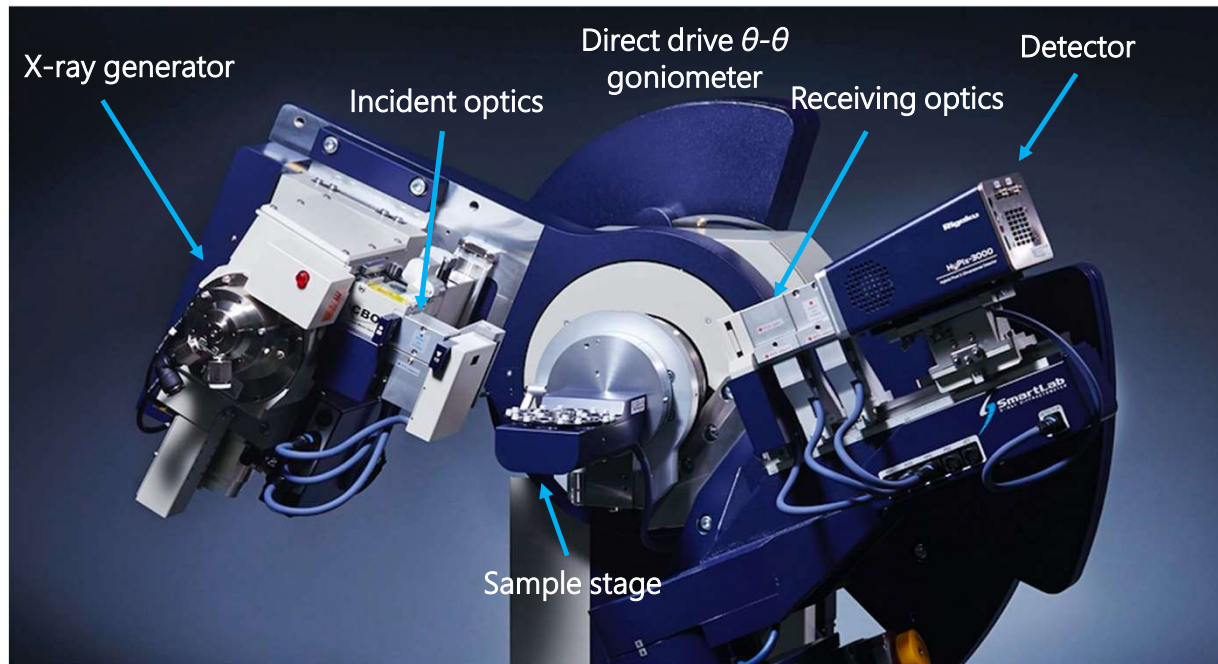
- The X-ray tube voltage is optimized to suppress bremsstrahlung and maximize characteristic X-ray emission.
- Biggest source of intensity fluctuations: cooling water  $t^\circ$  ( $1^\circ\text{C}$  change  $\sim 10\%$  variation).

# Generation of characteristic X-rays

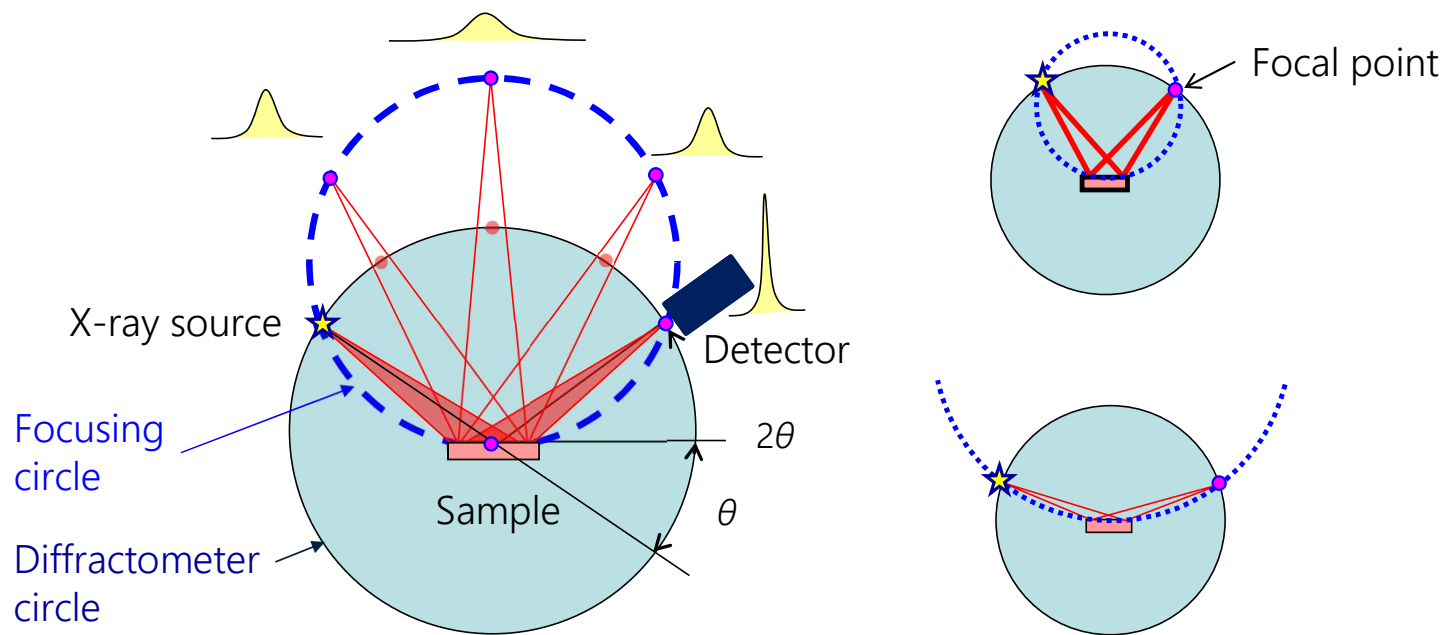


| Wavelength |                    |                    |                                 |                  |                 |
|------------|--------------------|--------------------|---------------------------------|------------------|-----------------|
| Target     | $K_{\alpha 1}$ , Å | $K_{\alpha 2}$ , Å | $K_{\alpha 2}/K_{\alpha 1}$ , Å | $K_{\alpha}$ , Å | $K_{\beta}$ , Å |
| ✓ Cu       | 1.540593           | 1.544414           | 0.497000                        | 1.541862         | 1.392246        |
| Co         | 1.789001           | 1.792886           | 0.497000                        | 1.790291         | 1.620823        |
| Mo         | 0.709317           | 0.713607           | 0.497000                        | 0.710741         | 0.632303        |
| Cr         | 2.289746           | 2.293652           | 0.500000                        | 2.291048         | 2.084912        |
| Fe         | 1.936081           | 1.940019           | 0.500000                        | 1.937394         | 1.756645        |
| V          | 2.503610           | 2.507430           | 0.500000                        | 2.504883         | 2.284446        |

# X-ray diffractometer



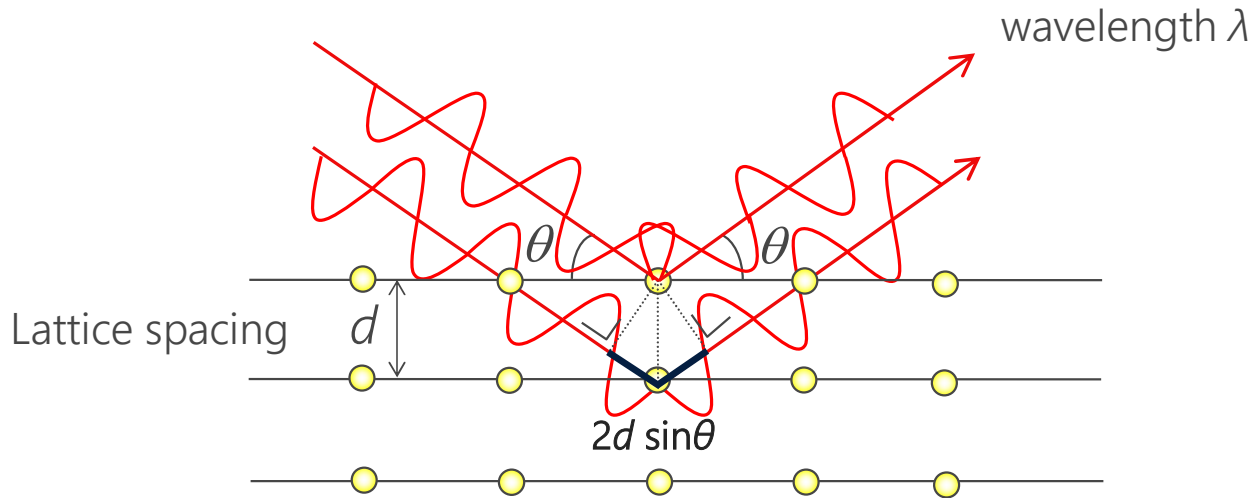
# Bragg-Brentano para-focusing geometry



The radius of focusing circle changes depending on the diffraction angle

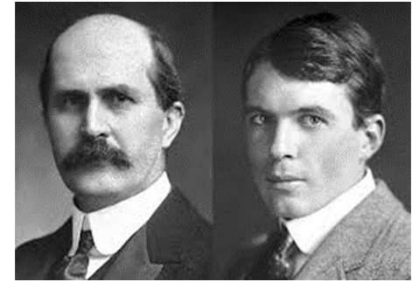
# Bragg's Law

Diffraction peaks occur when the path difference between waves scattered by planes spaced  $d$  equals an integer multiple of  $\lambda$ , fixing the observed  $2\theta$  angles.



$$\text{The Bragg's Equation: } n\lambda = 2d_{hkl} \sin \theta$$

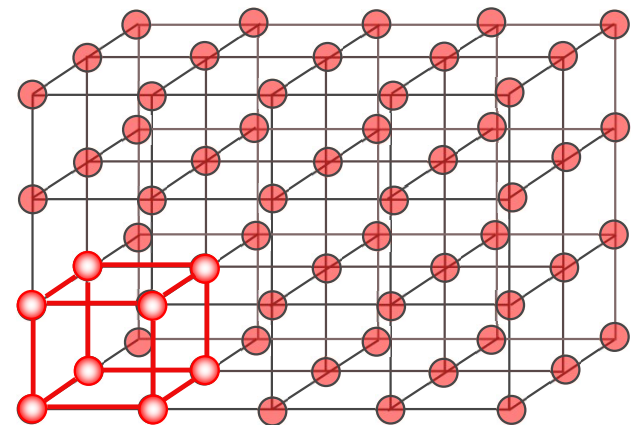
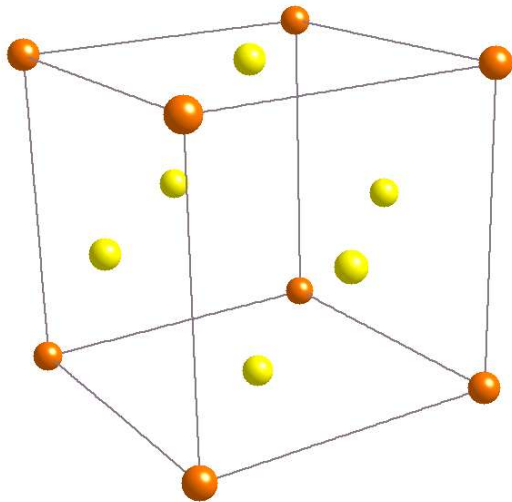
$n$ : integer



Sir William Henry Bragg and William Lawrence Bragg (Nobel Prize in 1915)

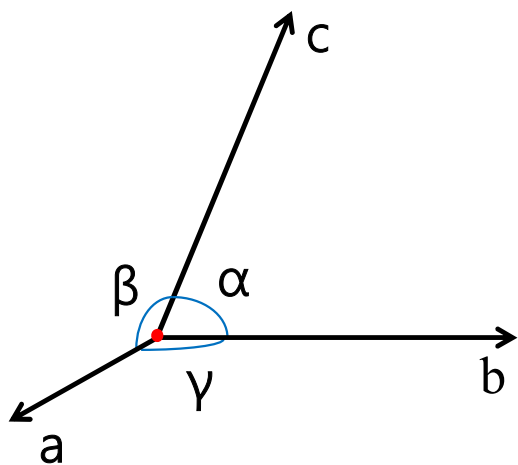
## What is a crystal?

A crystal is a solid in which atoms are arranged with long-range order, producing a repeating pattern.



Unit cell ← The smallest repeating unit of a crystal lattice

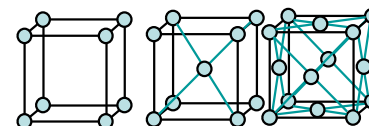
# Crystal systems



Cubic

$$a = b = c$$

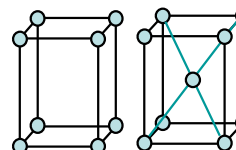
$$\alpha = \beta = \gamma = 90^\circ$$



Tetragonal

$$a = b \neq c$$

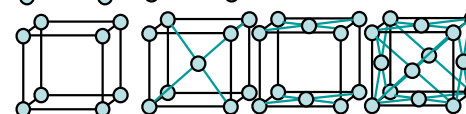
$$\alpha = \beta = \gamma = 90^\circ$$



Orthorhombic

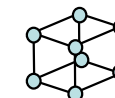
$$a \neq b \neq c$$

$$\alpha = \beta = \gamma = 90^\circ$$



Rhombohedral

$$a = b = c$$

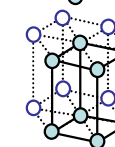


Trigonal and Hexagonal

$$\alpha = \beta = \gamma \neq 90^\circ$$

$$a = b \neq c$$

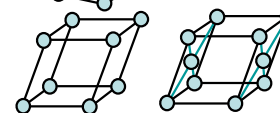
$$\alpha = \beta = 90^\circ \quad \gamma = 120^\circ$$



Monoclinic

$$a \neq b \neq c$$

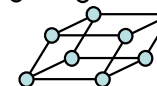
$$\alpha = \beta = 90^\circ \neq \gamma$$



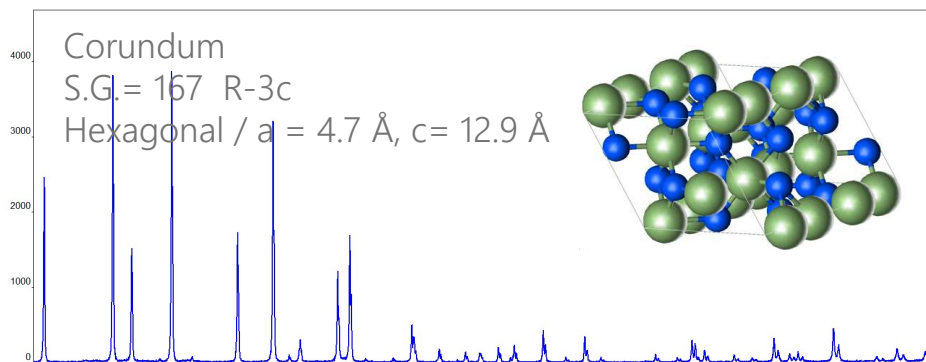
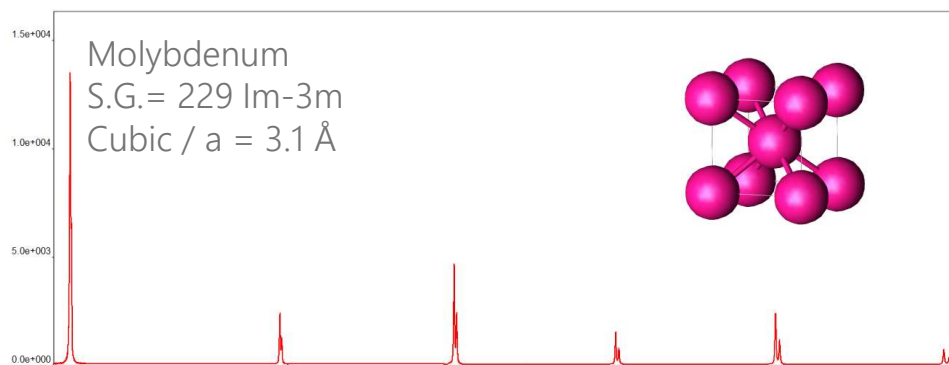
Triclinic

$$a \neq b \neq c$$

$$\alpha \neq \beta \neq \gamma \neq 90^\circ$$

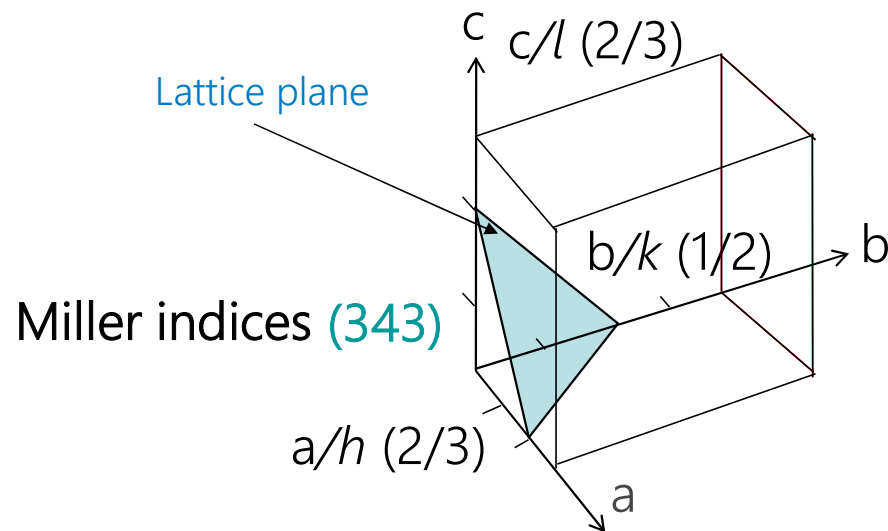


# High and low symmetries



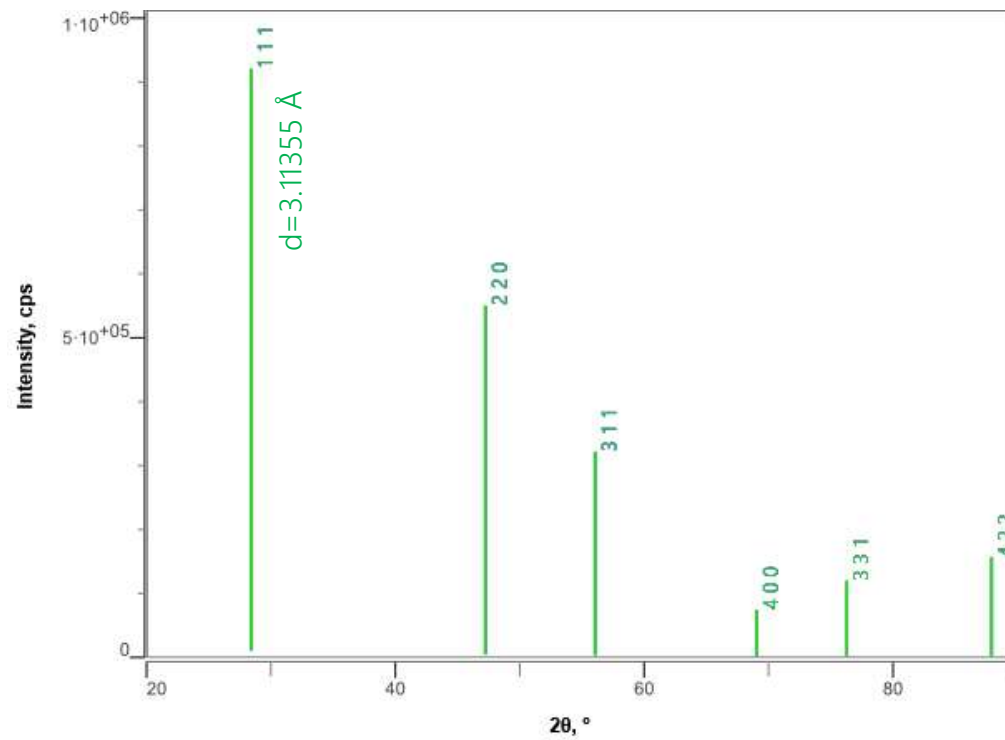
## Miller indices

- The Miller indices (h k l) specify the orientation of crystal planes relative to the unit cell.
- Obtained by taking the reciprocals of the plane's axis intercepts and reducing to the smallest integers.



# Database powder diffraction pattern from Si

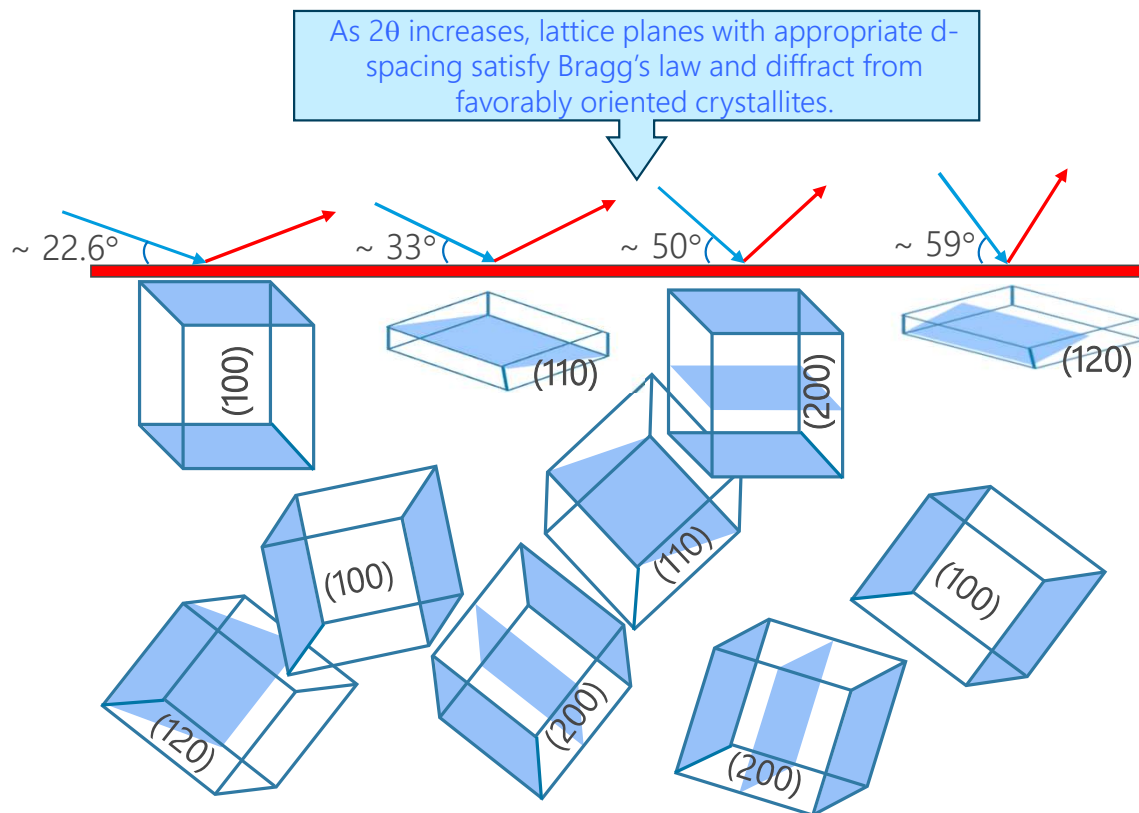
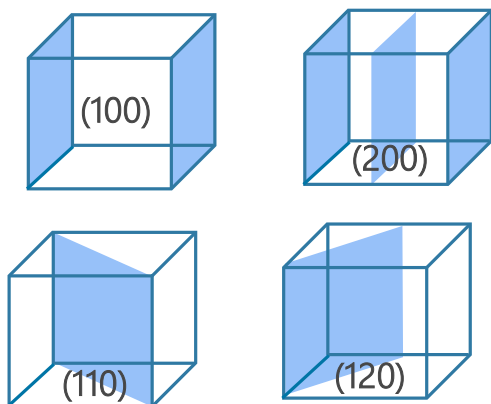
$$2d \sin\theta = n\lambda$$



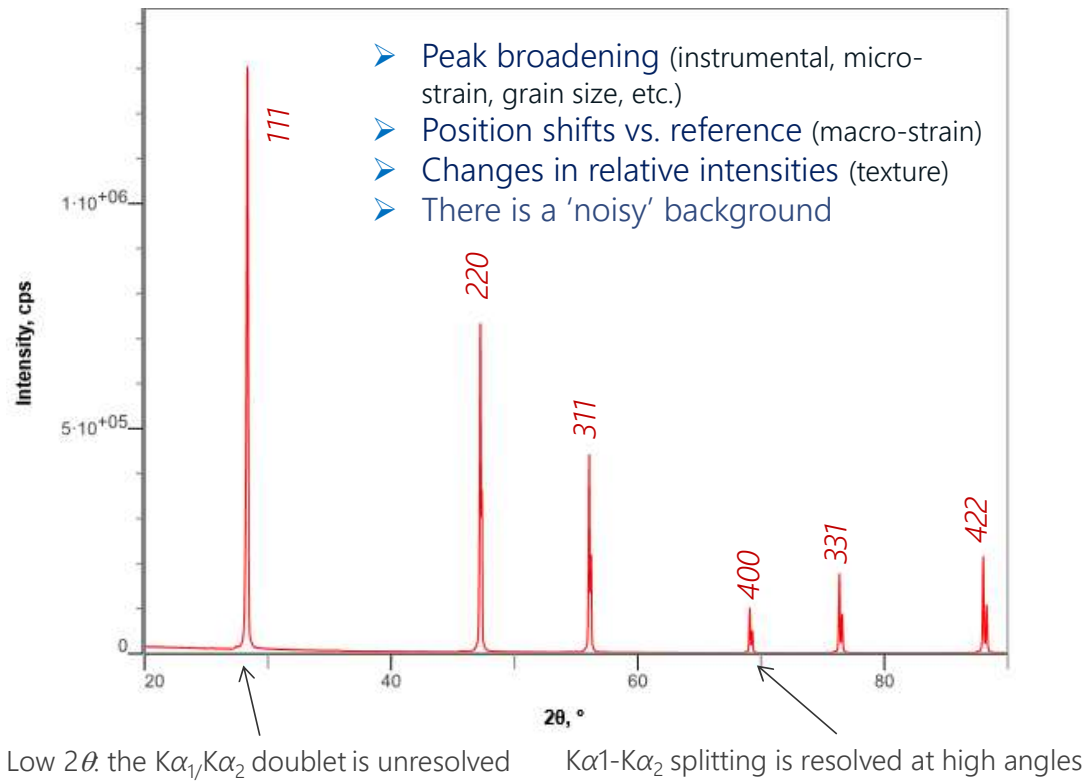
# Visualizing diffraction peaks

$$a = 2\text{\AA}; \lambda = 1.54\text{\AA}$$

| hkl | d    | Sin( $\theta$ ) | $\theta$ |
|-----|------|-----------------|----------|
| 100 | 2.00 | 0.38            | 22.64    |
| 110 | 1.41 | 0.54            | 33.00    |
| 200 | 1.00 | 0.77            | 50.36    |
| 120 | 0.89 | 0.86            | 59.40    |



# Experimental powder diffraction pattern from Si

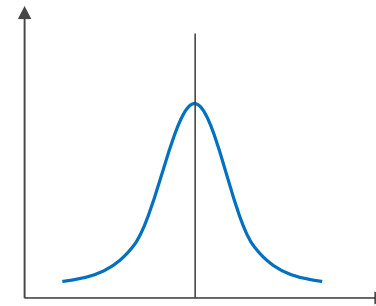


## Peak position

- Lattice parameter
  - Wavelength
- $$\left. \begin{array}{l} 2d \sin \theta = n\lambda, (n = 1, 2, 3 \dots) \\ d = f(a, b, c, \alpha, \beta, \gamma) \end{array} \right\}$$

- Instrument related systematic errors

- Flat sample
- Sample displacement
- Absorption
- Axial divergence
- Misalignment

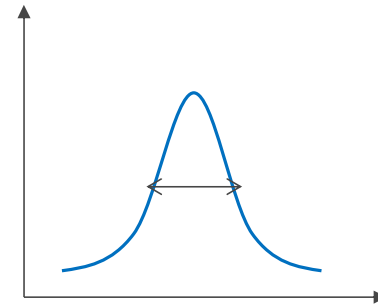


## Peak width

- Crystallite size
- Lattice distortion
- Instrument related broadening
  - Resolution
  - Flat sample
  - Absorption
  - Axial divergence
  - Misalignment

$$D = \frac{K\lambda}{\beta \cos \theta}$$

$$e = \frac{\beta}{4 \tan \theta}$$



## Peak intensity

- Crystal structure

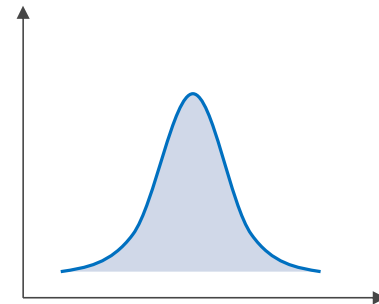
$$I_{hkl} \propto |F_{hkl}|^2 K_{LP}$$

$$F_{hkl} = \sum_{j=1}^N f_j e^{-B_j (\sin \theta_{hkl} / \lambda)^2} e^{2\pi i (hu_j + kv_j + lw_j)}$$

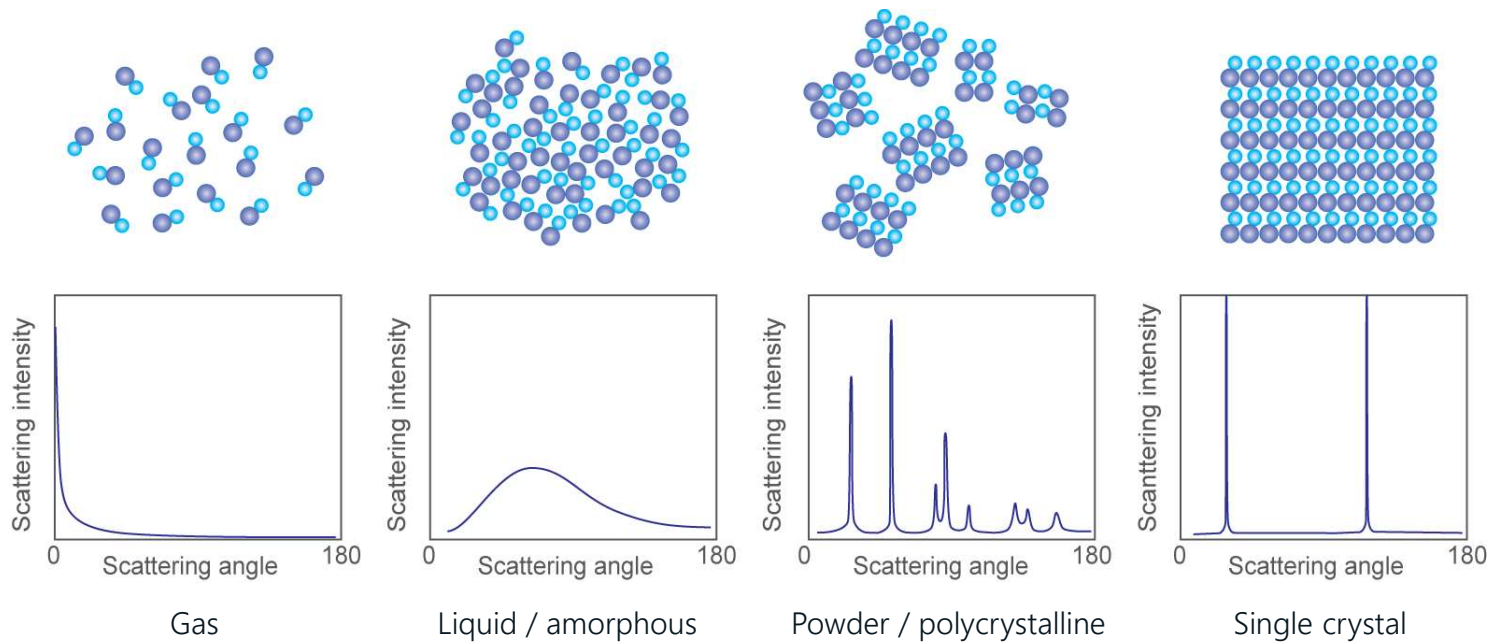
- Preferred orientation (March-Dollase)

$$P_{hkl} = \frac{1}{m_{hkl}} \sum_{j=1}^{m_{hkl}} \left( r^2 \cos^2 \alpha_j + r^{-1} \sin^2 \alpha_j \right)^{-3/2}$$

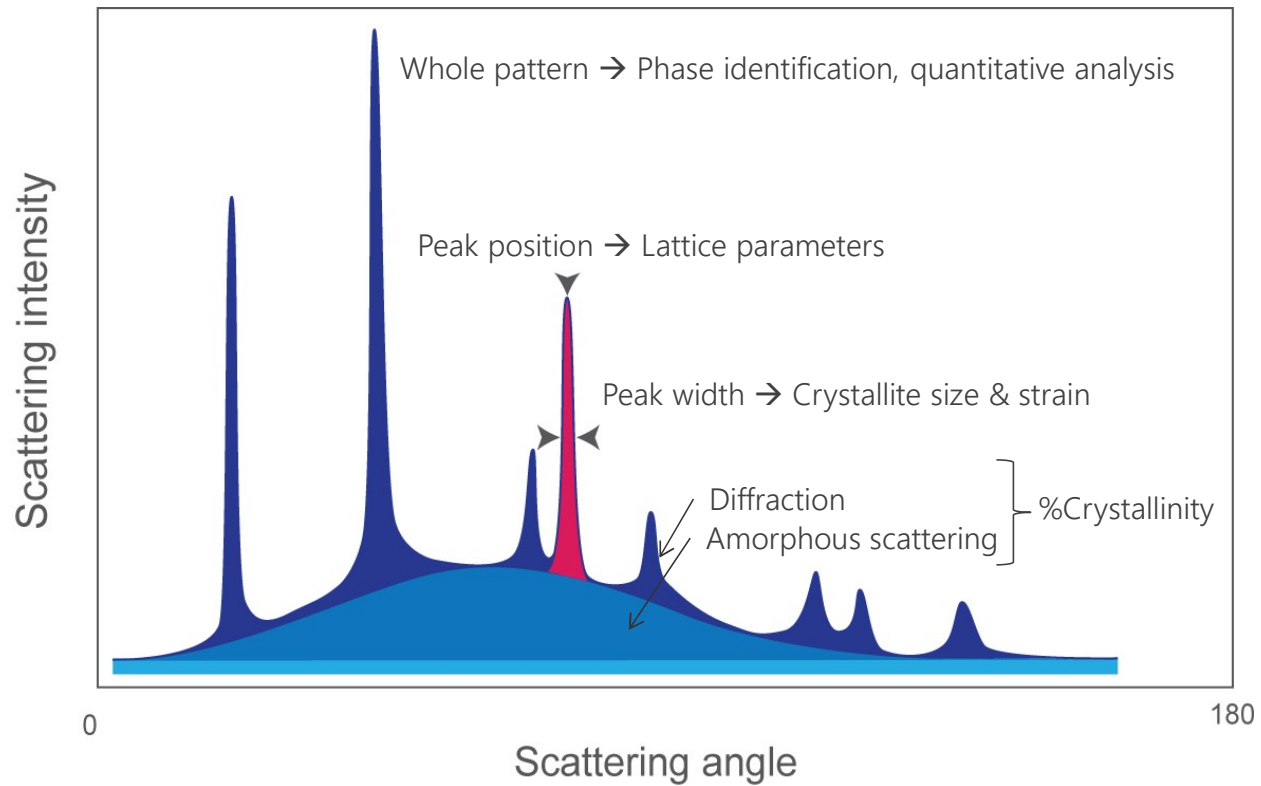
- Particle statistics



# Types of diffraction patterns



# XRD pattern / Sample information



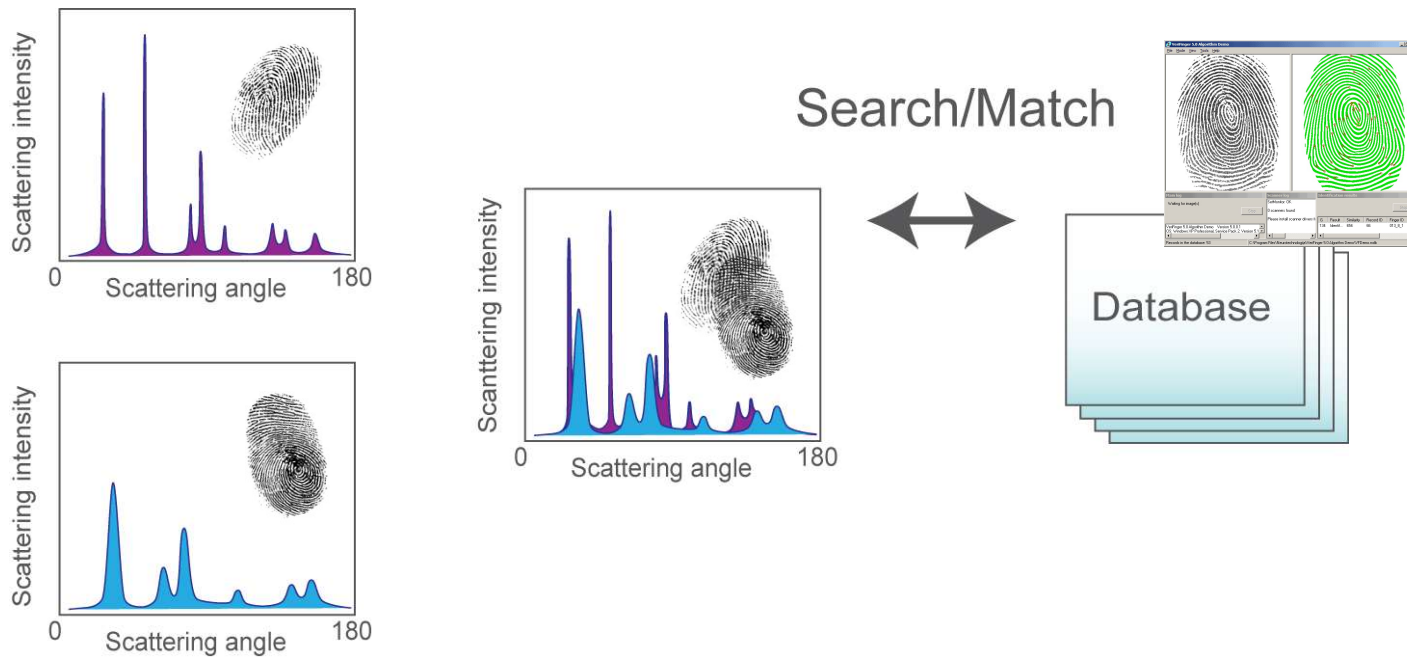
## Phase identification – Search/Match

Powder XRD identifies crystalline phases from the positions and relative intensities of diffraction peaks.

- ✓ Database of reference patterns
- ✓ Experimental diffraction data
- ✓ Search/Match software

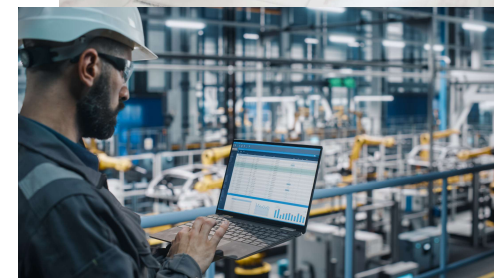
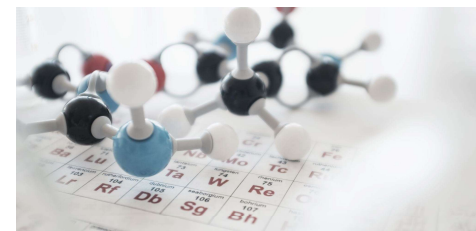


# Phase identification



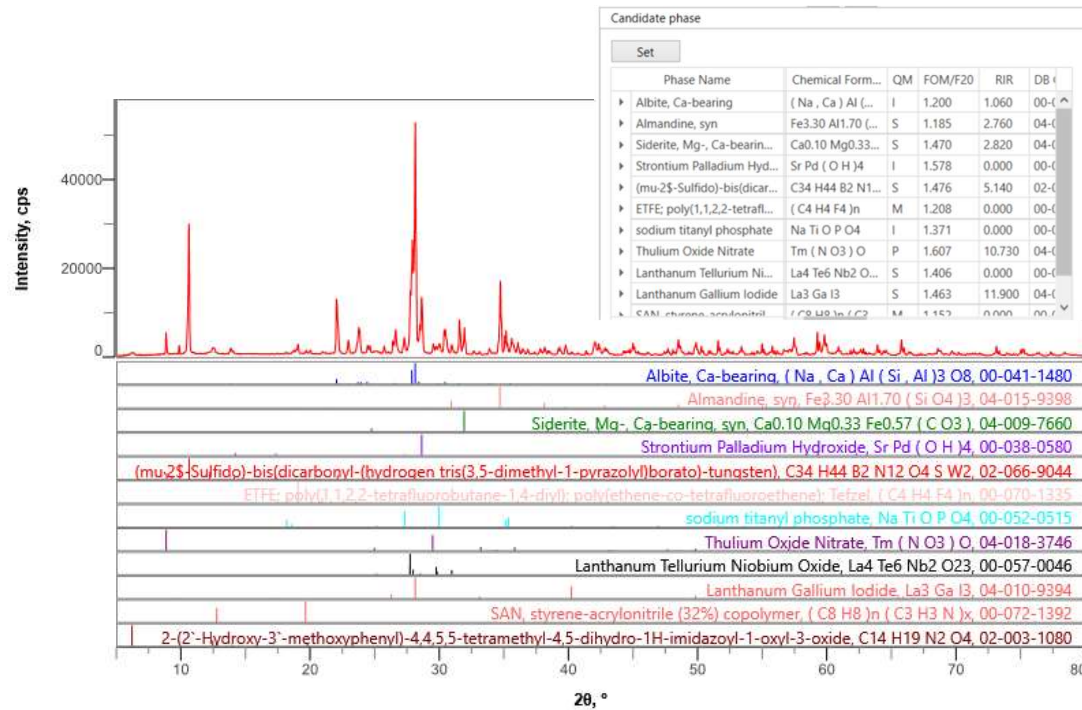
## Crystallographic databases

- ICDD (International Centre for Diffraction Data):
  - ✓ PDF-5+: 1,126,200+ entries (1-year license)
  - ✓ PDF-4/Axiom: 114,000+ entries (3-year license)
  - ✓ PDF-2: 369,200+ entries (5-year license)
- ICSD (Inorganic Crystal Structure Data)  
FIZ Karlsruhe – Leibniz Institute for Information Infrastructure  
Structure Database of Inorganic Crystal Compounds
  - ✓ 318,000+ entries
- Crystal Open Database (COD): open-access  
Administrated by Journal of Applied Crystallography
  - ✓ 530,000+ entries



# User subfiles

Search/Match returns too many phases – how do we narrow to the likely ones



**d-I database**

Search conditions: PDF-2 2026, PDF-S+ 2026

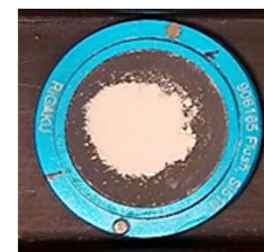
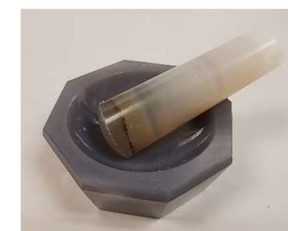
Search results table:

| No. | Name                     | Chemical Formula  | QM | RIR   | DB Card Number  |
|-----|--------------------------|---|----|-------|-----------------|
| 1   | Ye'elimite, syn          | Ca <sub>4</sub> Al <sub>6</sub> O <sub>12</sub> S O <sub>4</sub>          | I  | 0.000 | 00-016-0335 ... |
| 2   | Ye'elimite, syn          | Ca <sub>3</sub> Al <sub>6</sub> O <sub>12</sub> - Ca S O <sub>4</sub>     | I  | 0.000 | 00-016-0440 ... |
| 3   | Ye'elimite, syn          | Ca <sub>4</sub> Al <sub>6</sub> O <sub>12</sub> S O <sub>4</sub>          | I  | 0.000 | 00-033-0256 ... |
| 4   | Calcium Aluminum Oxid... | Ca <sub>12</sub> Al <sub>14</sub> O <sub>32</sub> S                       | B  | 0.000 | 00-036-0677 ... |
| 5   | Calcium Aluminum Oxid... | Ca <sub>4</sub> Al <sub>6</sub> O <sub>12</sub> S O <sub>4</sub>          | I  | 0.000 | 00-042-1478 ... |
| 6   | Ye'elimite, syn          | Ca <sub>4</sub> ( Al <sub>6</sub> O <sub>12</sub> ) ( S O <sub>4</sub> )  | I  | 3.410 | 01-083-7086 ... |
| 7   | Ye'elimite, syn          | Ca <sub>8</sub> ( Al <sub>12</sub> O <sub>24</sub> ) ( S O <sub>4</sub> ) | I  | 1.060 | 01-083-9042 ... |
| 8   | Ye'elimite, syn          | Ca <sub>4</sub> Al <sub>6</sub> ( S O <sub>4</sub> ) O <sub>12</sub>      | B  | 3.360 | 04-009-7268 ... |
| 9   | Calcium Aluminum Oxid... | Ca <sub>4</sub> Al <sub>6</sub> ( S O <sub>4</sub> ) O <sub>12</sub>      | S  | 1.020 | 04-011-1786 ... |
| 10  | Calcium Aluminum Oxid... | Ca <sub>4</sub> Al <sub>6</sub> S O <sub>12</sub>                         | I  | 3.510 | 04-011-2231 ... |

Buttons: Add to User Subfile...

## Sample preparation for powder XRD

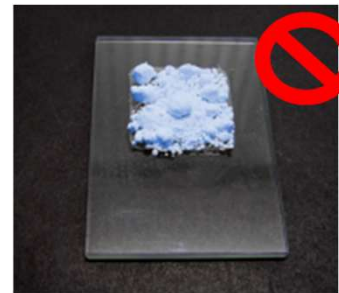
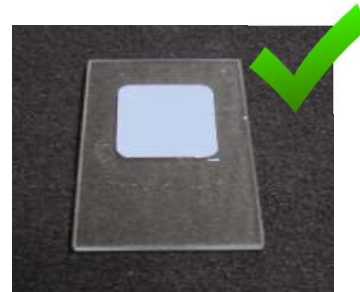
- Use finely ground powder ( $< 45\mu\text{m}$  / 325 mesh) to produce a flat surface and allow X-rays penetration.
- Ensure the sample surface is level with the holder rim.
- The sample holder cavity should be completely filled, unless using a low background holder).



## Sample preparation

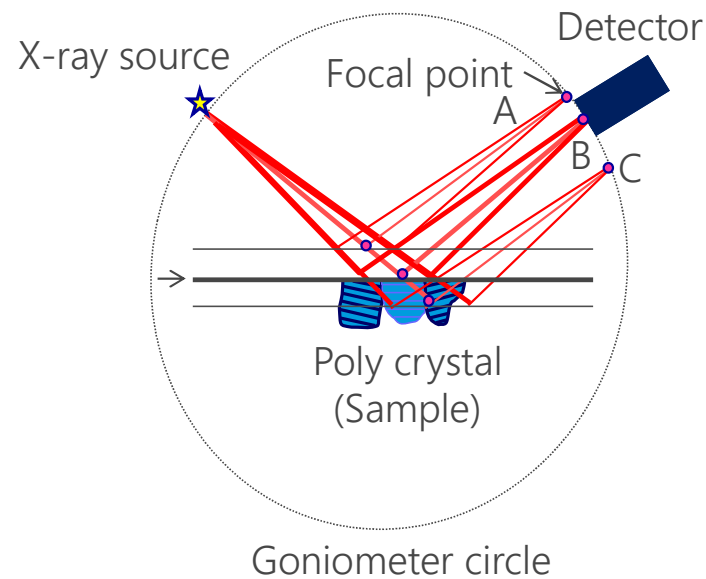
### Common factors and errors

- Sample height displacement
- Sample flatness
- Sample particle size
- Preferred orientation
- Sample transparency

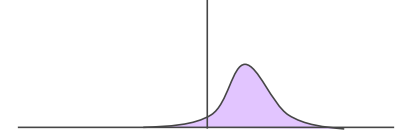


# Sample displacement

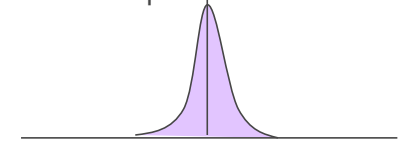
- The biggest source of error
- Peak position shifts  $\sim 0.02^\circ$  for every  $30\ \mu\text{m}$
- $\Delta 2\theta = 2h \cos \frac{\theta}{R}$ ,  
where  $R$  – goniometer radius.



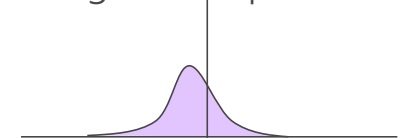
A. Positive displacement



B. No displacement



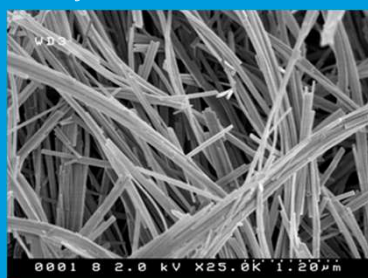
C. Negative displacement



# Preferred orientation

- Common in samples with plate/needle/fiber morphologies
- Alters relative peak intensities
- Can compromise phase ID
- Certain sample preparation methods can increase the effect

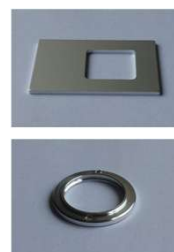
Halloysite



Kaolinite



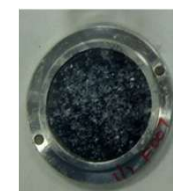
Back loading



Side loading



- Randomized mounting
- Slurry mounting with acetone



Spinning

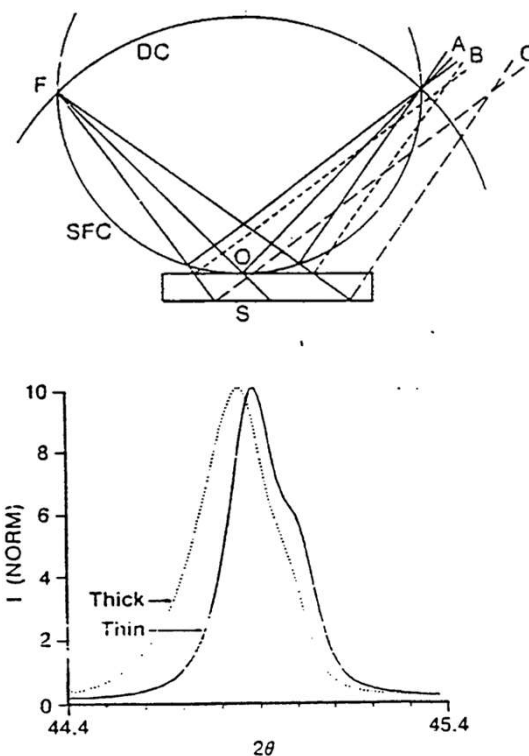


Transmission geometry



## Sample transparency

- Diffracted intensity arises from multiple depths
- Common in low-attenuation (low- $\mu$ ) materials: organics, polymers, low-Z compounds
- Shifts peaks (to lower  $2\theta$ ) and makes them asymmetric



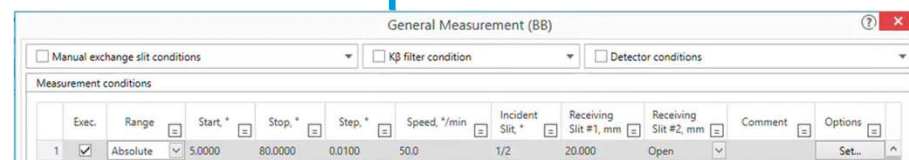
### Solutions:

- Thin sample layer
- Transmission geometry

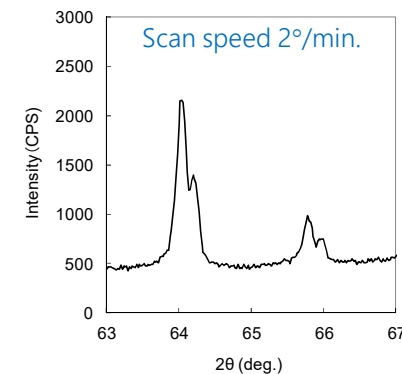
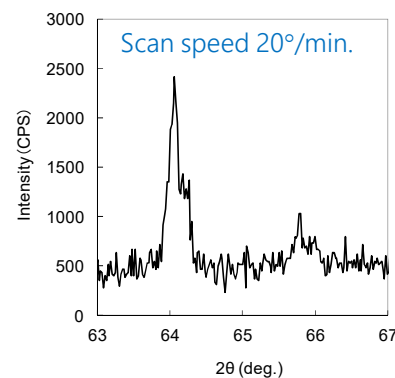
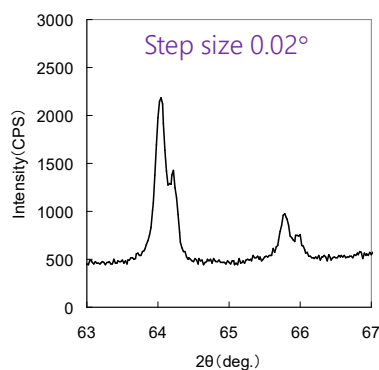
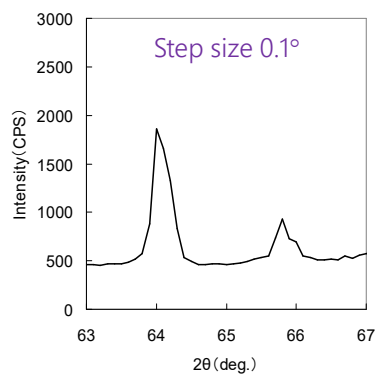


## Data collection strategies

- ✓ Pre-scan your sample (5-80° 2θ, 0.02° step, 50°/min).
- ✓ Adjust the scan range, step size and speed.
  - Step size: FWHM x 1/10
  - Adjust the scan speed to have high diffraction intensities and decrease statistical fluctuation.



Scan duration: 2 min.



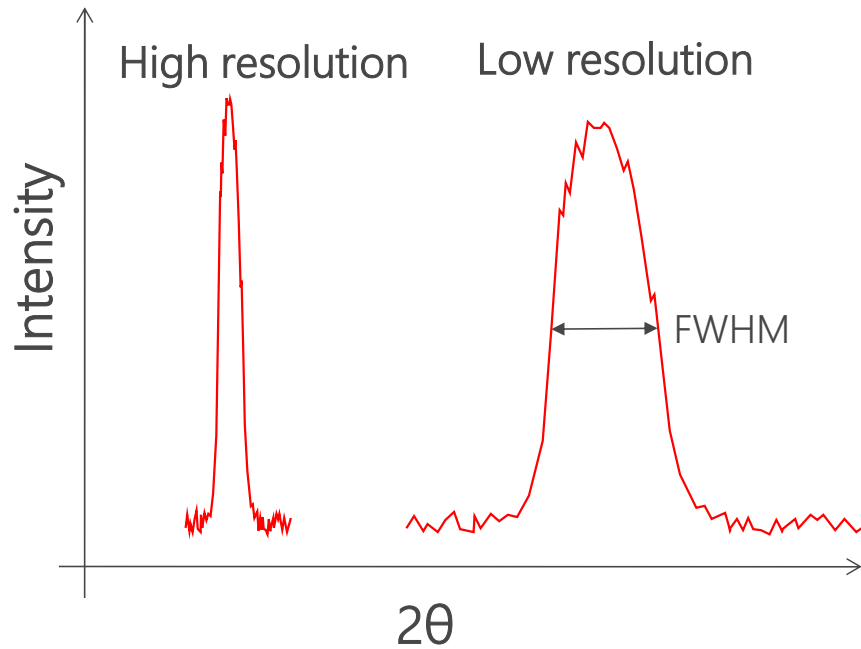
## Data quality

- Resolution (peak width and shape)
- Peak-to-background ratio
- Signal-to-noise ratio



# Peak resolution

- = Angular resolution
- Diffraction maps  $d \rightarrow 2\theta$



Peak List

Only displayed in chart Phase name:

| No. | 2θ, °     | d, Å        | Height, cps | FWHM, °  |
|-----|-----------|-------------|-------------|----------|
| 1   | 25.244(5) | 3.5250(6)   | 21862(191)  | 0.153(4) |
| 2   | 27.372(3) | 3.2556(4)   | 8049(123)   | 0.111(3) |
| 3   | 36.019(3) | 2.49139(18) | 3845(90)    | 0.100(3) |

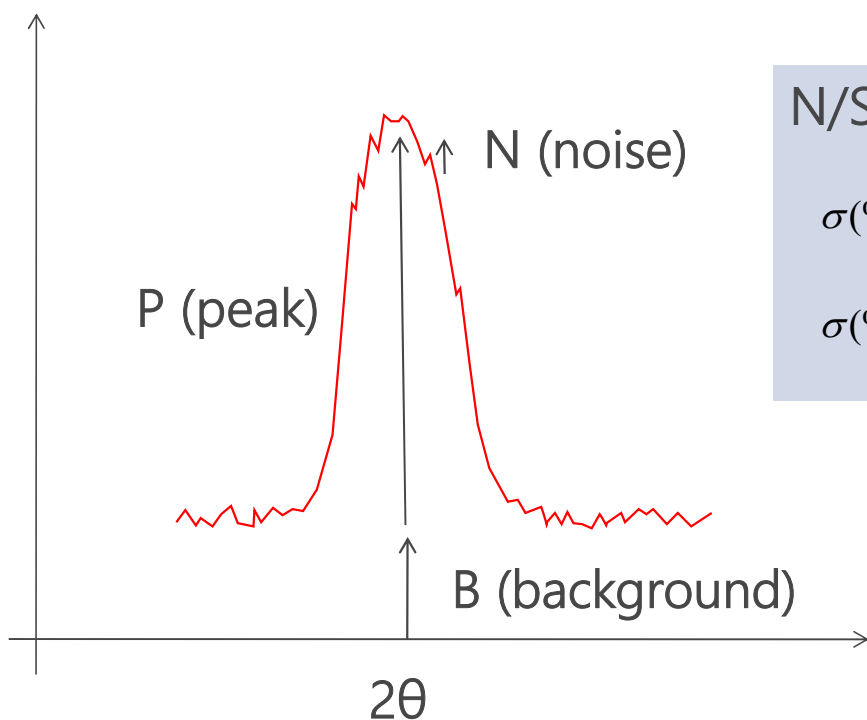


# Signal-to-noise ratio

Statistical confidence of the peak



Use SNR to decide if a weak peak is truly detected



N/S:  $\sigma$  %

$$\sigma(\%) = \frac{1}{\sqrt{P}} \sqrt{\frac{P+2B}{P}} \times 100$$

$$\sigma(\%) = \frac{1}{\sqrt{P}} \times 100 \quad (\text{B is negligible.})$$

Peak Evaluation

Run

Apply data reduction

Peak search

Preprocess:

Method:

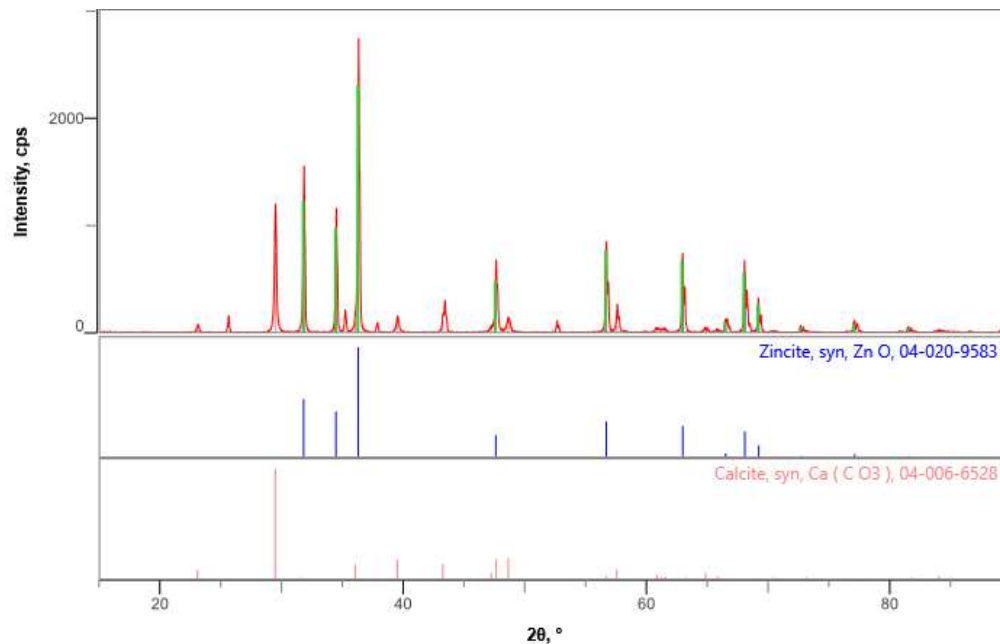
$\sigma$  cut

Find amorphous peak

- Count longer ( $\text{SNR} \propto \sqrt{\text{time}}$ )
- Repeat N scans and average ( $\text{SNR} \propto \sqrt{N}$ )

## Phase identification / Pattern matching

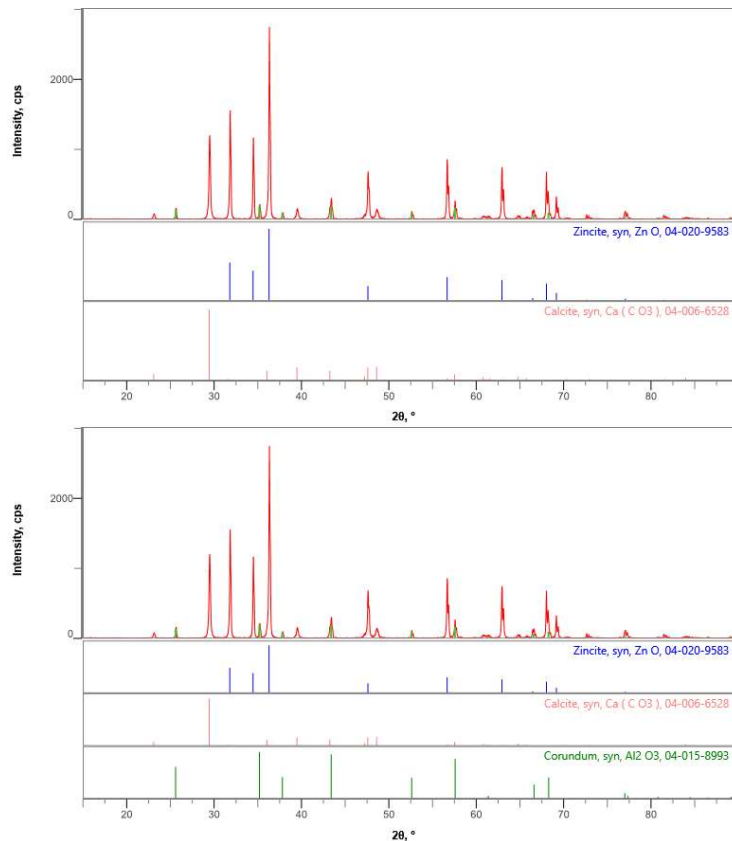
All reference peaks must be present in the data; unmatched peaks indicate no valid match.



### Exceptions

- Preferred orientation (check if absences are *hkl*-specific)
- Bad statistics / coarse grains (regrind to reduce size)
- Below detection limit (very weak peaks not visible above background or overlap)

## Phase identification / Additional peaks

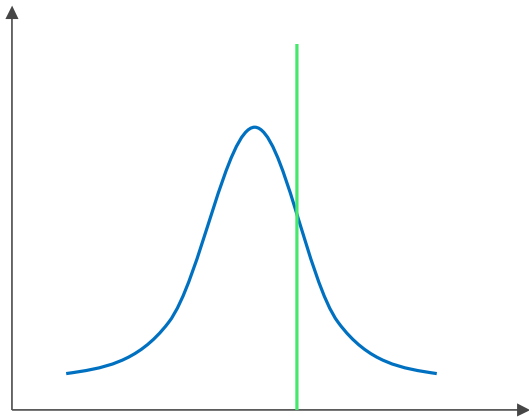


### Possible reasons

- Extra peaks may indicate an impurity in the sample (continue searching for references)
- Very small peaks can be spectral artifacts (e.g.  $K\beta$ , W L-lines (verify with the  $K\alpha$ ,  $K\beta$ ,  $L\alpha$  Lines Display cursor))

## Phase identification / Peak shift

### Diagnosing peak shifts



- Larger at low angles, smaller at high angles ( $\propto \cos\theta$ )
  - Likely: sample displacement
- Grows with angle ( $\propto \tan\theta$ )
  - Likely: true lattice-parameter change (thermal expansion/contraction, chemical composition)
  - Direction: expansion  $\rightarrow$  lower  $2\theta$ ; contraction  $\rightarrow$  higher  $2\theta$

# Questions?





We'll follow up with your questions.



Recording will be available tomorrow.



Register for webinar.



# Practical XRD with Confidence

Episode 2 – Quantitative XRD in Practice:  
From Phase Estimates to Defensible Results

Wednesday, May 20, 2026, at 1 pm CDT

**Presenter:** Akhilesh Tripathi, PhD

**Host:** Tom Concolino, PhD

