



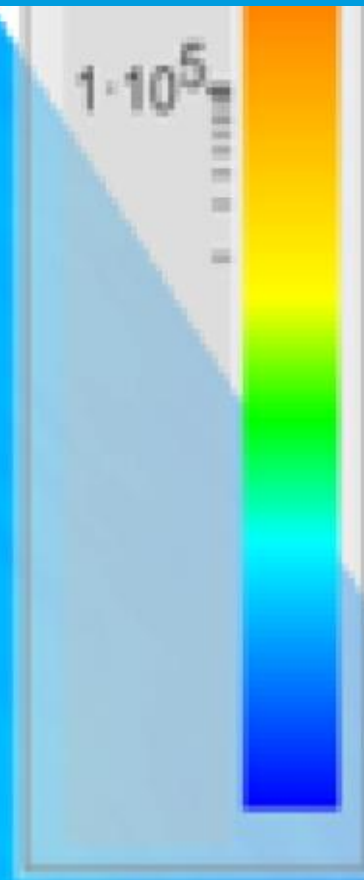
# Practical XRD with Confidence

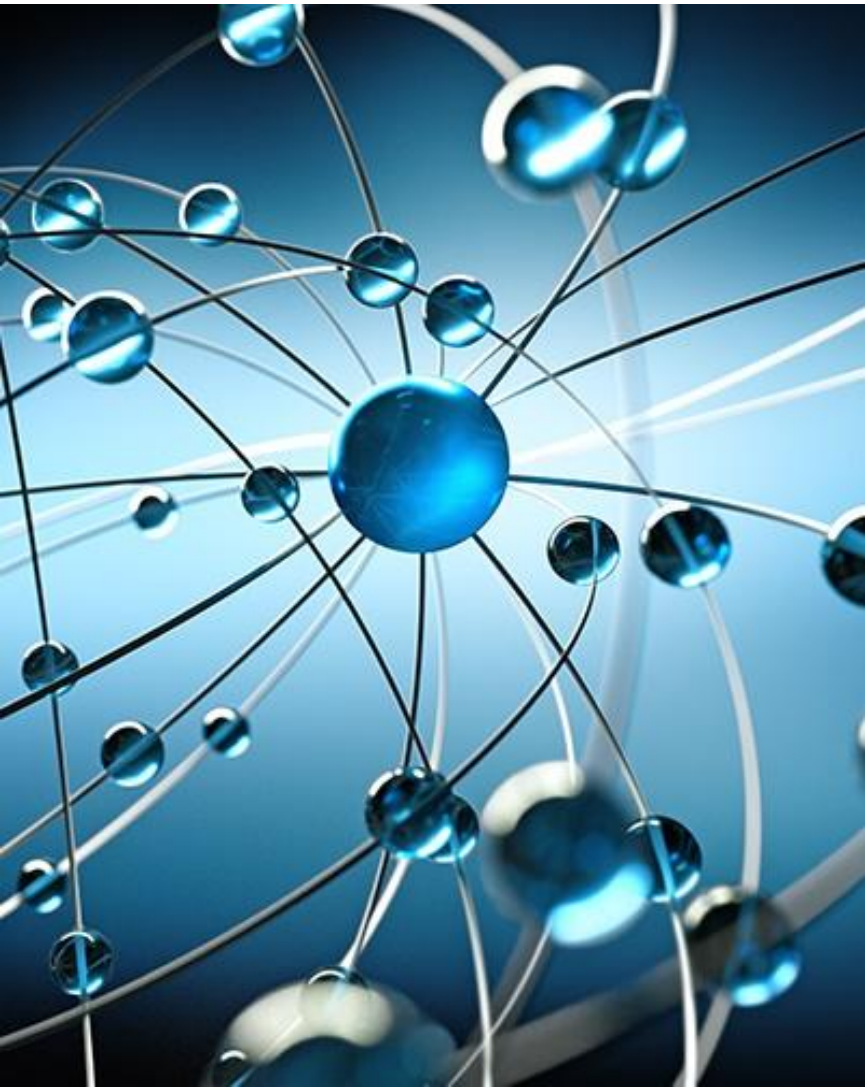
Episode 3 – XRD for Thin Films: Choosing the Right  
Measurement for Structure, Strain, and Thickness

Wednesday, June 3, 2026, at 1 pm CDT

**Presenter:** Keisuke Saito, PhD

- *You will be muted during the workshop*
- *You can ask questions using the Q&A tool.*
- *You should hear music if your sound is working*



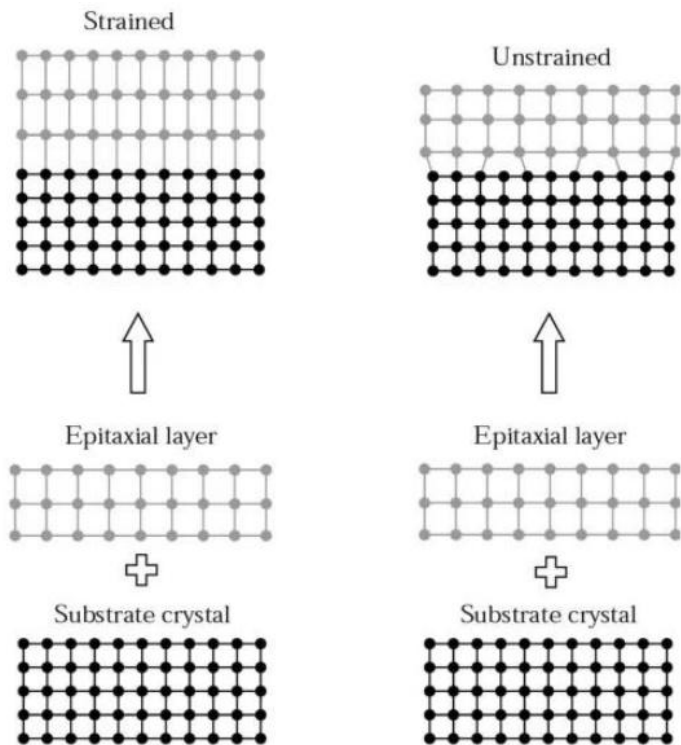


# What I will cover today:

1. Preferred orientation of crystals
2. XRD scans for polycrystalline thin films
3. Phase identification of a 5 nm thick Fe–O thin film
4. HRXRD measurement types and instrument configurations
5. HRXRD scans in reciprocal space
6. Epitaxial perovskite thin films: structure and thickness

# *1. Preferred orientation of crystals*

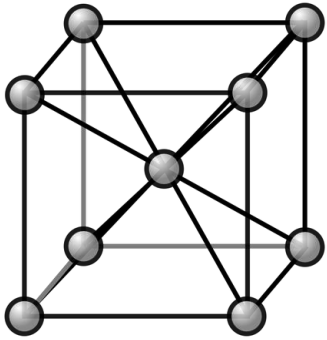
# Thin film growth



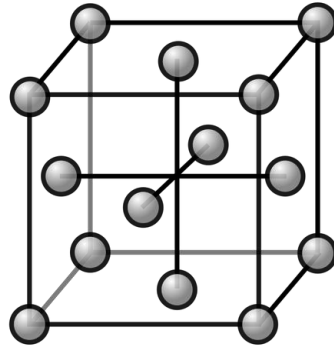
- Epitaxial growth: crystal-aligned thin film growth on a single-crystal substrate

# Thin film growth

Body centered cubic



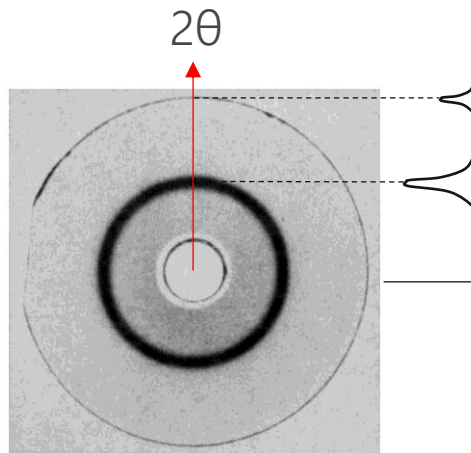
Face centered cubic



- Amorphous or mismatched substrates: lattice plane with lowest surface energy will orient  $\parallel$  to surface
  - Surface energy  $E_{hkl}$ 
    - B.C.C crystals:  $E_{110} < E_{100} < E_{111}$
    - F.C.C crystals:  $E_{111} < E_{100} < E_{110}$

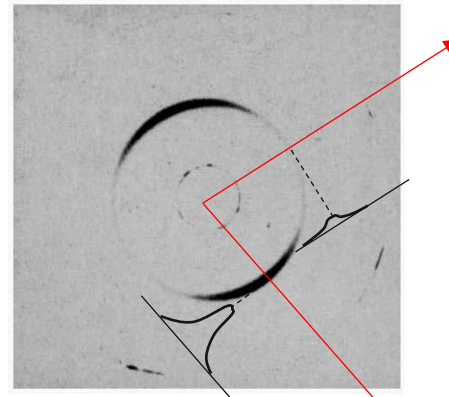
# X-ray diffraction pattern & crystal orientation

Powder



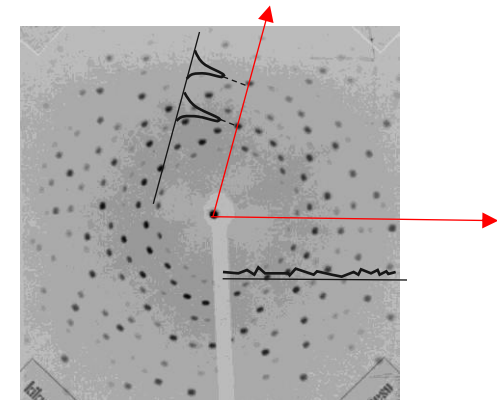
Full Debye  
ring

Crystals with  
orientation



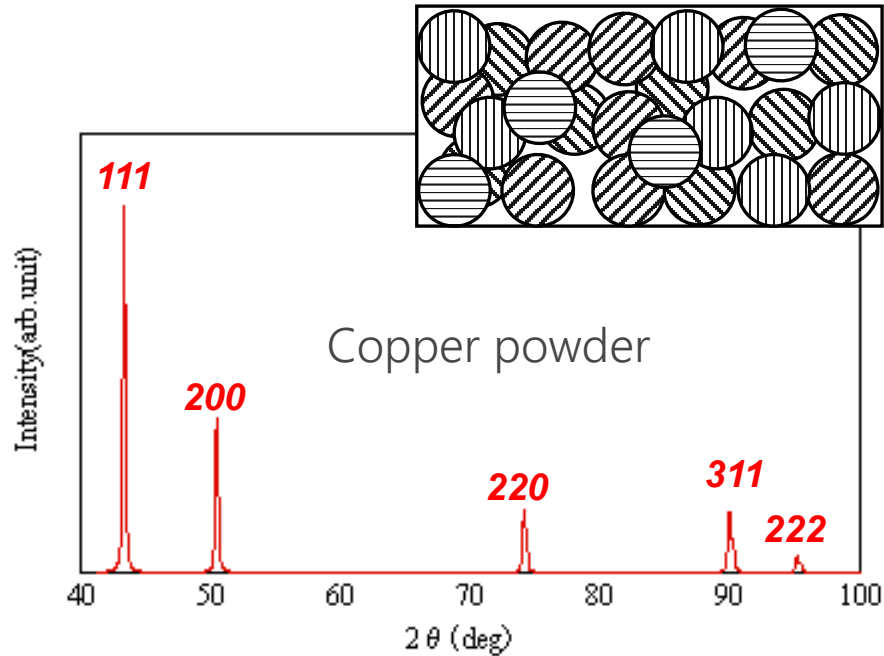
Only partial  
rings visible

Single crystal  
or  
epitaxial film

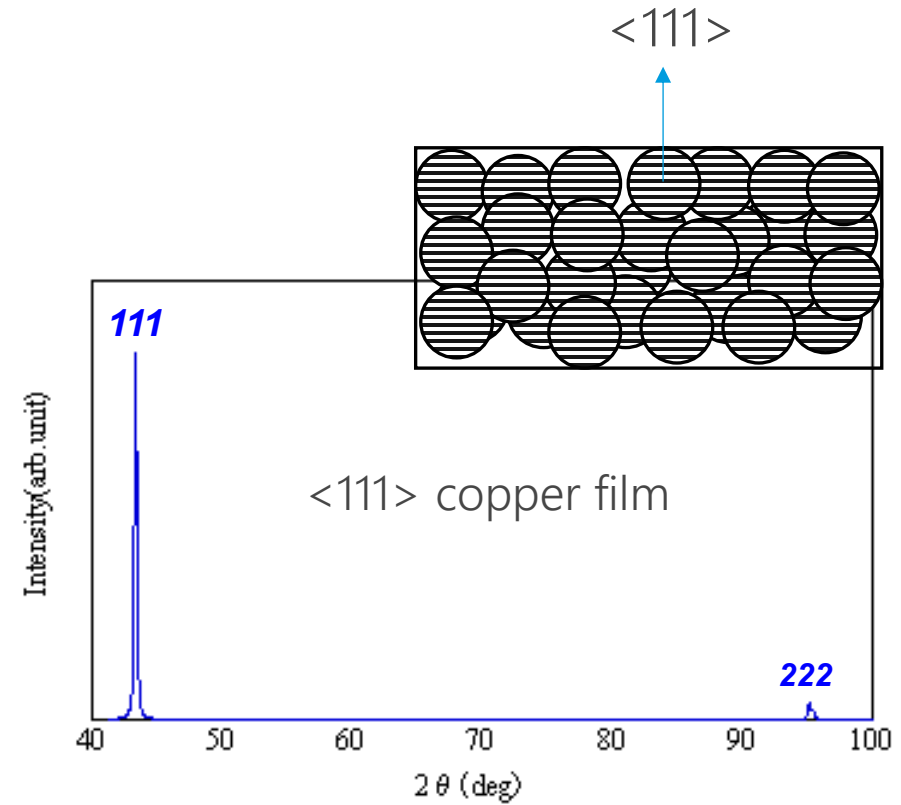


Sharp spots

# Preferred orientation



No preferred orientation

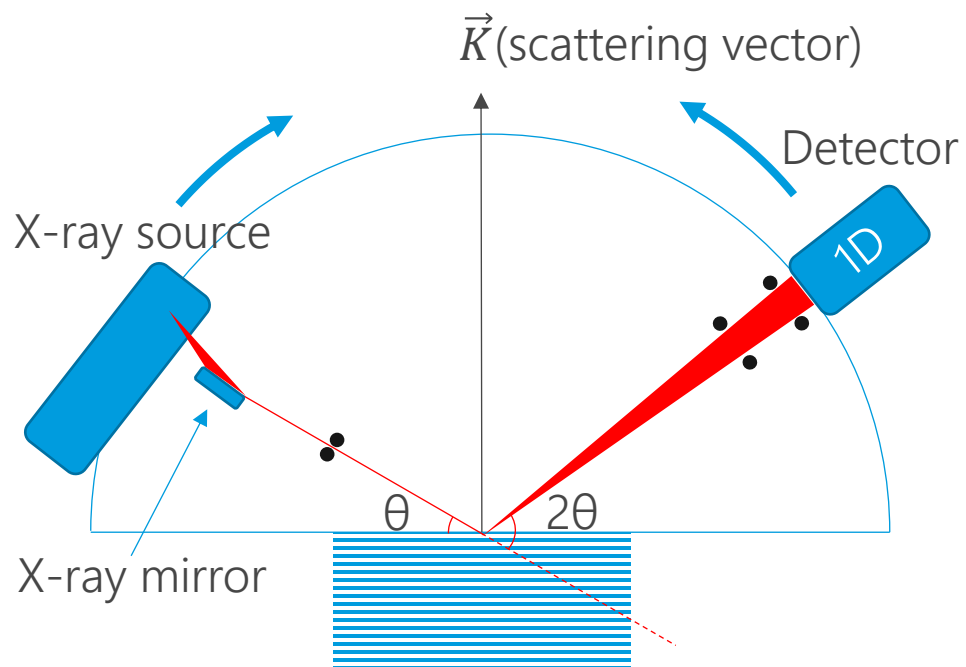


With preferred orientation



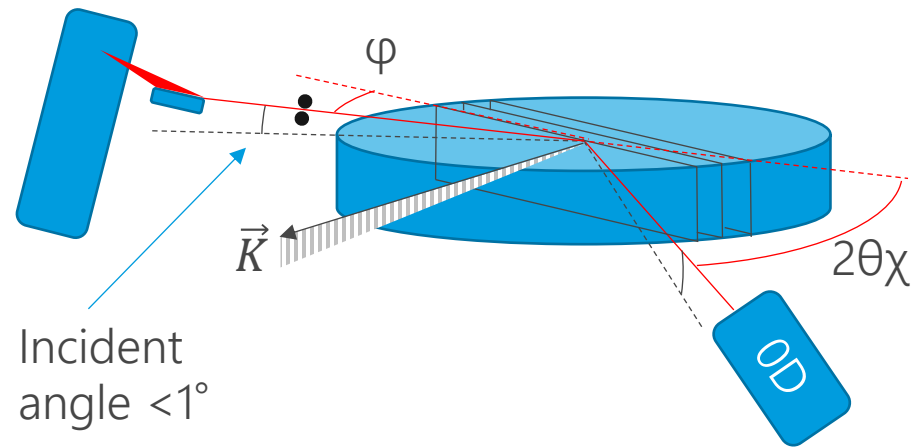
*2. XRD scans  
for  
polycrystalline thin films*

# Symmetric $2\theta$ - $\theta$ scan



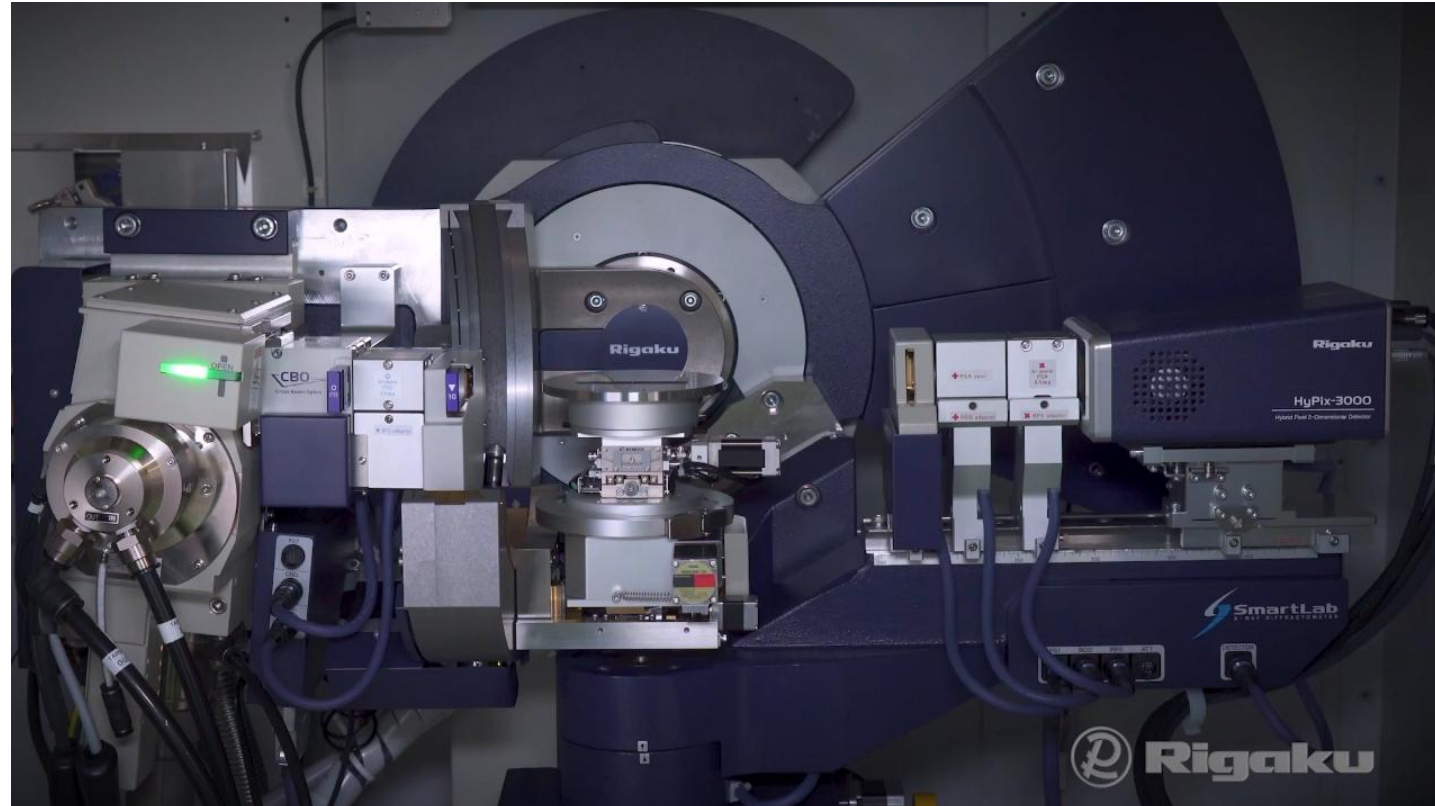
- d-spacings  $\perp$  from surface
- Preferred orientation
- Phase identification
- Large penetration depth of X-ray

# In-plane scan ( $2\theta\chi$ - $\varphi$ scan)



- d-spacings  $\parallel$  to surface
- Preferred orientation  $\parallel$  to surface
- Phase identification  $\parallel$  to surface
- Depth profile

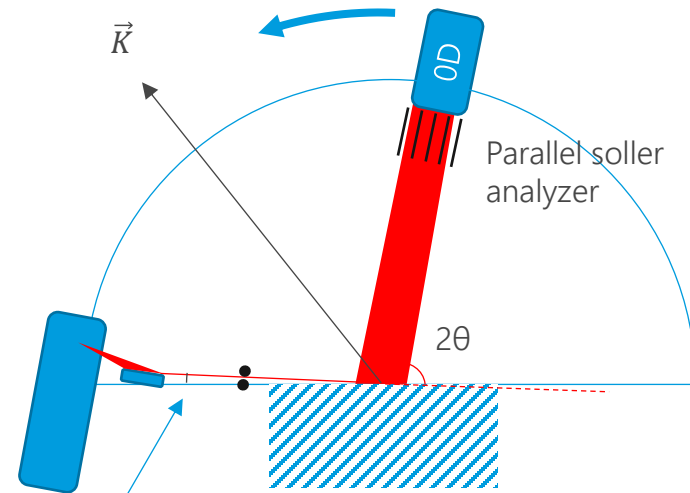
# In-plane scan



<https://www.youtube.com/watch?v=rzC5amxVaf0&t=1s>

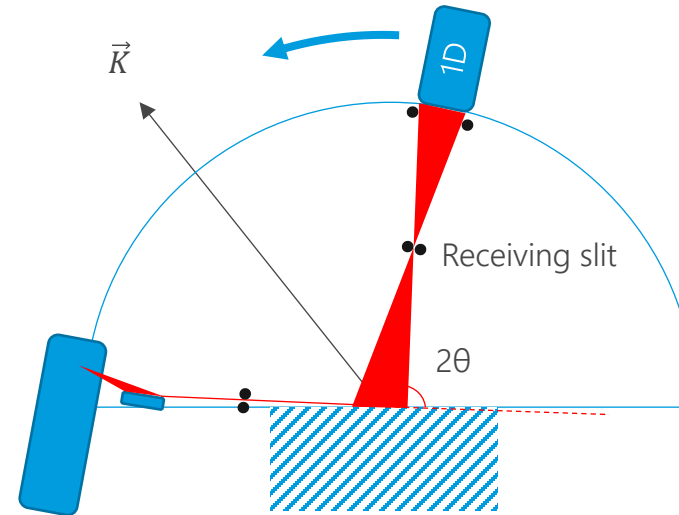
# Grazing incidence XRD ( $2\theta$ scan)

Traditional PSA scan



Incident angle  $< 1^\circ$

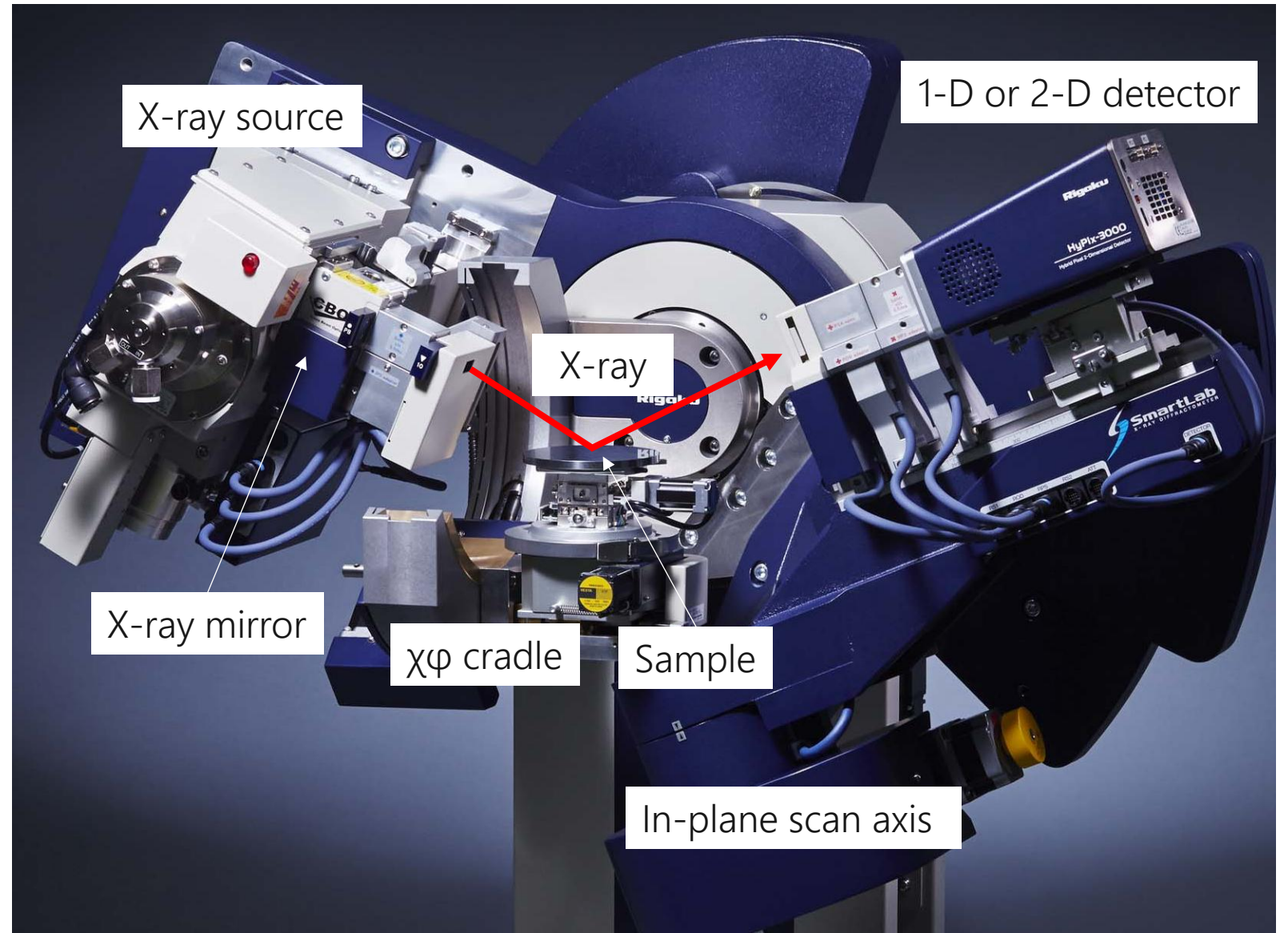
1D fast GID scan



[Next-Gen GI-XRD Technology Cuts Scan Times to Minutes](#)

- d-spacings from tilted direction
- Tilt angle depends on  $2\theta$
- Phase identification for non-textured thin film

# Configurations



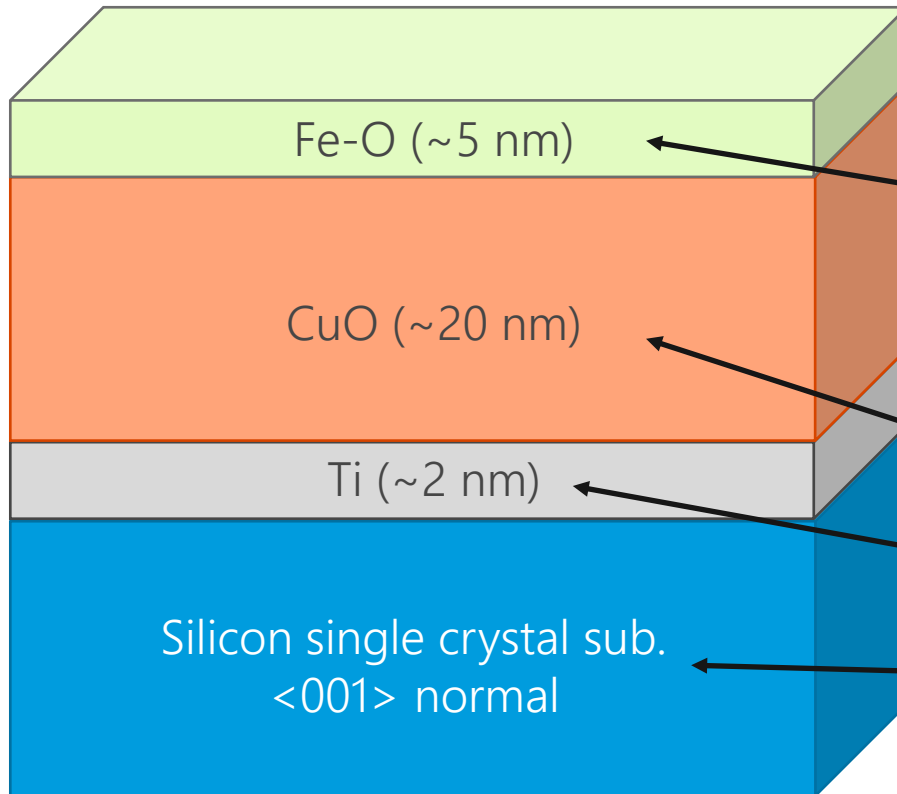
## 2D detector

- Hybrid pixel array detector
- Attenuator free
- Pixel resolution  $\sim 0.02^\circ$
- $>29^\circ$  ( $2\theta$ ) coverage

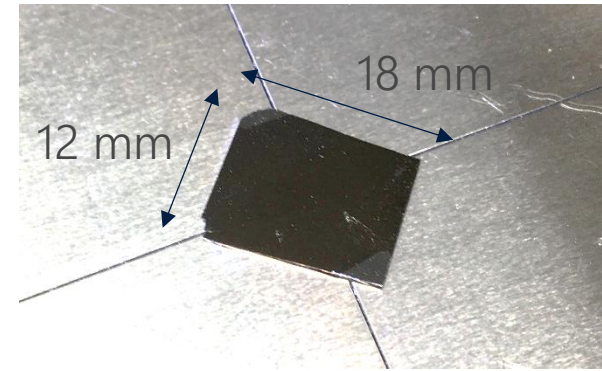


### *3. Phase identification of a 5 nm thick Fe–O thin film*

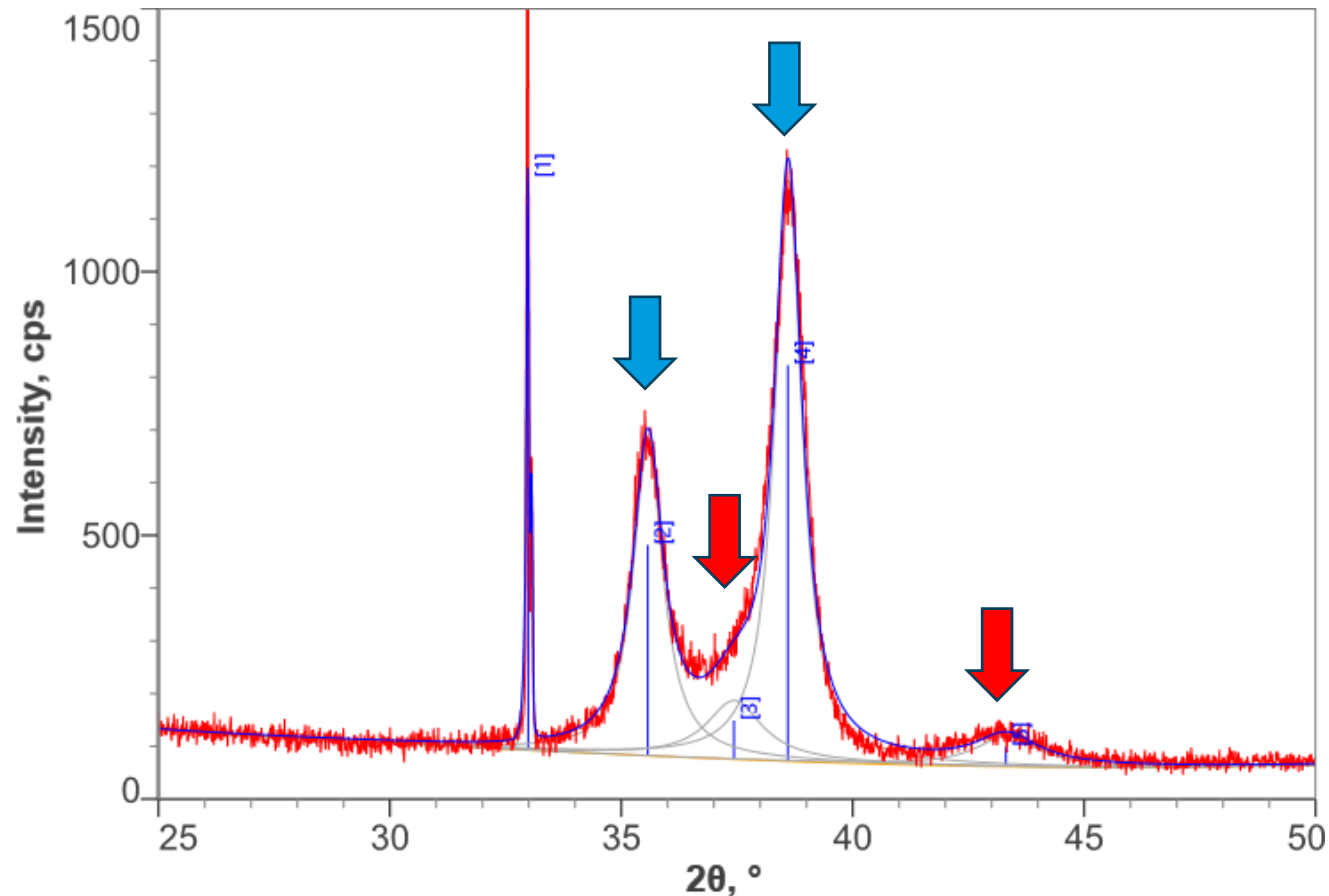
# Sample structure



- Fe-O
  - FeO wüstite
  - Fe<sub>2</sub>O<sub>3</sub> hematite
  - Fe<sub>2</sub>O<sub>3</sub> maghemite
  - Fe<sub>3</sub>O<sub>4</sub> magnetite
- CuO: tenorite
- Ti: adhesive layer, amorphous
- Si: substrate



# Symmetric 2 $\theta$ - $\theta$ scan



- Peak FWHM → Crystallite size
  - Scherrer equation
- Peaks 2&4
  - Peak FWHMs ~ 0.9°
  - Crystallite sizes ~ 11 nm
- Peaks 3 & 5
  - Peak FWHMs ~ 1.9°
  - Crystallite sizes ~ 5 nm

# Scherrer equation

$$D = \frac{K \cdot \lambda}{\beta \cdot \cos \theta}$$

$D$ : average crystallite size

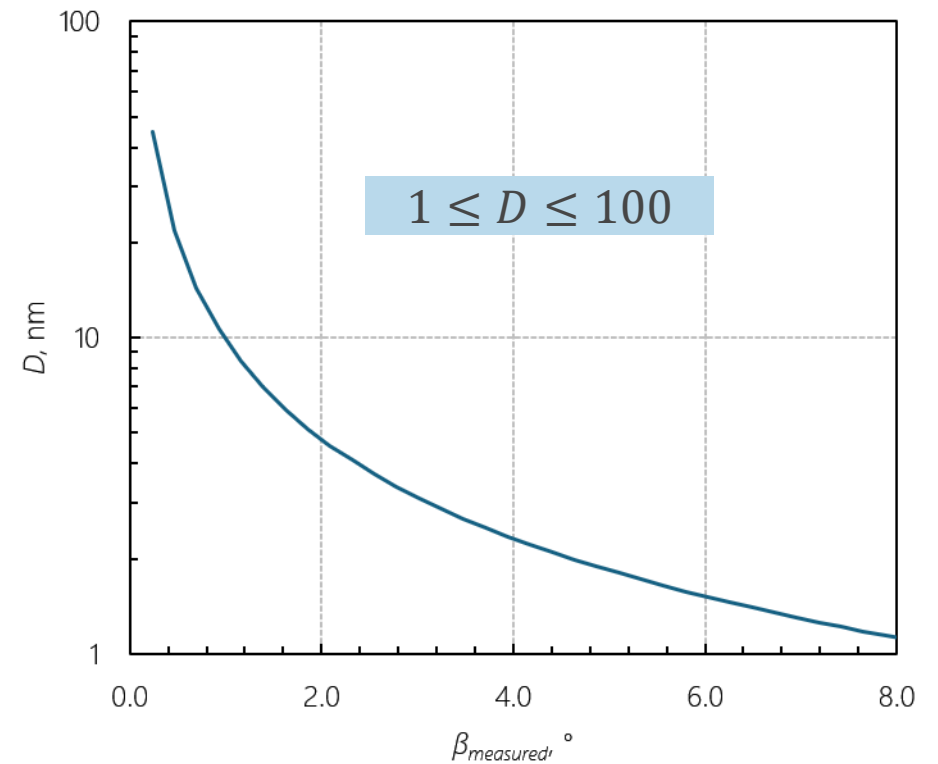
$K$ : shape factor (1.0 oriented thin film, 0.89 sphere)

$\lambda$ : wavelength of X-ray

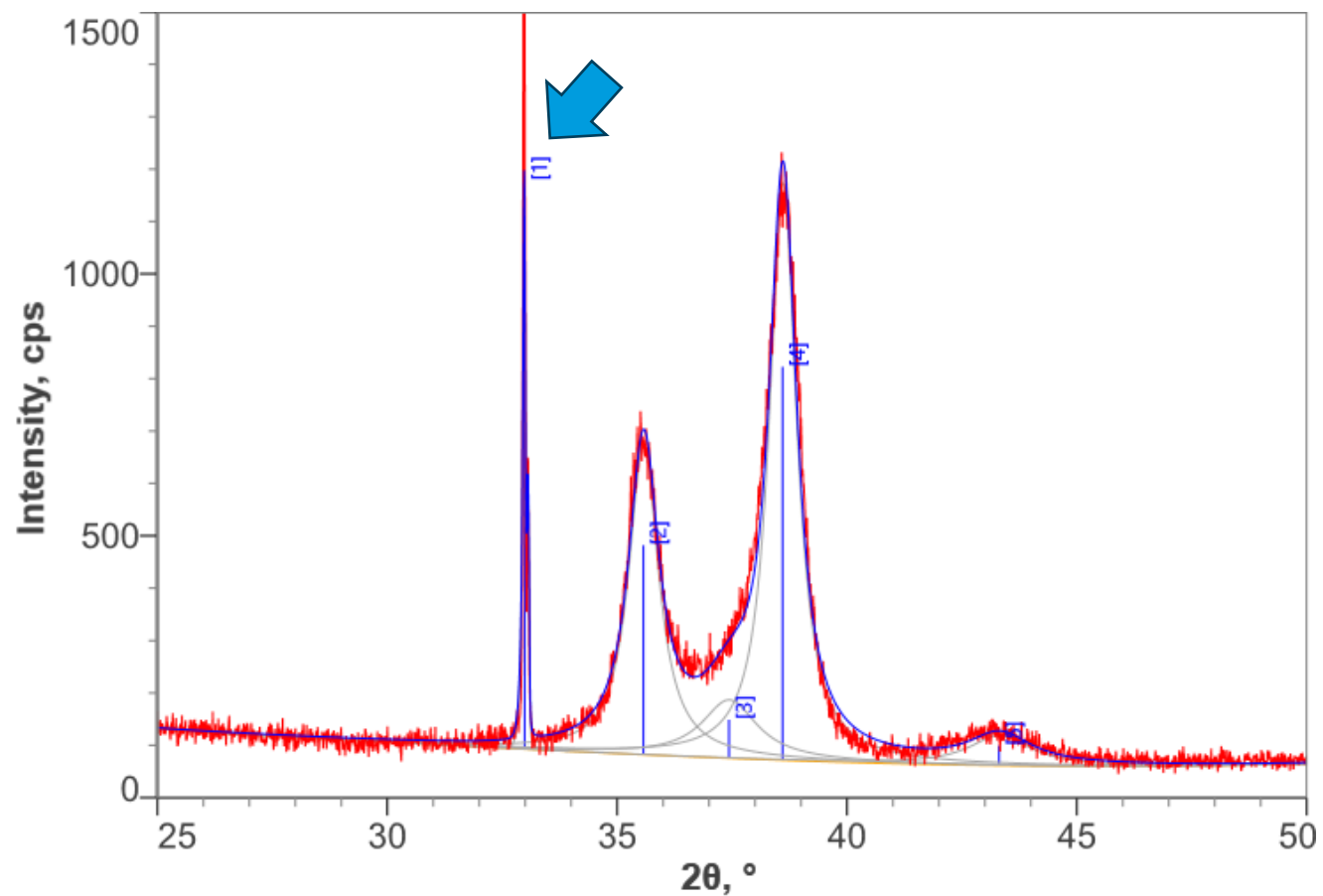
$\beta = \sqrt{\beta_{measured}^2 - \beta_{instrument}^2}$  : FWHM of Gaussian

$\theta$ : Bragg angle

Peak FWHM  $\beta_m$  dependence of crystallite size  $D$   
 $\beta_I = 0.2^\circ$ ,  $2\theta = 40^\circ$ , CuK $\alpha$



# Symmetric $2\theta$ - $\theta$ scan



- Is peak 1 silicon 002, which is a forbidden diffraction?

## research papers



Journal of  
**Applied  
Crystallography**

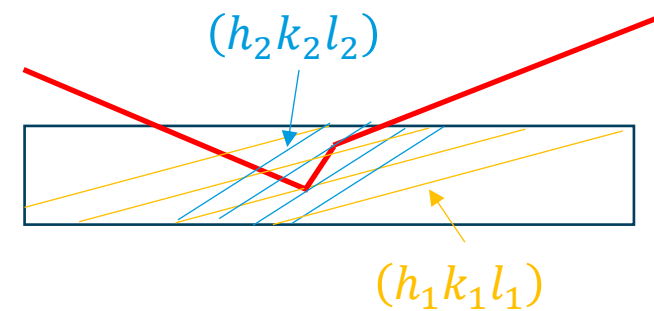
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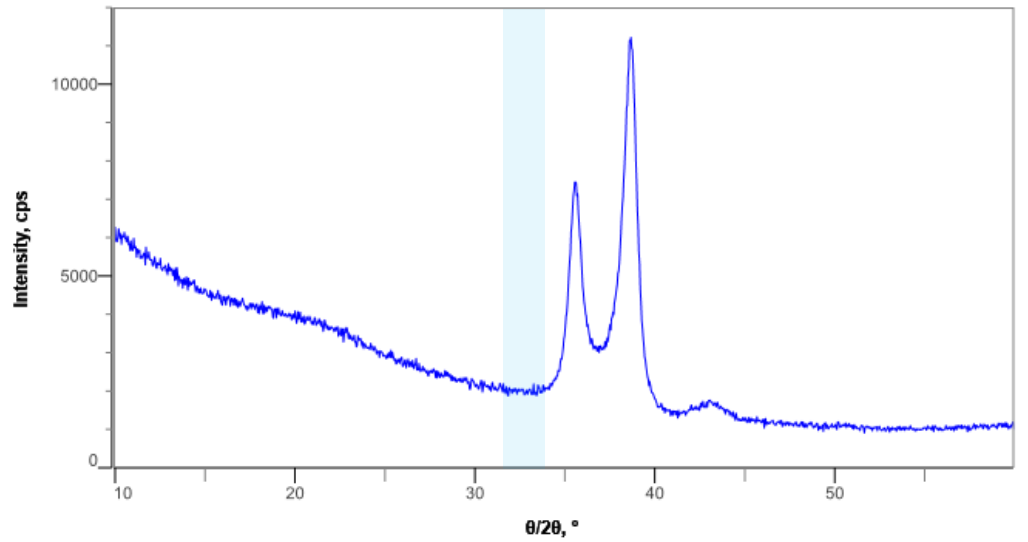
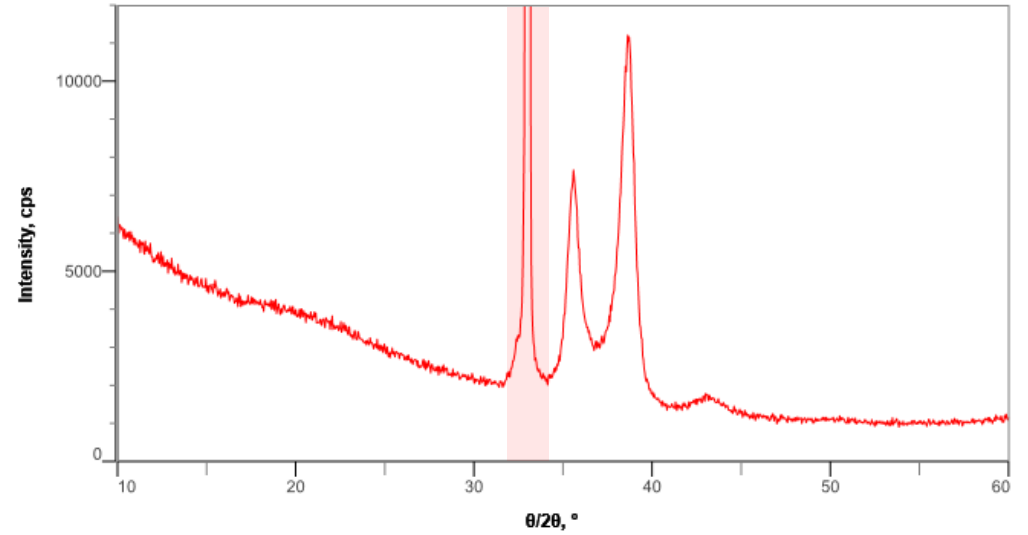
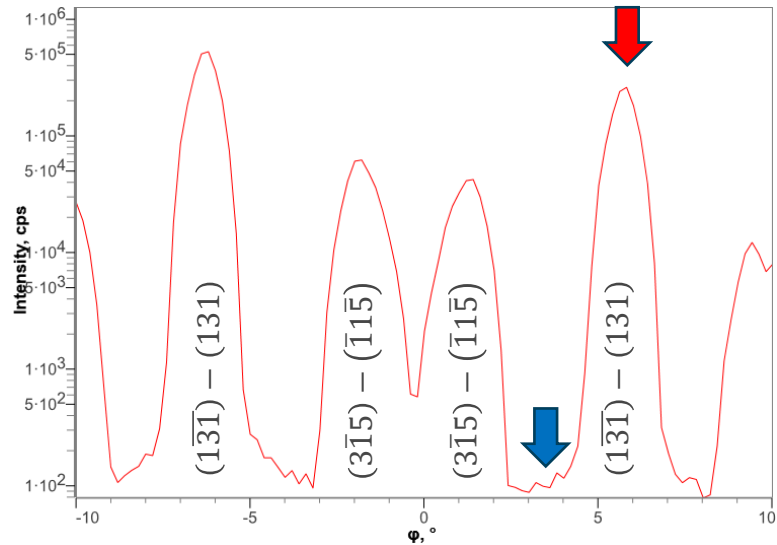
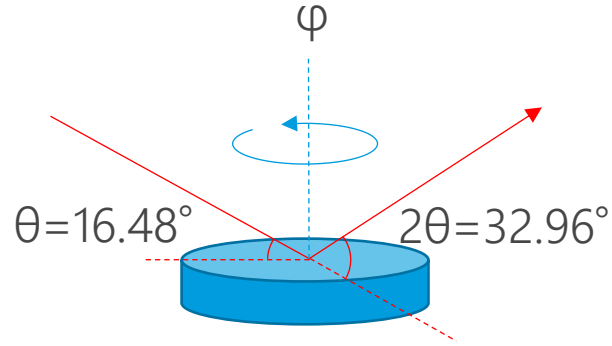
# High-resolution characterization of the forbidden Si 200 and Si 222 reflections<sup>1</sup>

**Peter Zaumseil**

IHP, Im Technologiepark 25, Frankfurt (Oder), 15236, Germany. Correspondence e-mail:  
zaumseil@ihp-microelectronics.com

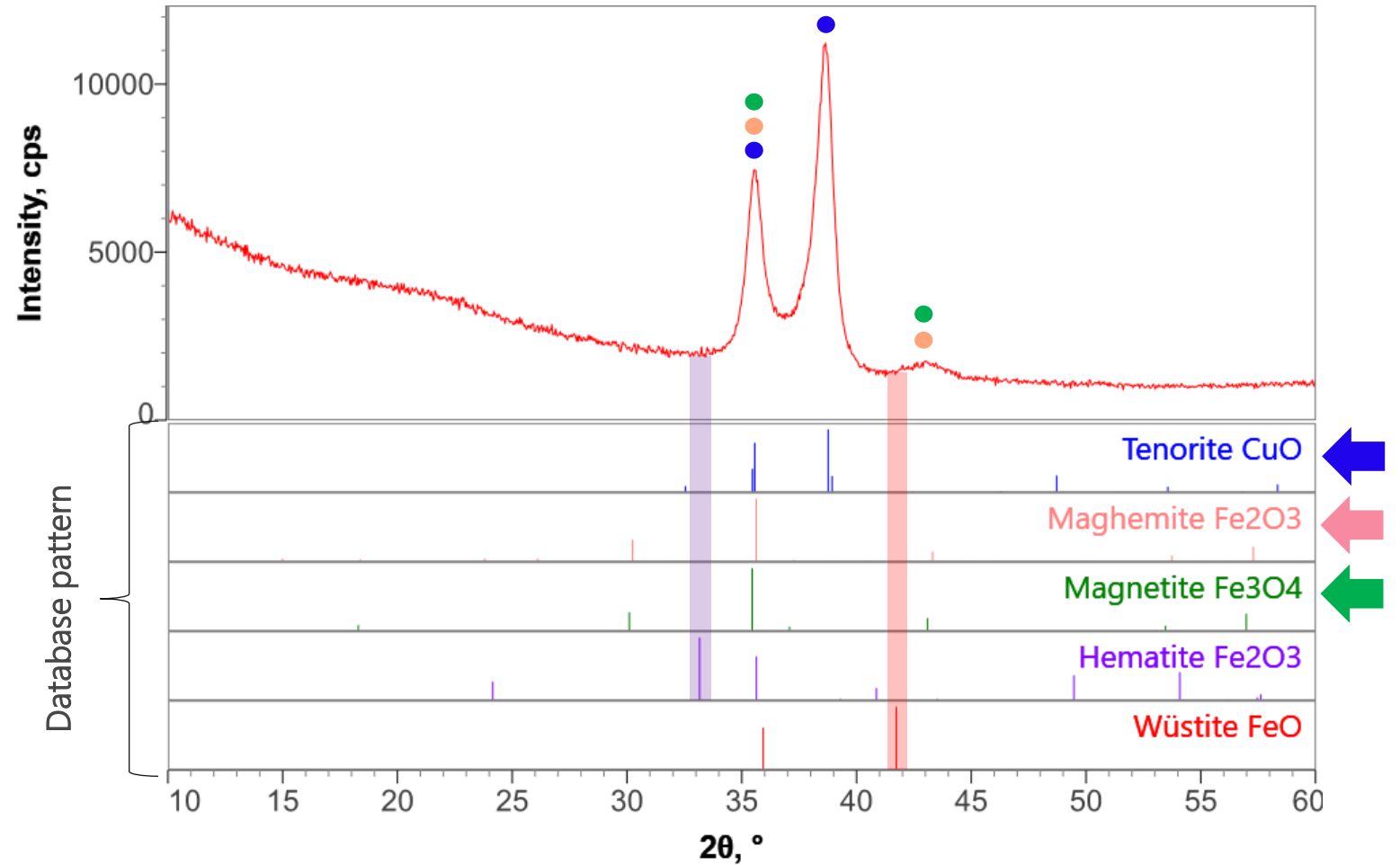


# Multiple diffraction



# 2 $\theta$ - $\theta$ scan - phase identification

- Cu-O layer is Tenorite.
- Fe-O layer could be **Maghemite** or/and **Magnetite**.
- Can we probe surface layer only?



# Probing surface layers with X-rays

- Penetration depth ( $D$ ) of X-ray around  $\theta_c$

$$D = \frac{\lambda}{4\pi B}$$

$$2B^2 = (-\omega^2 + 2\delta) + [(\omega^2 - 2\delta)^2 + 4\beta^2]^{1/2}$$

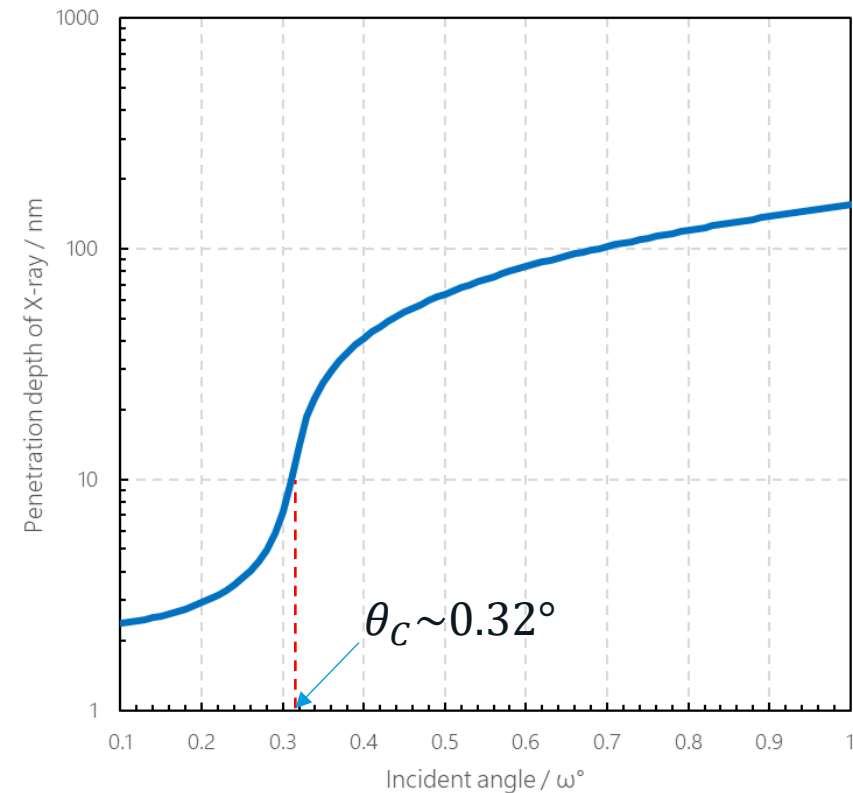
$n = 1 - \delta + i\beta$  : refractive index

$\theta_c \sim \sqrt{2\delta}$  : critical angle for total reflection

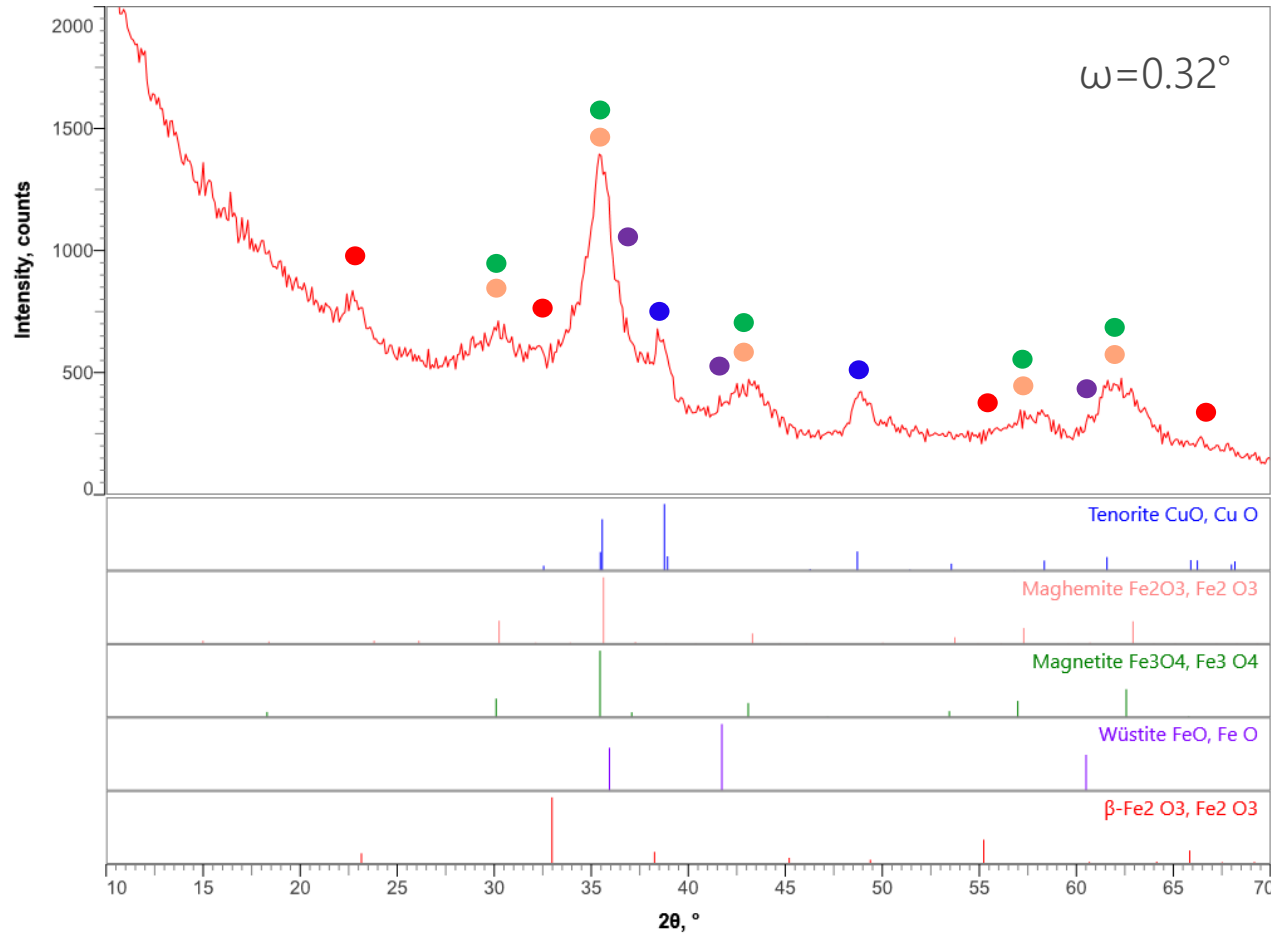
$\lambda$ : wavelength of X-ray

$\omega$ : incident angle of X-ray

Penetration depth of Cu-K $\alpha$  X-ray for Fe<sub>2</sub>O<sub>3</sub> Maghemite

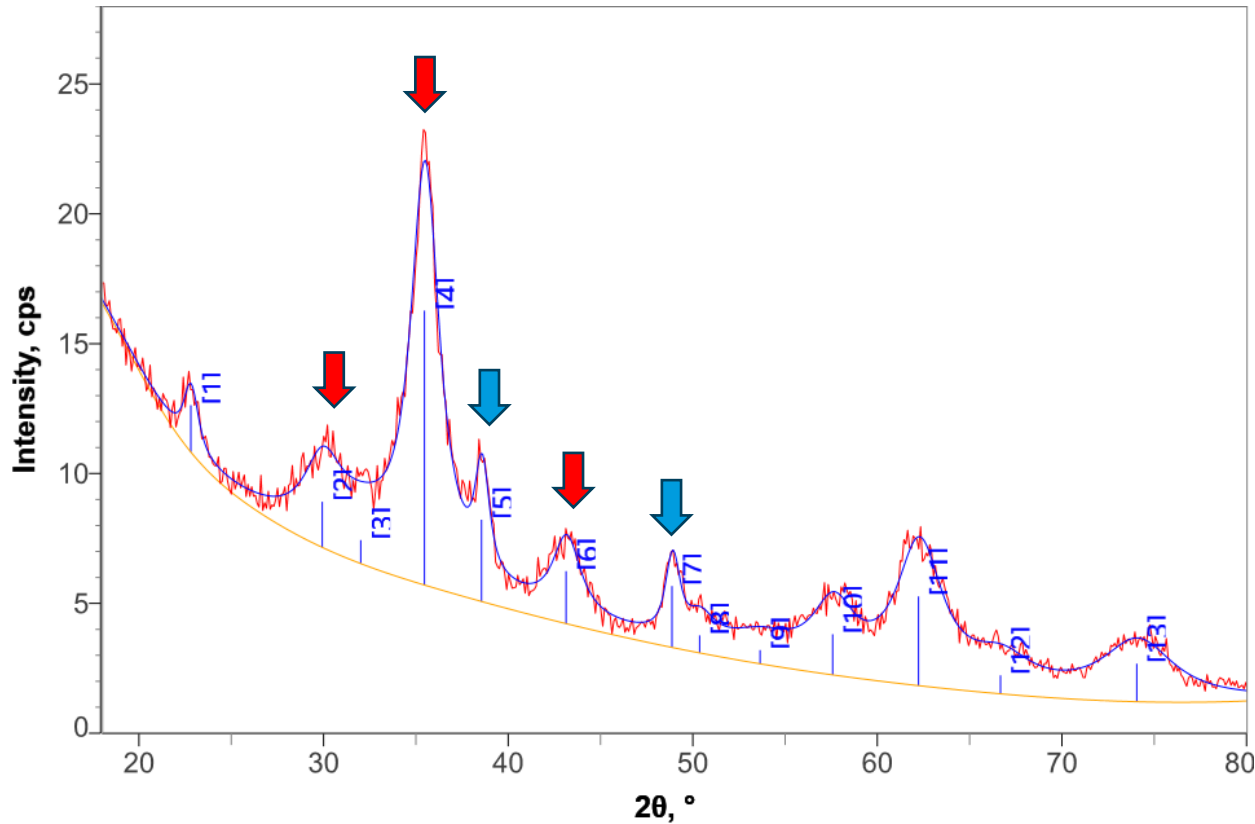


# In-plane scan & phase identification



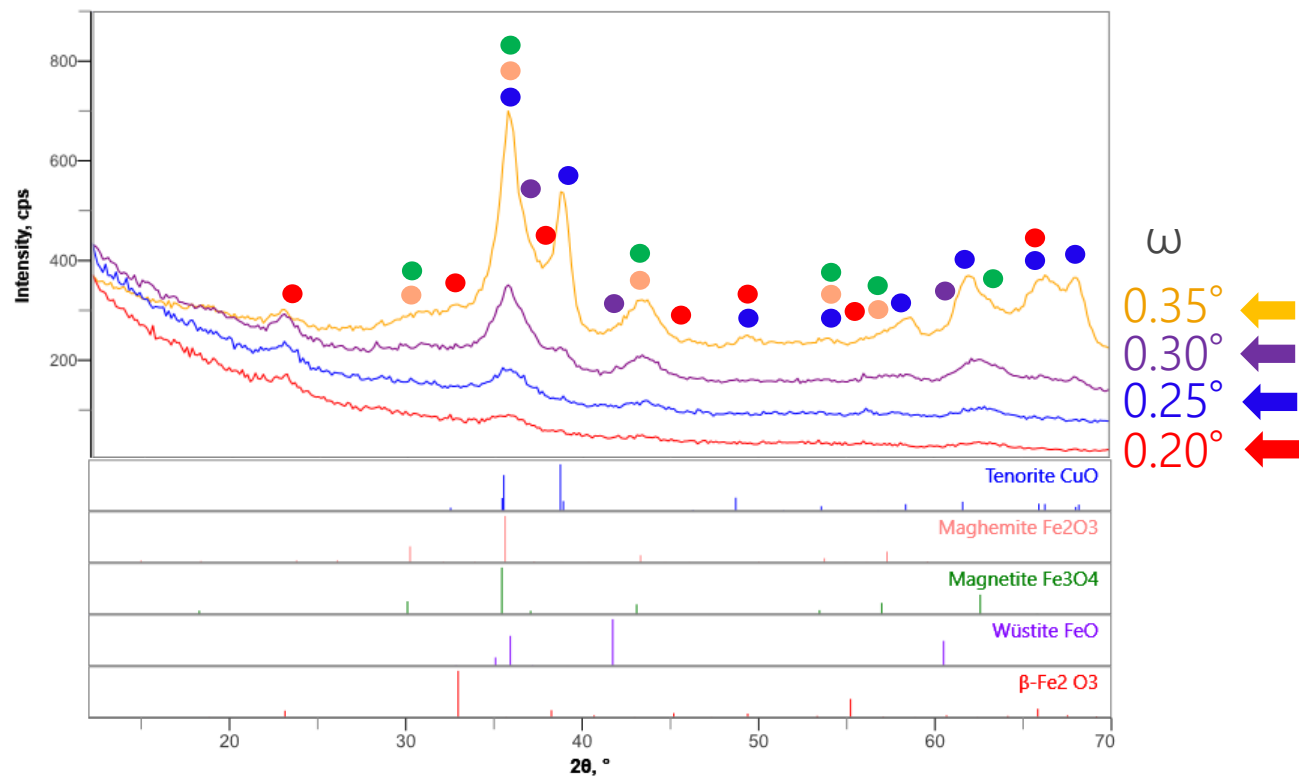
- Small CuO peaks
- Fe-O layer
  - Maghemite
  - Magnetite
  - Wüstite
  - $\beta$ -Fe<sub>2</sub>O<sub>3</sub>

# In-plane XRD

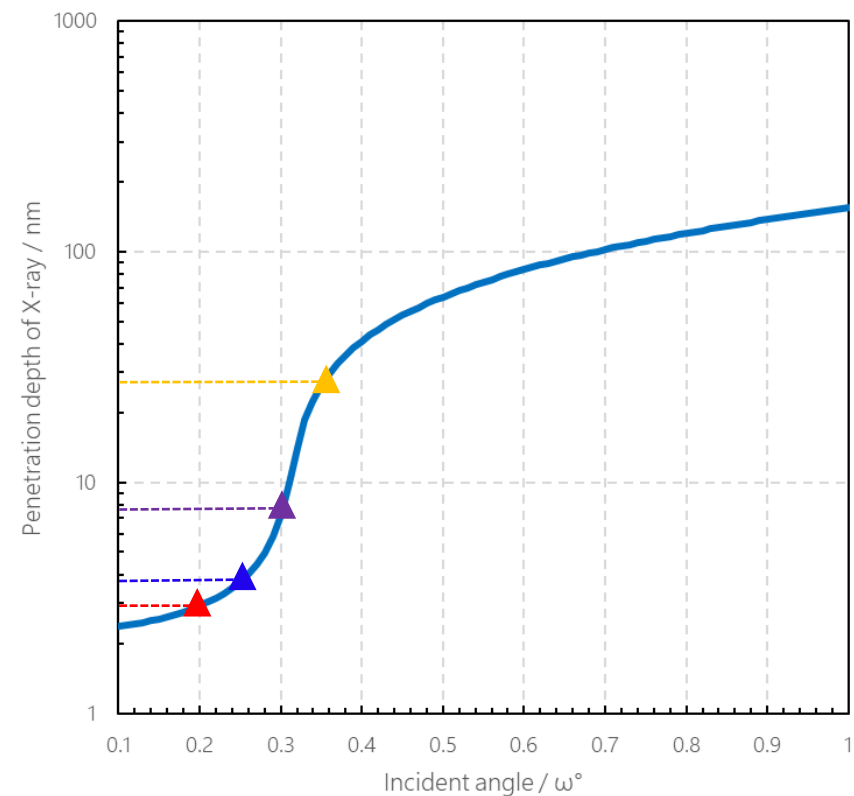


- Peak FWHM → Crystallite size
  - Scherrer equation
- Peaks 5 & 7 (CuO)
  - Peak FWHMs ~ 0.9°
  - Crystallite sizes ~ 11 nm  
~Half thickness
- Peaks 2 & 4 & 6 (Fe-O)
  - Peak FWHMs ~ 2.3°
  - Crystallite sizes ~ 4 nm  
~ Thickness

# GIXRD – incident angle dependence



Penetration depth of Cu-K $\alpha$  X-ray for Fe<sub>2</sub>O<sub>3</sub> Maghemite

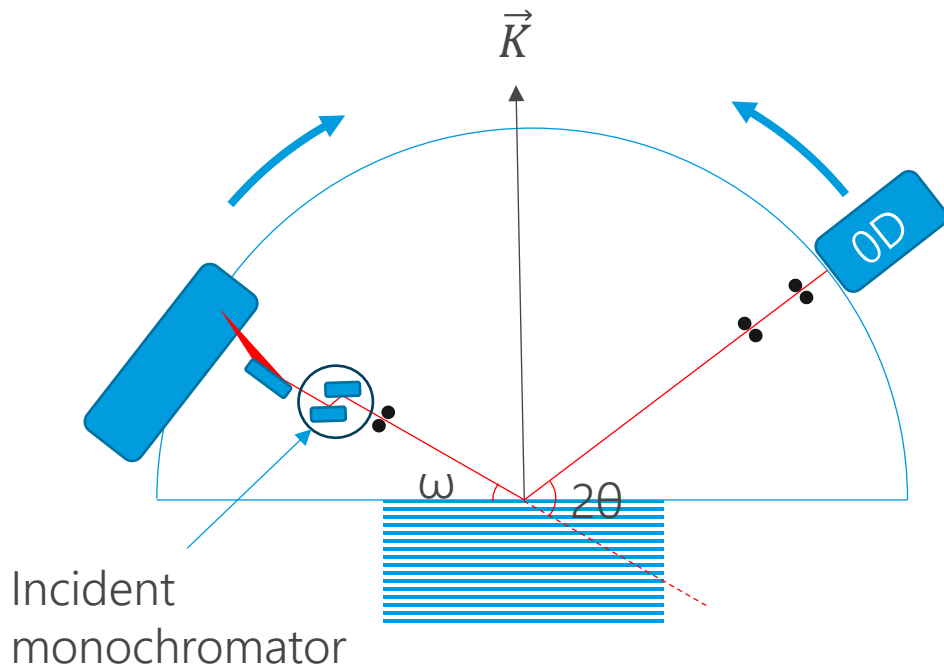


# Summary

- $2\theta$ - $\theta$  scan
  - Surface normal information
  - Not sensitive to thin layer(s) but should be done first (!)
- In-plane XRD
  - Sensitive to surface layer(s), depth profile
- Grazing incidence XRD
  - Same as in-plane scan but only applicable on non-textured thin films

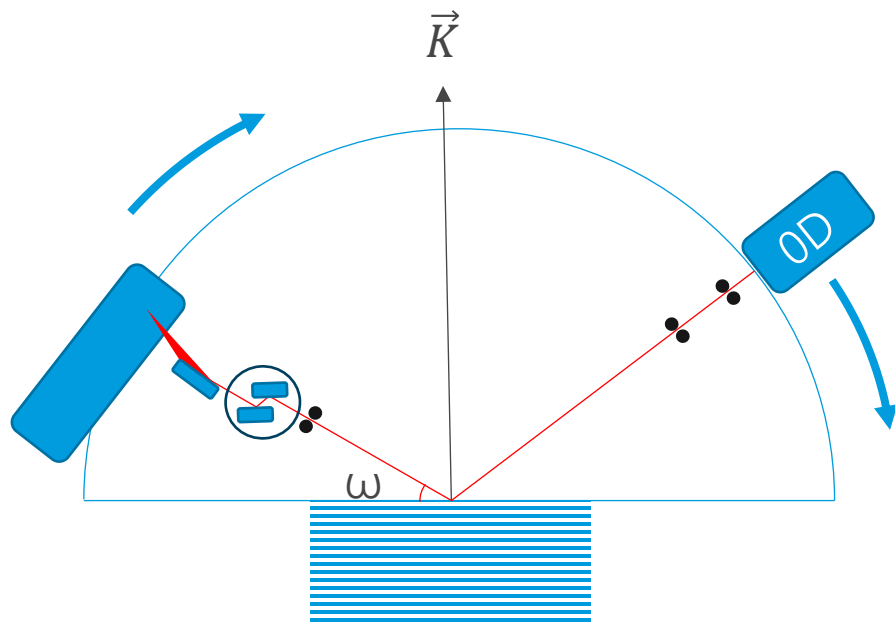
# *4. High-resolution X-ray diffraction measurement types & instrument configurations*

# Symmetric $2\theta$ - $\omega$ scan, $\omega = \theta$

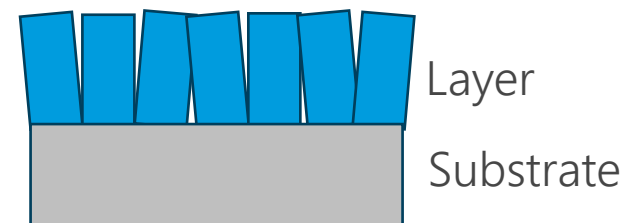


- d-spacings  $\perp$  from surface
- Lattice mismatch
- Thickness from fringes / satellites

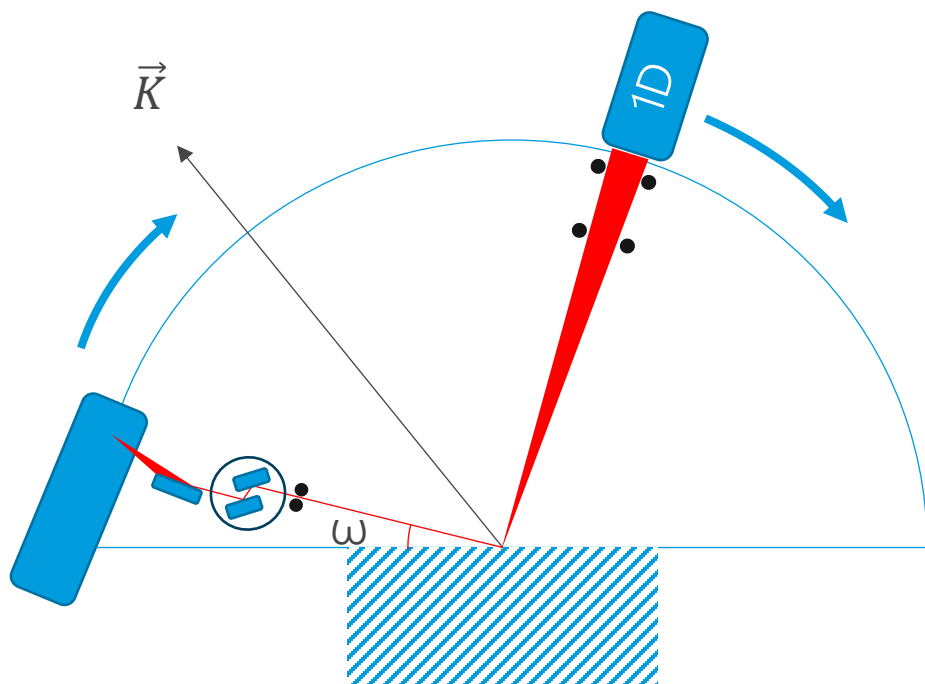
# Rocking curve ( $\omega$ scan)



- Mosaicity
- Dislocation density

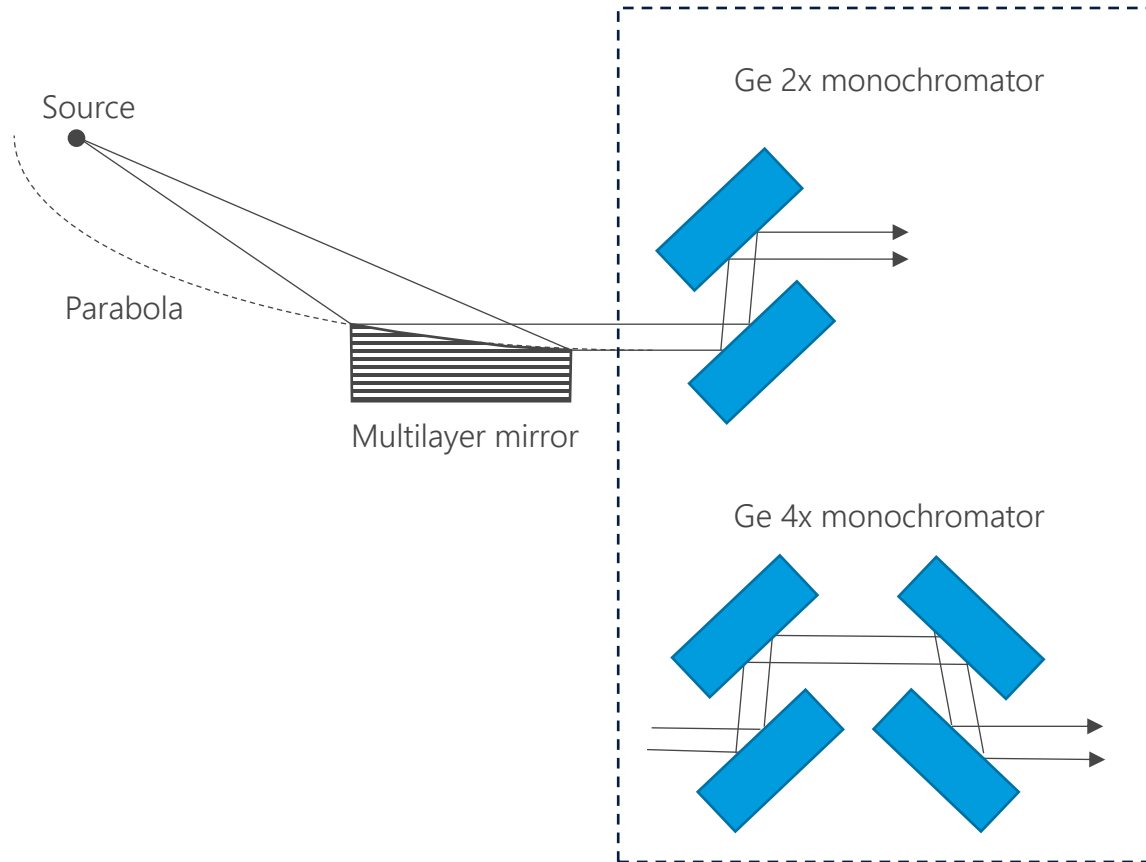


# Reciprocal space mapping ( $\omega$ scan + $2\theta$ covered by 1D det.)



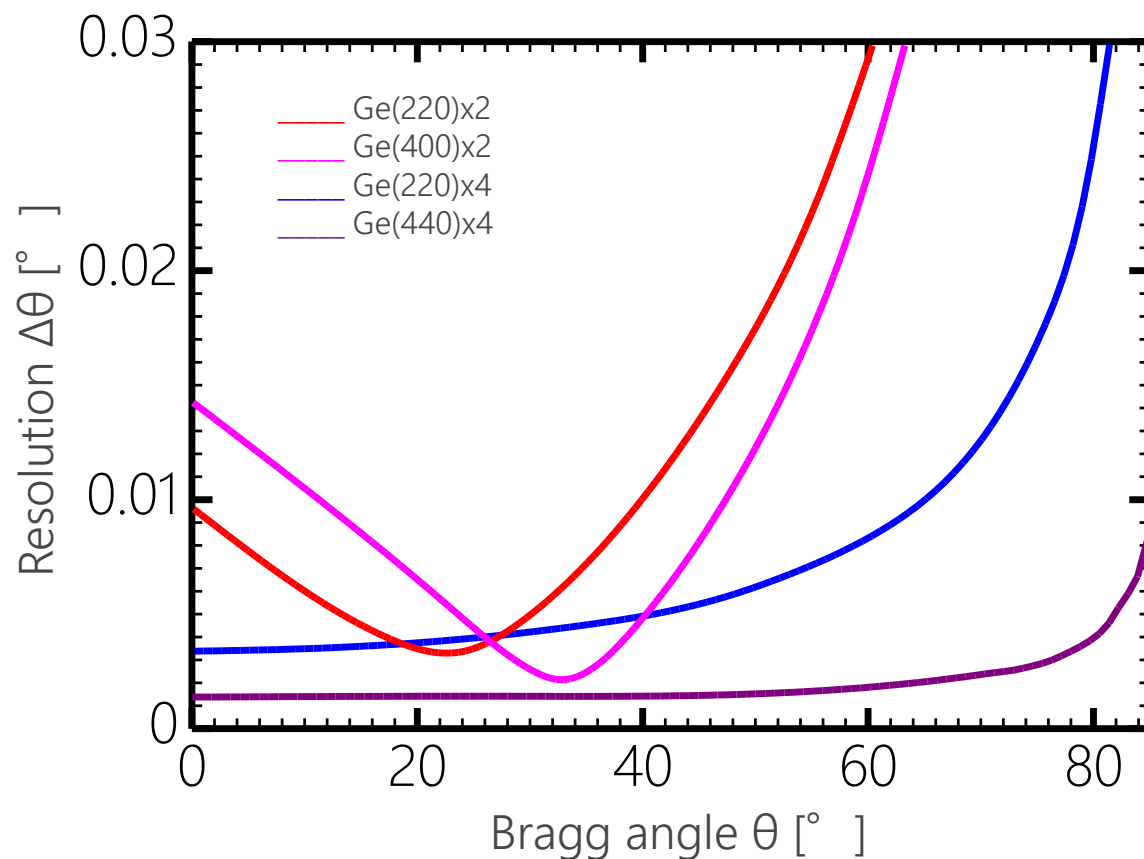
- d-spacings both  $\perp$  &  $\parallel$  to surface
- Stress & relaxation

# Channel cut monochromator



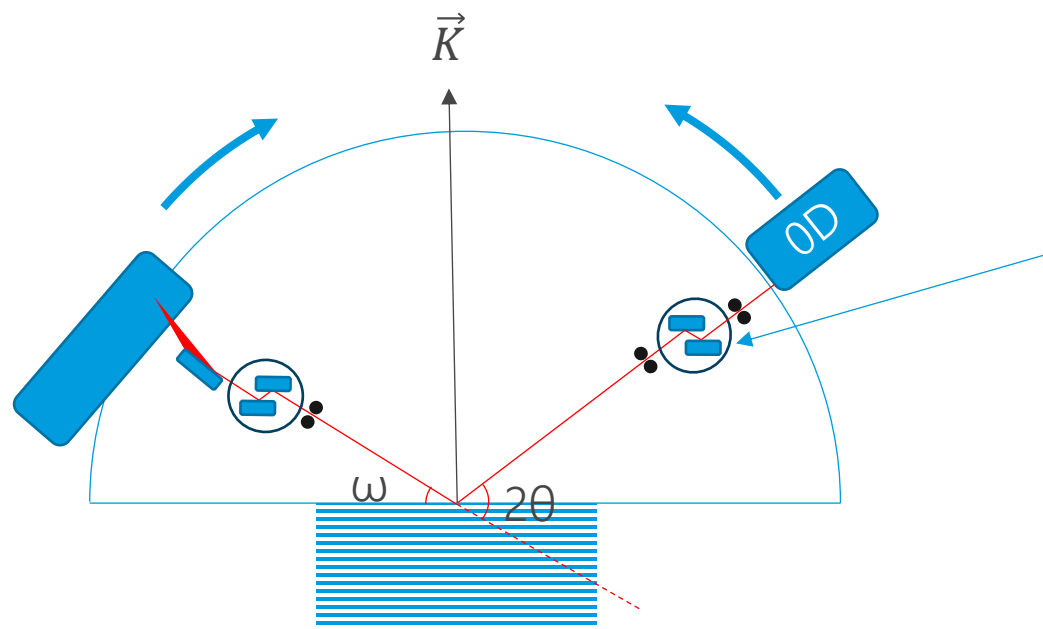
- 2 bounce monochromator
  - Pros: High intensity
  - Cons: Large dispersion
- 4 bounce monochromator
  - Pros: Small dispersion
  - Cons: Low intensity

# Monochromator resolution



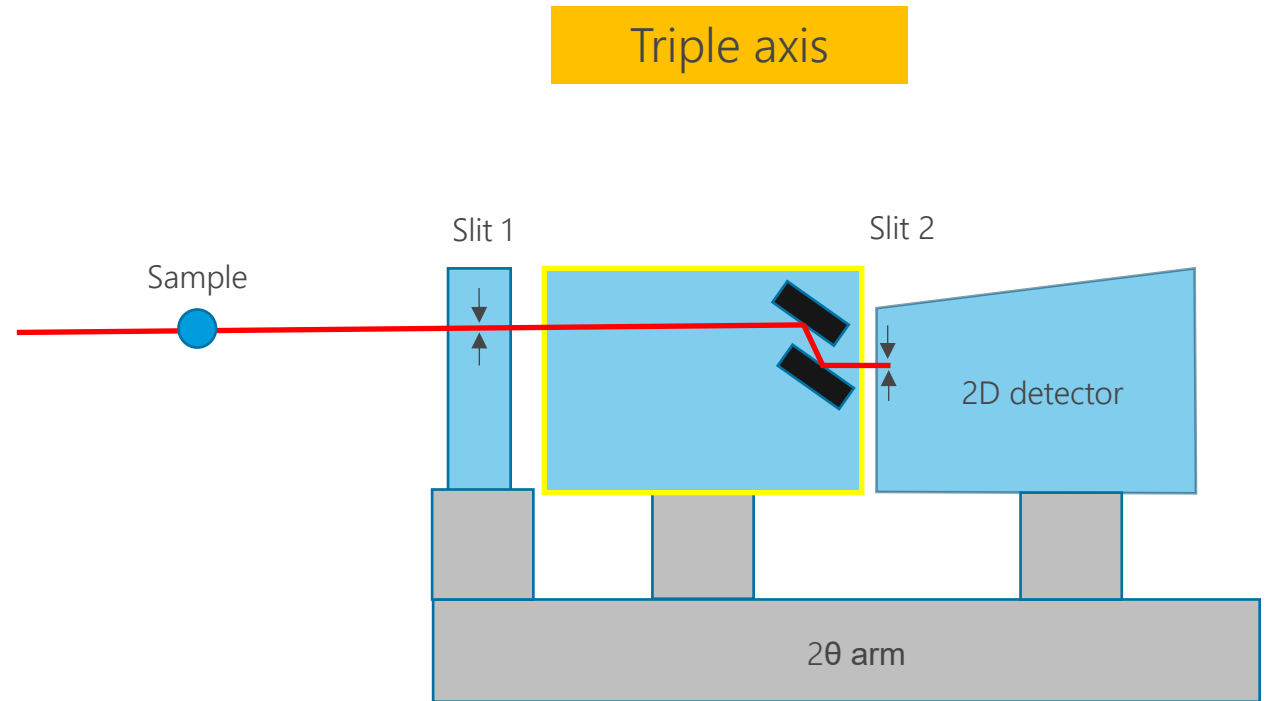
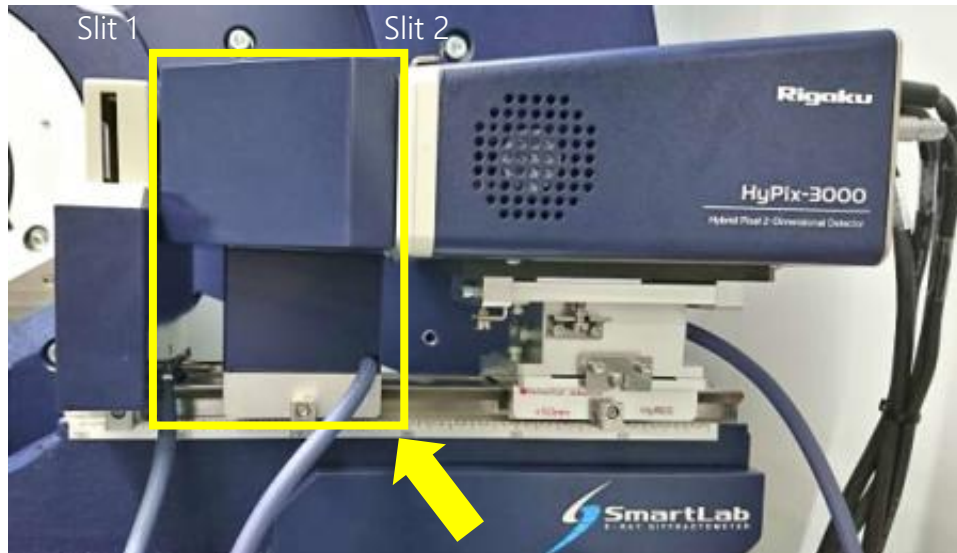
Primary monochromator	Relative intensity	Applications examples
Ge(220) 2-bounce	~ 360	GaN, ZnO, SiC, oxides, metals
Ge(400) 2-bounce	~ 100	Si, GaAs, InP, ZnSe, SiGe
Ge(220) 4-bounce	~ 20	Epitaxial thin films from low to high $2\theta$ angle
Ge(440) 4-bounce	1	Substrate analysis

# Symmetric $2\theta$ - $\omega$ triple axis scan

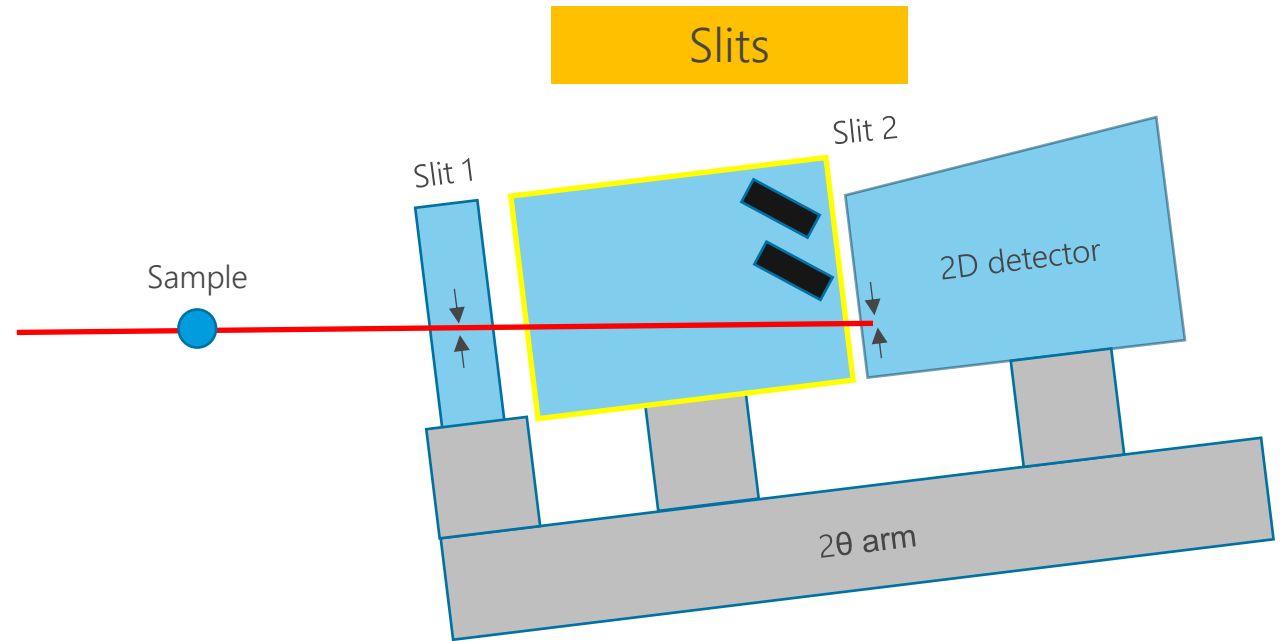
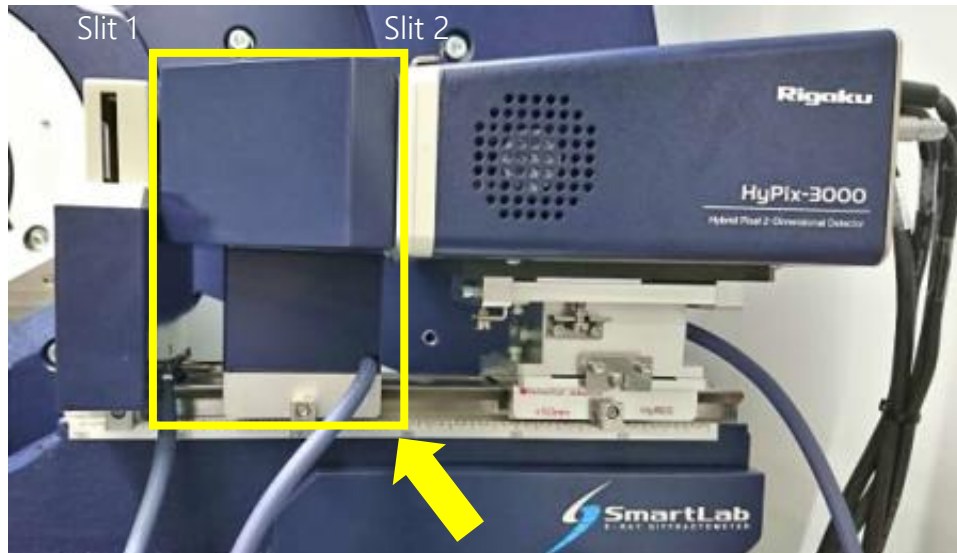


- Ge-channel cut analyzer crystal
- $\Delta 2\theta \sim 0.003^\circ$
- Resolution is too high for sample alignment

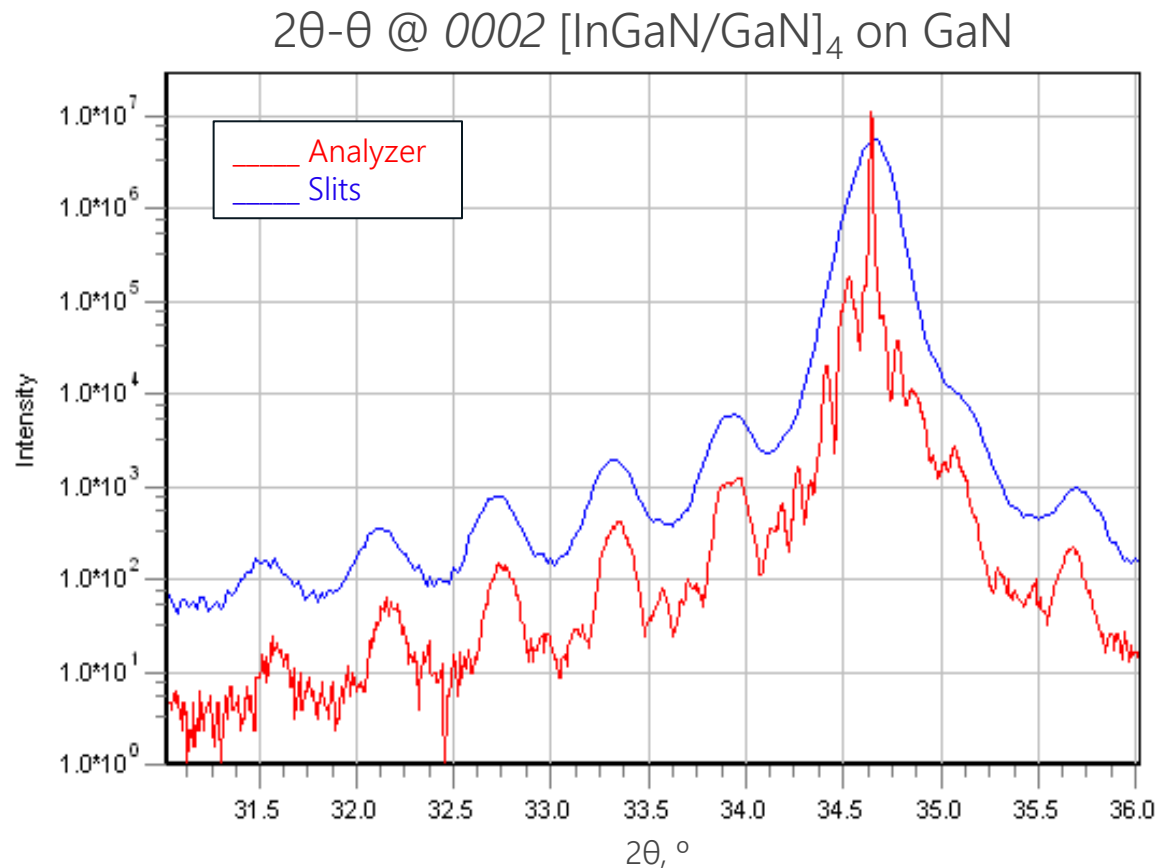
# Receiving analyzer crystal



# Receiving analyzer crystal



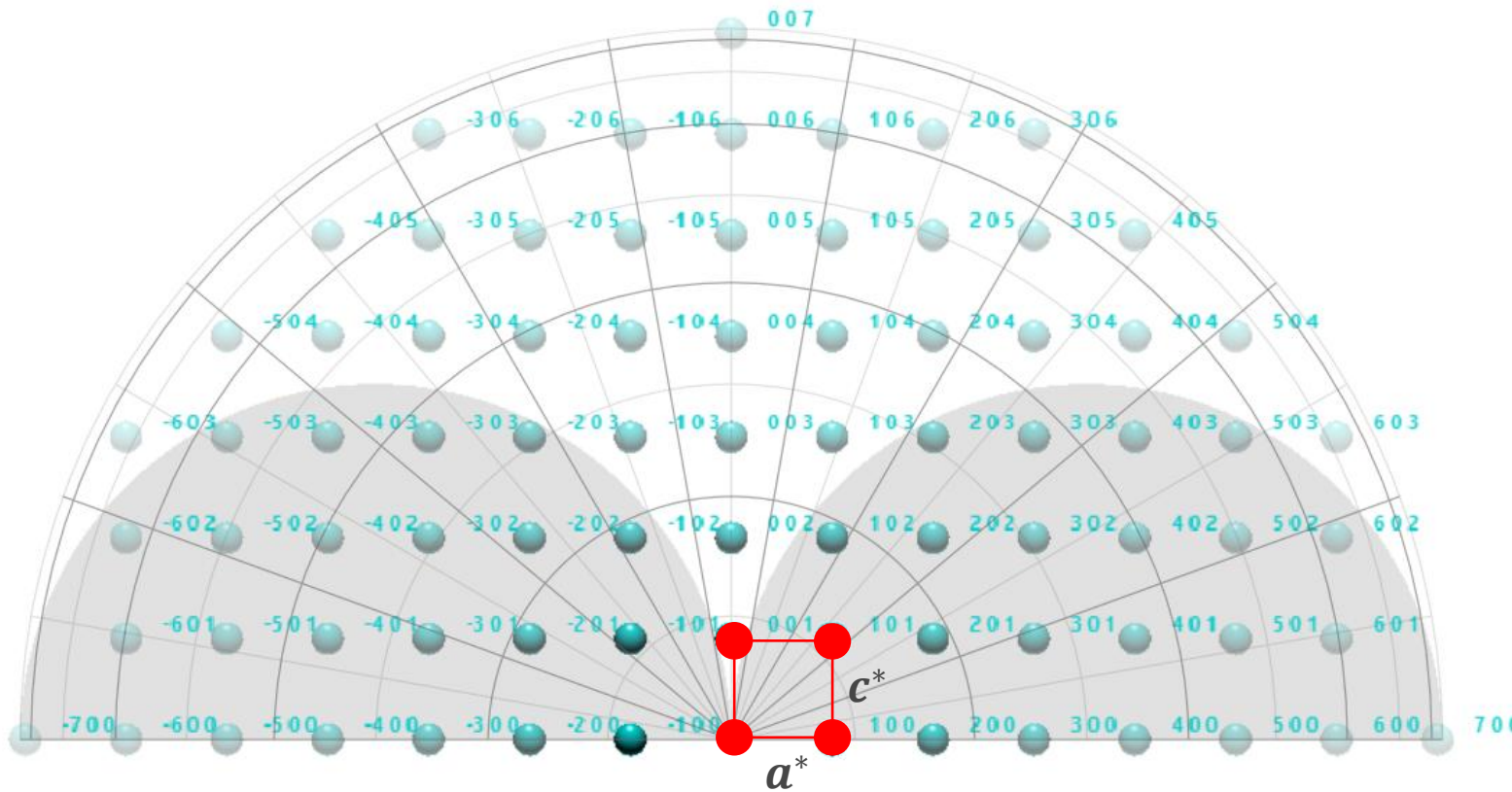
# Slit path vs. Analyzer path



- Analyzer
  - Suppresses peak broadening along  $\omega$
  - Film with mosaicity, e.g. GaN
- Slits
  - High quality film, e.g. III-V compound semiconductors

# *5. HRXRD scans in reciprocal space*

# Reciprocal lattice point



Crystal: a cubic crystal

$$\vec{g}_{hkl} = h\vec{a}^* + k\vec{b}^* + l\vec{c}^*$$

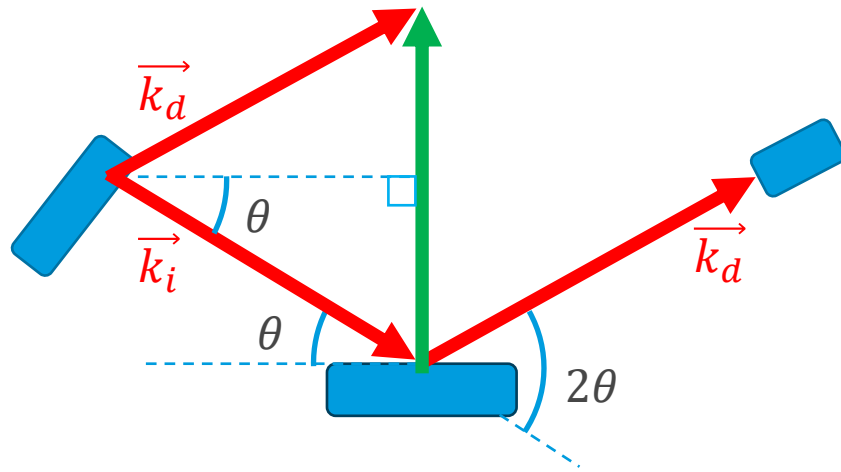
$$\vec{a}^* = \frac{\vec{b} \times \vec{c}}{v_c}, \vec{b}^* = \frac{\vec{a} \times \vec{c}}{v_c}, \vec{c}^* = \frac{\vec{a} \times \vec{b}}{v_c}$$

$$|\vec{g}_{hkl}| = \frac{1}{d_{hkl}}$$

$\vec{g}_{hkl}$ : reciprocal lattice vector  
 $\vec{a}^*, \vec{b}^*, \vec{c}^*$ : unit vector of a reciprocal lattice

$v_c$ : Volume of a unit cell

# Bragg equation



Bragg equation

$$2 \cdot d_{hkl} \cdot \sin \theta = \lambda$$

$$\lambda = \frac{1}{|\vec{k}_i|} = \frac{1}{|\vec{k}_d|}$$

$$\therefore 2 \cdot |\vec{k}| \cdot \sin \theta = \frac{1}{d_{hkl}} = |\vec{k}_d - \vec{k}_i|$$

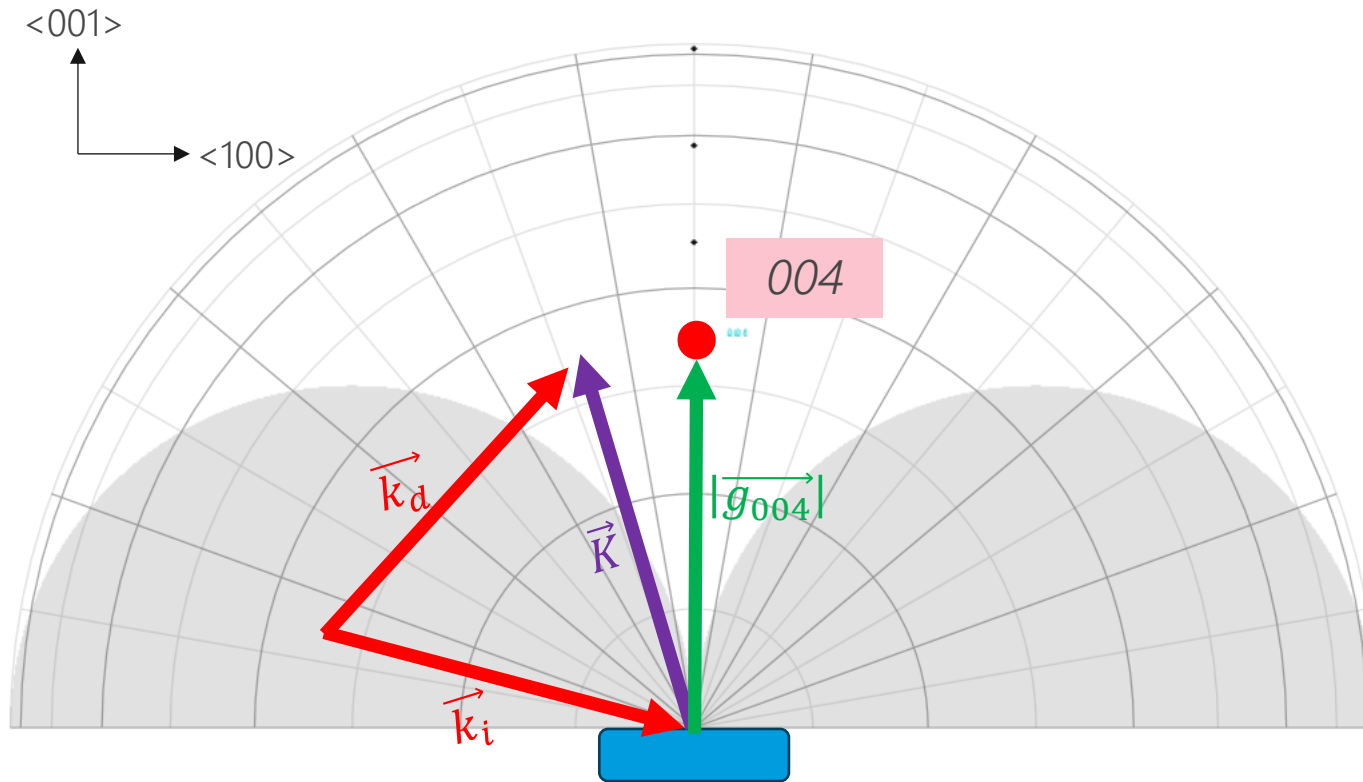
$d_{hkl}$ : interplanar spacing of  $hkl$  plane

$\lambda$ : wavelength of X-ray

$\vec{k}_i$ : incident X-ray wave vector

$\vec{k}_d$ : diffracted X-ray wave vector

# Diffraction condition in reciprocal space



Scattering vector  $\vec{K}$

$$\vec{K} = \vec{k}_d - \vec{k}_i$$

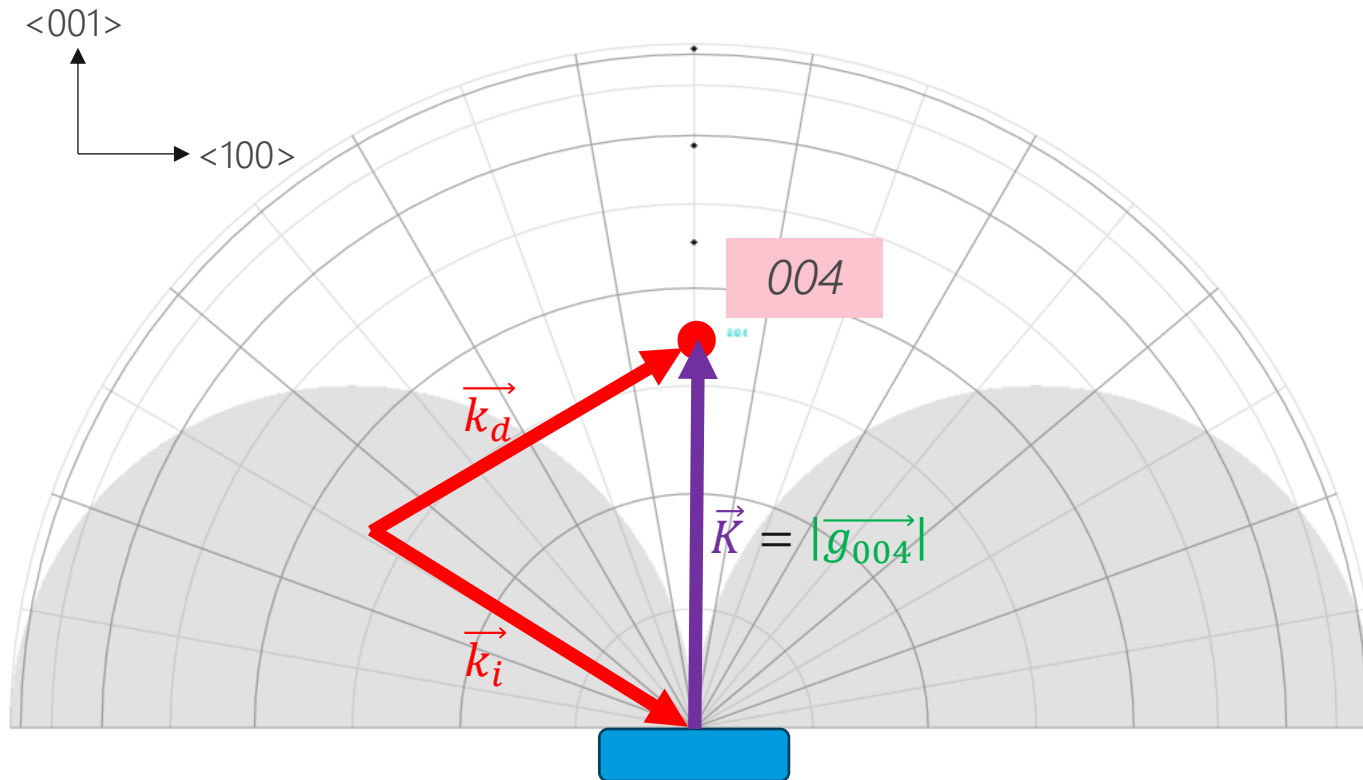
Diffraction is observed when

$$\vec{K} = \vec{g}_{hkl}$$

$\vec{K}$  : scattering vector

$\vec{g}_{hkl}$  : reciprocal lattice vector

# Diffraction condition in reciprocal space



Scattering vector  $\vec{K}$

$$\vec{K} = \vec{k}_d - \vec{k}_i$$

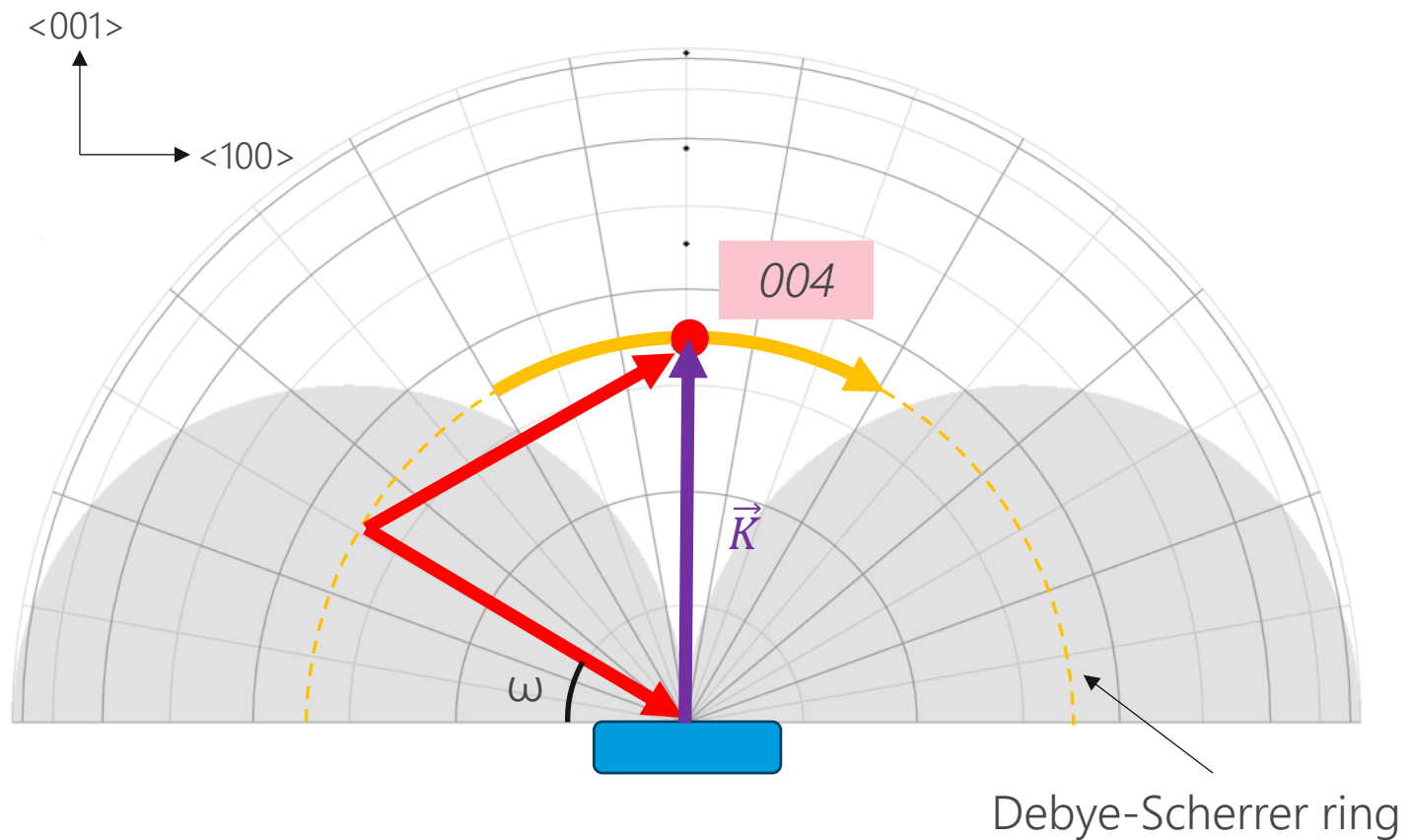
Diffraction is observed when

$$\vec{K} = \vec{g}_{hkl}$$

$\vec{K}$  : scattering vector

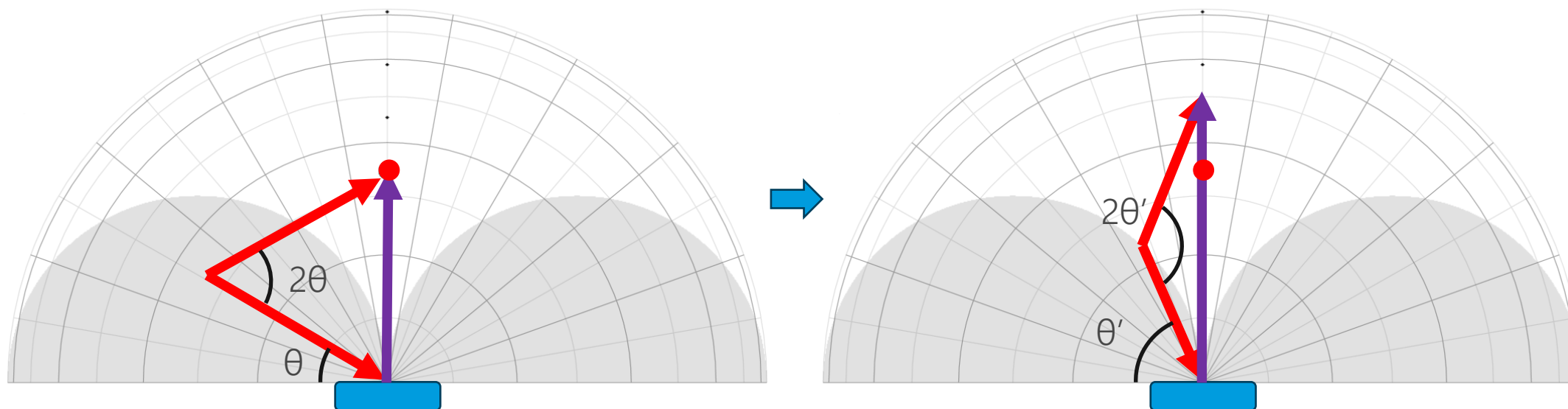
$\vec{g}_{hkl}$  : reciprocal lattice vector

# Omega scan



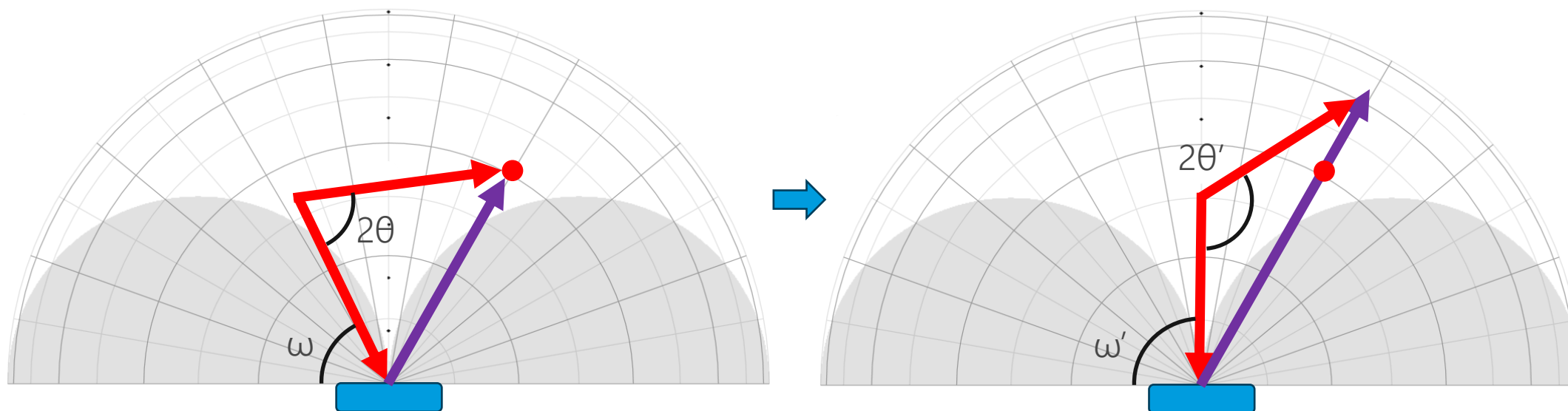
- $\omega$  scan || Debye-Scherrer ring
- Also called "rocking curve"

## $2\theta$ - $\theta$ scan



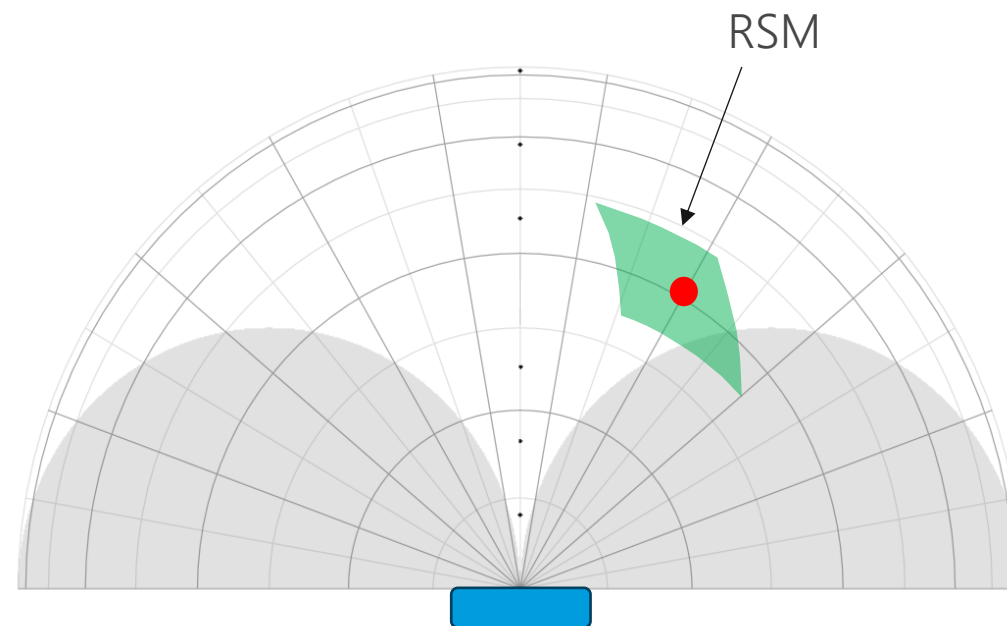
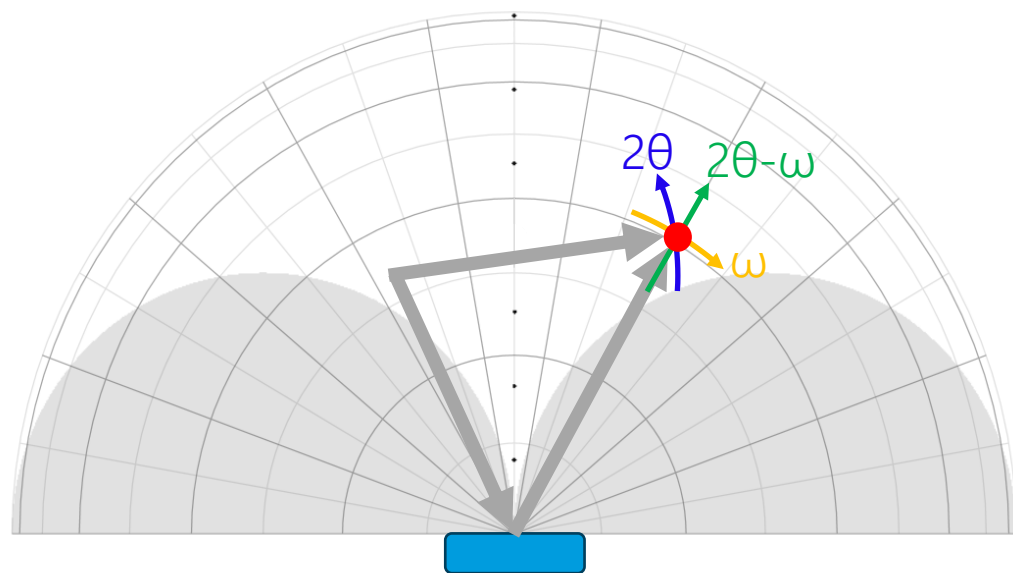
- $2\theta$ - $\theta$  scan  $\perp$  sample surface
- Also called “radial scan”

# 2 $\theta$ - $\omega$ scan



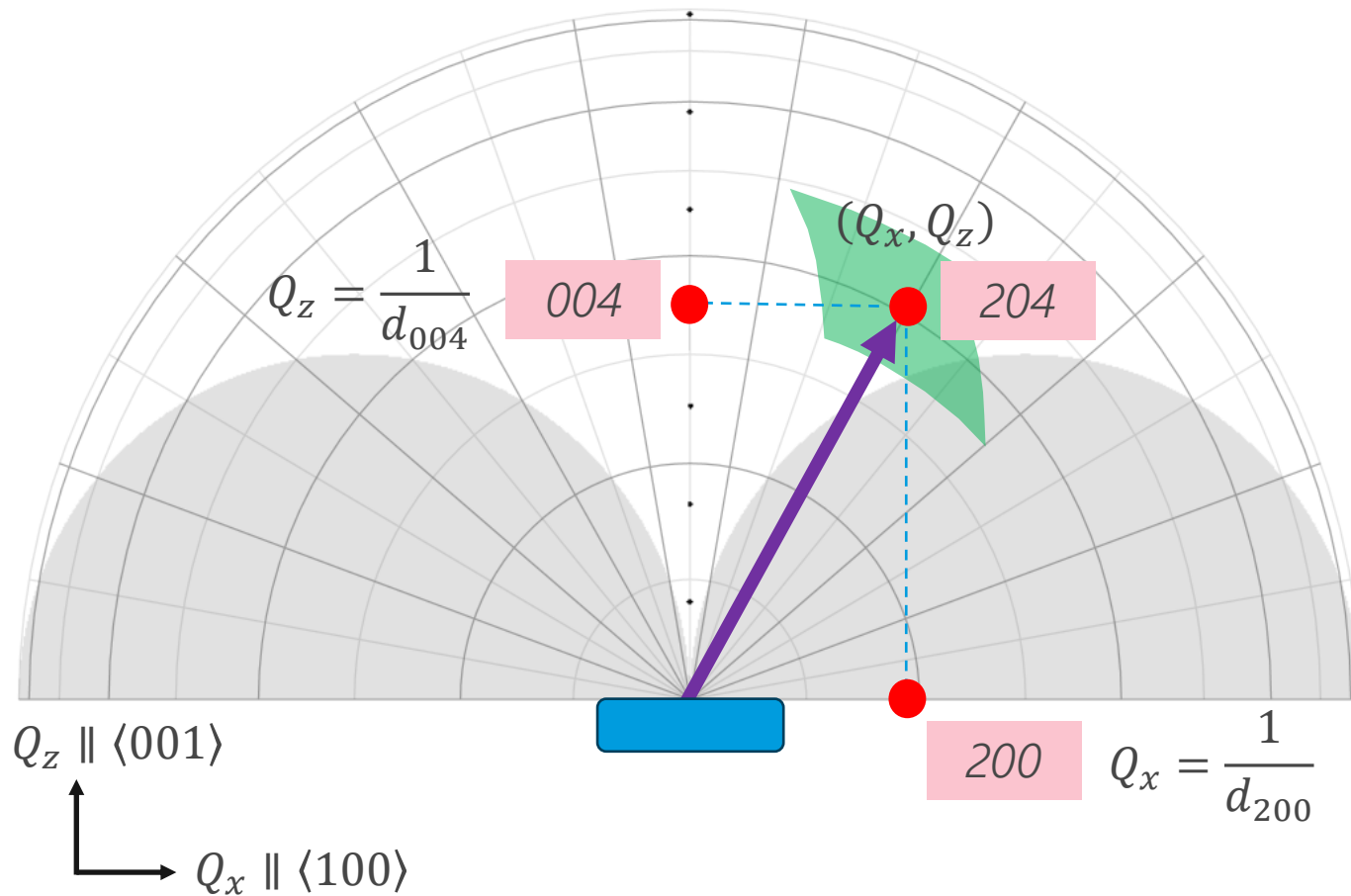
- $\theta \neq \omega$
- Radial scan along tilted direction

# Reciprocal space mapping (RSM)



- $2\theta - \omega$  scan +  $\omega$  offset
- $2\theta$  scan +  $\omega$  offset
- $\omega$  scan + 1D detector ( $2\theta$ )

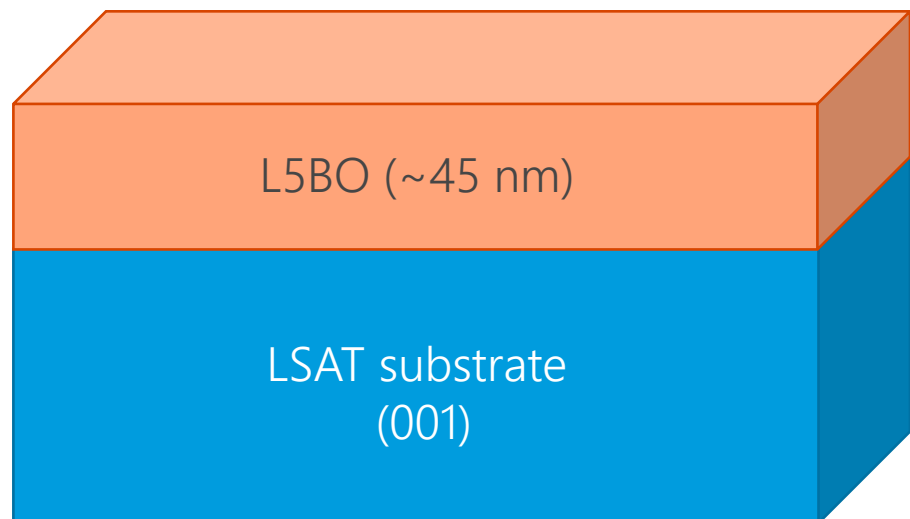
# Reciprocal space mapping (RSM)



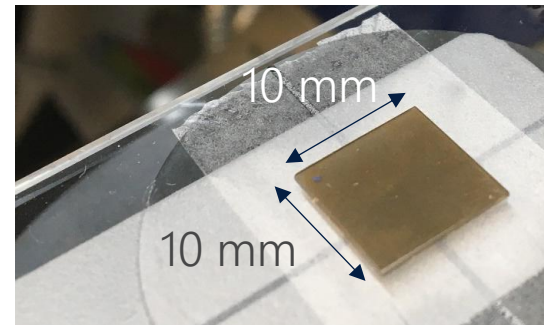
- RSM tells  $(Q_x, Q_z)$  of diffraction spot
- $Q_x \propto$  in-plane d-spacing
- $Q_z \propto$  out of plane d-spacing

# *6. Epitaxial perovskite thin films: structure and thickness*

# Sample

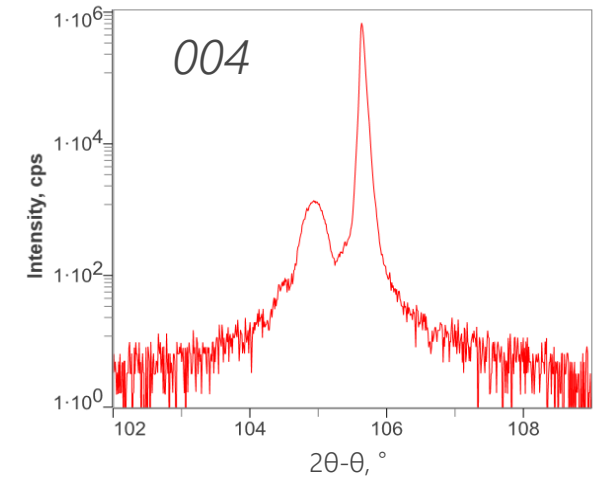
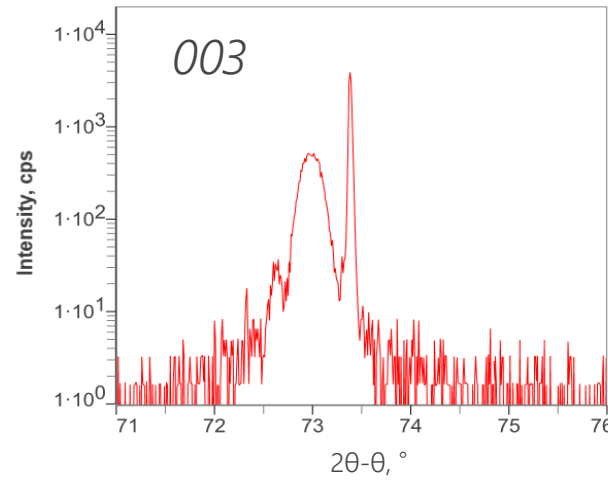
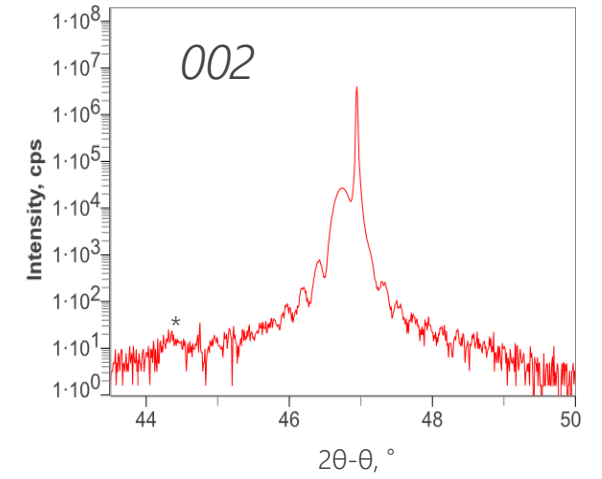
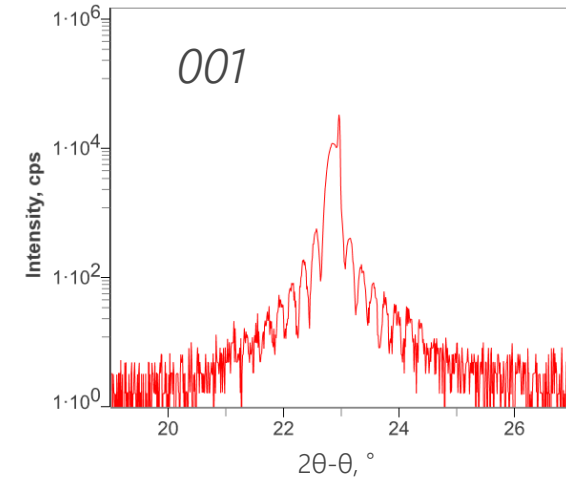
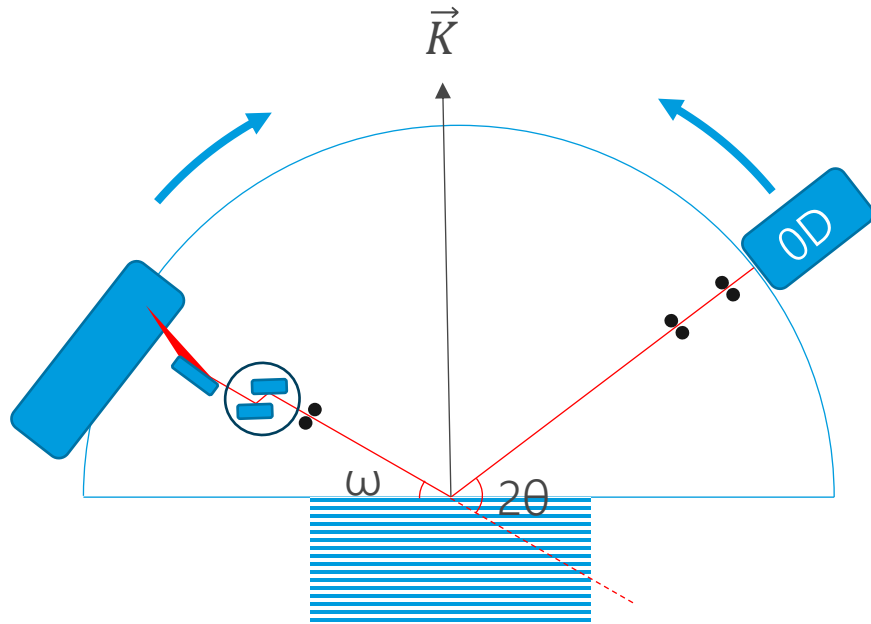


Sample courtesy:  
 Dr. Alessandro Mazza (Los Alamos Nat. Lab.)  
 Dr. Jim Browning (Oak Ridge Nat. Lab.)



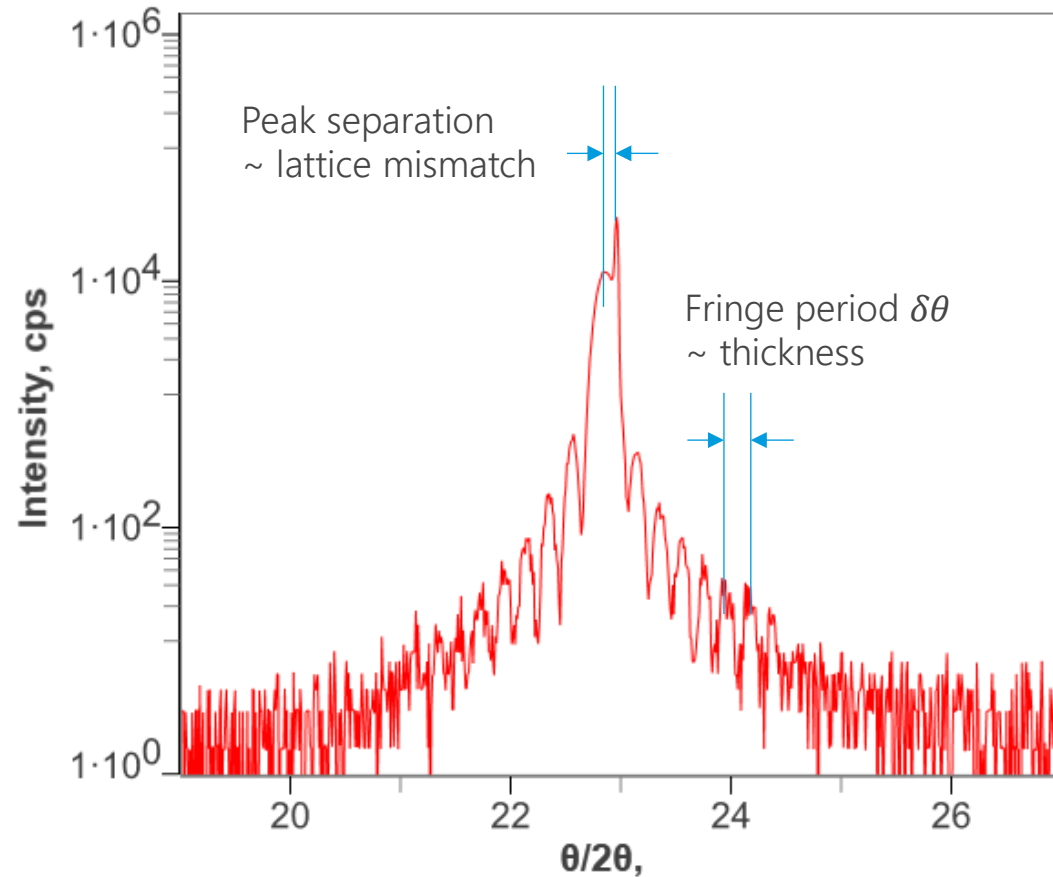
- L5BO
  - $\text{La}(\text{Cr}_{0.2}\text{Mn}_{0.2}\text{Fe}_{0.2}\text{Co}_{0.2}\text{Ni}_{0.2})\text{O}_3$
  - $\text{LaCrO}_3, \text{LaMnO}_3, \text{LaFeO}_3 \rightarrow$   
orthorhombic
  - $\text{LaCoO}_3, \text{LaNiO}_3 \rightarrow$   
rhombohedral
- LSAT
  - $(\text{La}_{0.18}\text{Sr}_{0.82})(\text{Al}_{0.59}\text{Ta}_{0.41})\text{O}_3$
  - $Pm\bar{3}m, a=0.3868 \text{ nm}$

# HR 2θ-θ scan



\* Unassigned peak

# Lattice mismatch & Thickness



- Lattice mismatch:  $M$

$$M = \frac{\Delta c}{c} = \frac{C_L - C_S}{C_S} = \frac{\sin \theta_S}{\sin \theta_L} - 1$$

$$= 0.00471 = 0.471 \%$$

- Layer's c-lattice parameter:  $C_L$

$$C_L = (1 + M) \cdot C_S = 0.3887 \text{ nm}$$

- Layer's thickness:  $t$

$$t = \frac{\lambda \cdot \sin \theta}{\sin 2\theta \cdot \delta\theta}$$

$$= 44.2 \text{ nm}$$

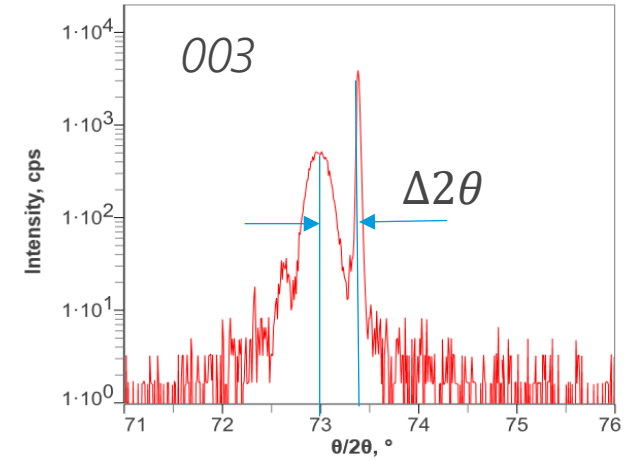
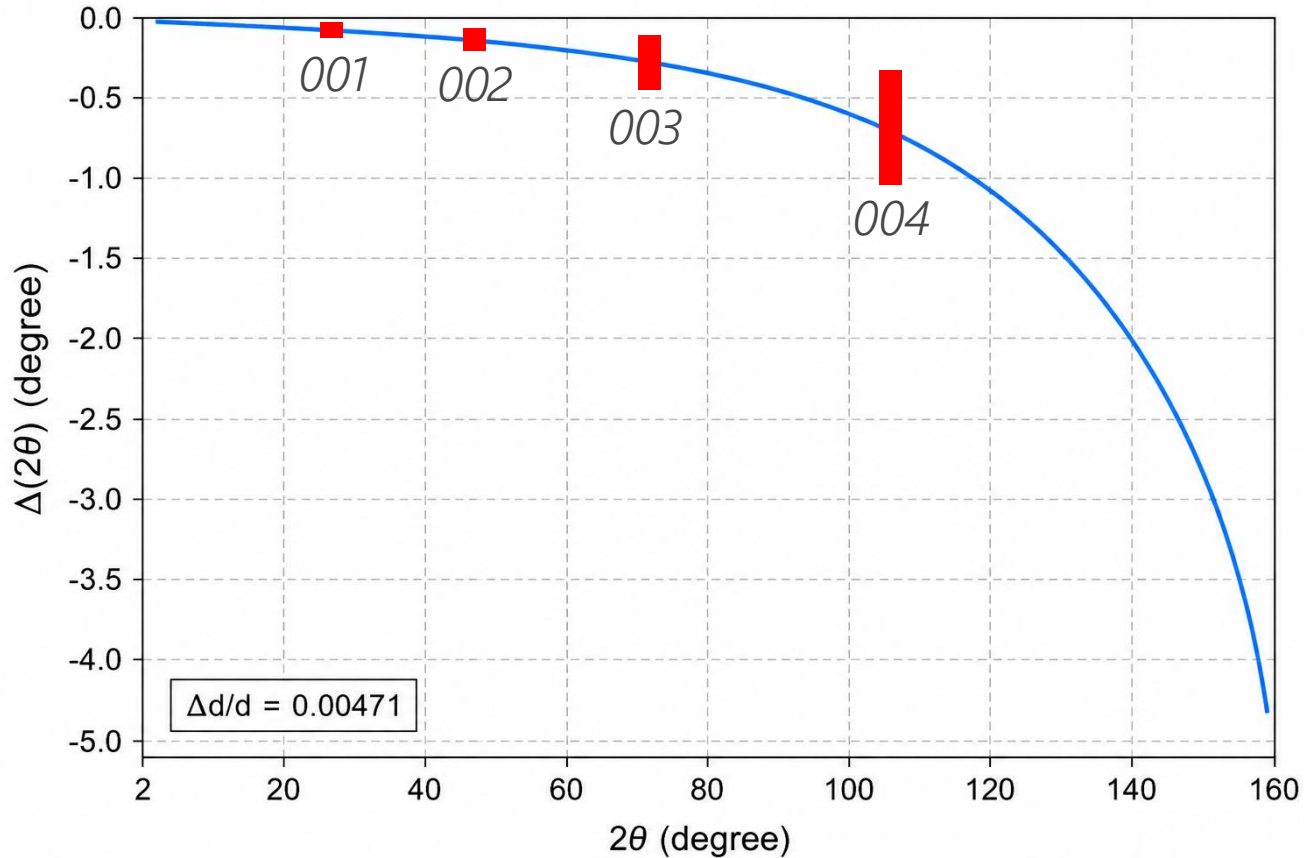
$C_L$ : layer's c-lattice constant

$C_S$ : substrate's c-lattice constant

$\theta$ : Bragg angle

$\lambda$ : X-ray wavelength (0.15406 nm)

# Peak separation layer & substrate



- 1<sup>st</sup> derivative Bragg equation

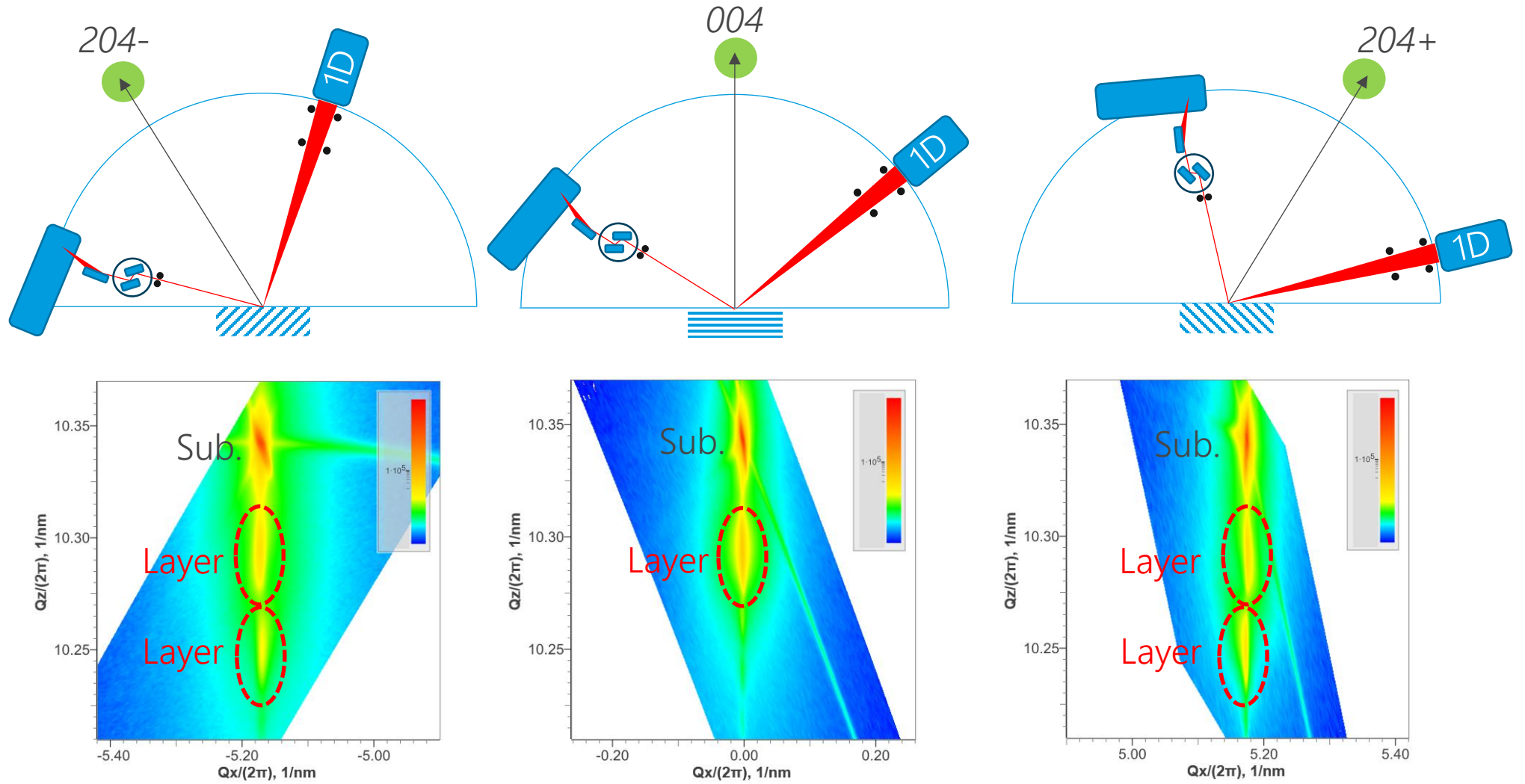
$$\frac{\Delta d}{d} = -\cot \theta \cdot \Delta \theta$$

- Peak separation  $\Delta \theta$

$$\Delta 2\theta = -2 \cdot \tan \theta \cdot \frac{\Delta d}{d}$$

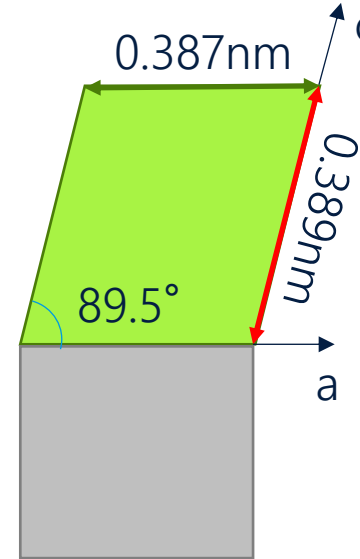
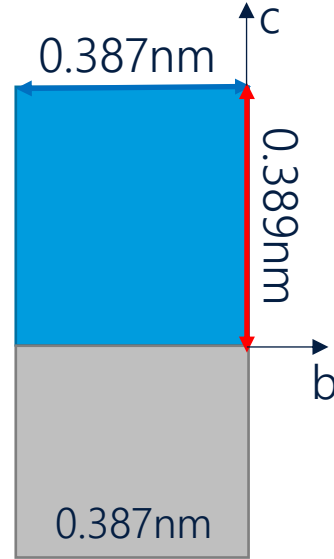
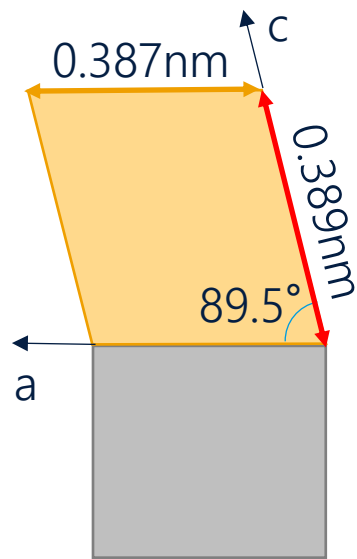
$$\frac{\Delta d}{d} = M = 0.00471$$

# RSMs

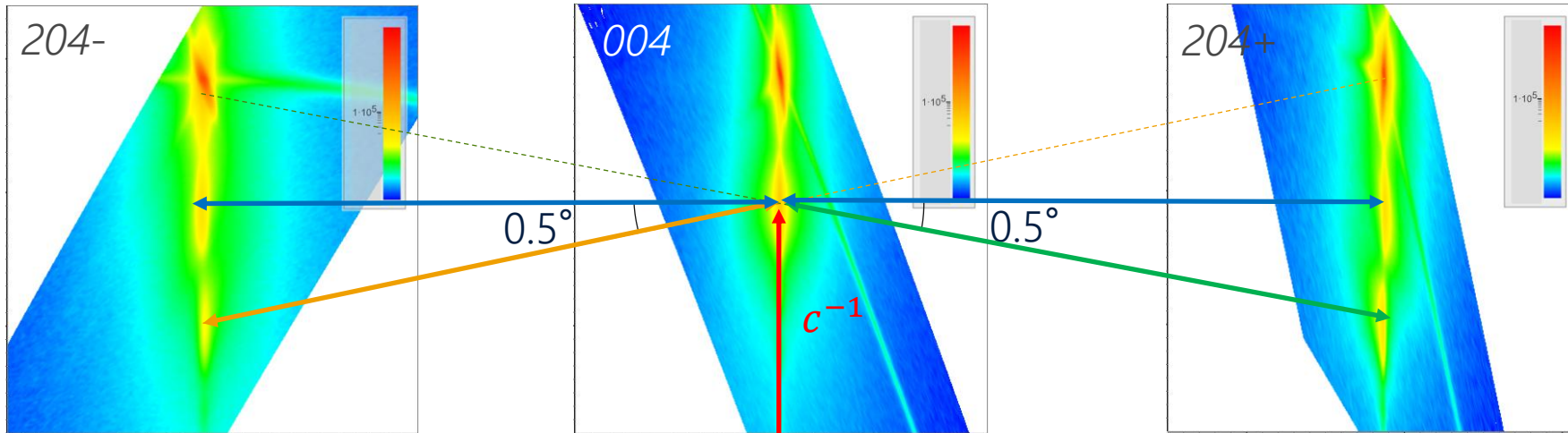


L5BO layer

LSAT sub.



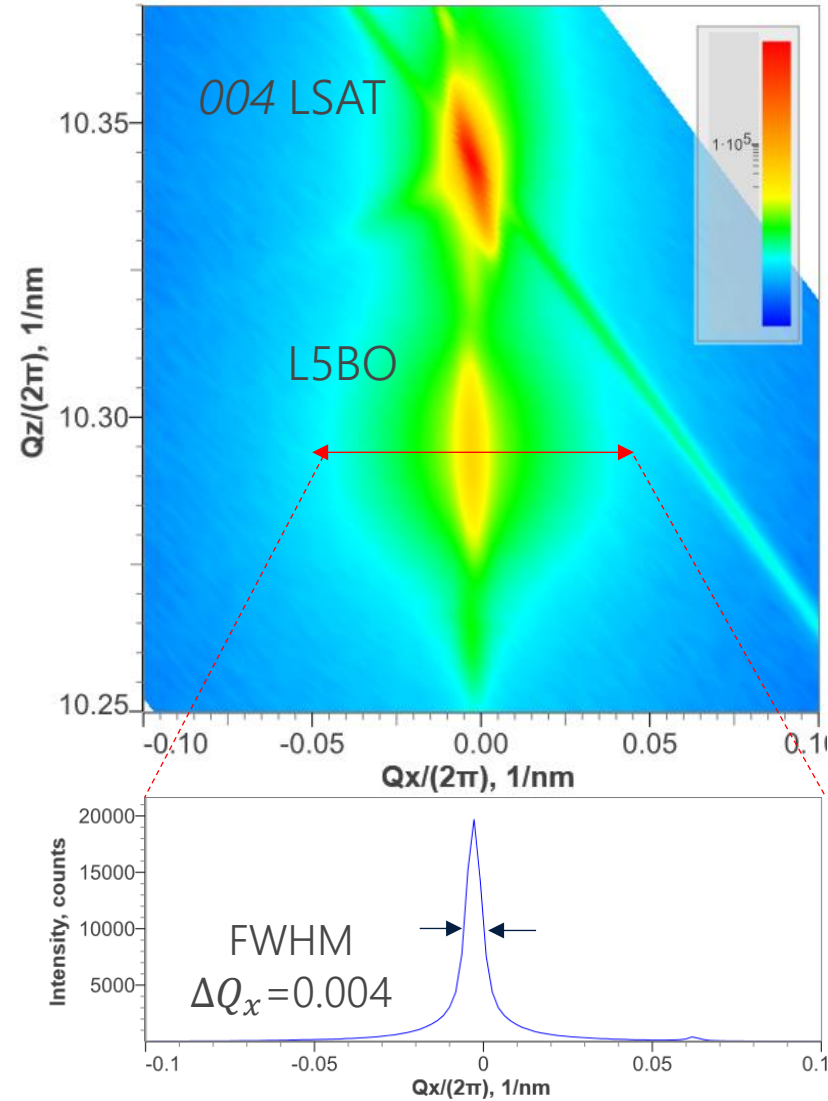
- Monoclinic symmetry
- $a=b=0.387$  nm
- $c=0.389$  nm
- $\beta=89.5^\circ$



# Rocking curve from RSM

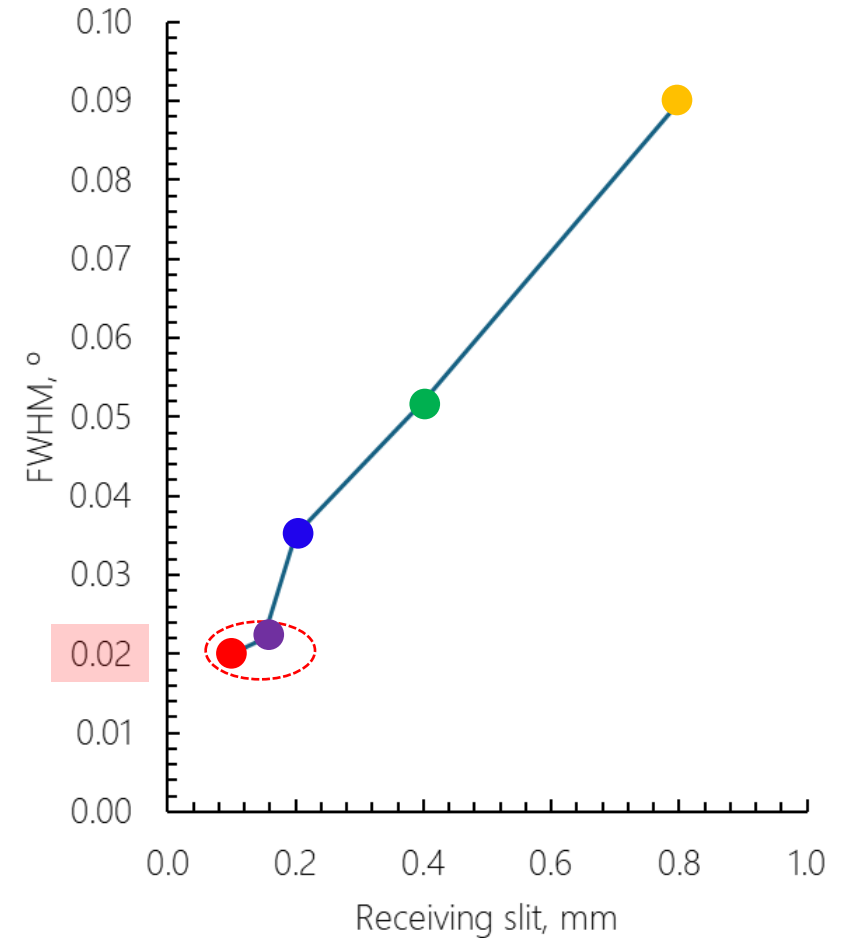
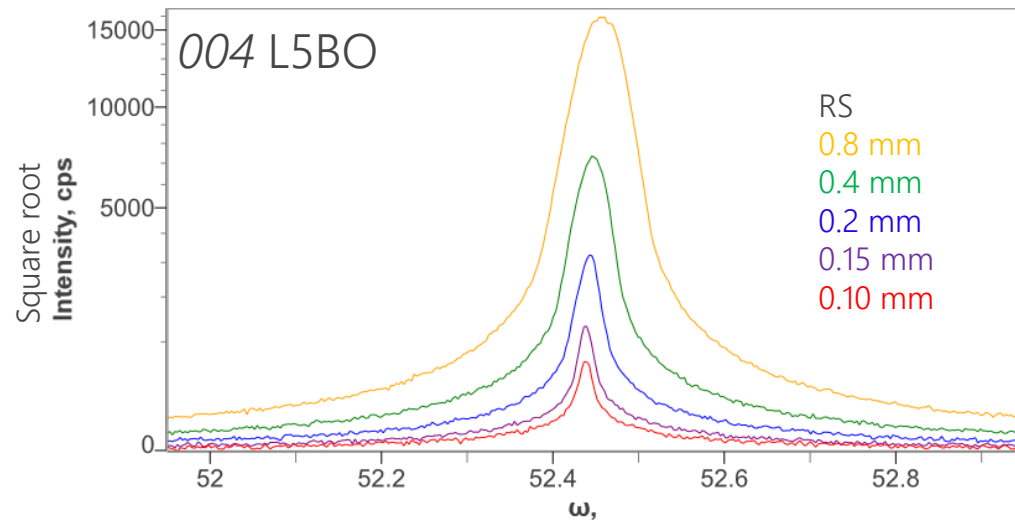
- Conversion of  $\Delta Q_x$  to rocking curve FWHM ( $\Delta\omega$ ).

$$\Delta\omega \sim \frac{\Delta Q_x}{Q_z} = \frac{0.004}{10.29} = 0.02^\circ$$



# XRC: receiving slit dependence

- Wider receiving slits convolves  $\Delta Q_z$  with  $\Delta Q_x$ .
- Narrow receiving slit is required to measure the real XRC peak width  $\sim 0.02^\circ$ .



# Summary

- $2\theta$ - $\theta$  scan
  - Lattice mismatch  $\perp$  to sample surface
  - Thickness from thickness fringes
- Reciprocal space mapping
  - Lattice mismatch both  $\perp$  &  $\parallel$  to sample surface
  - Crystal symmetry & domain structure determination
  - Rocking curve extraction